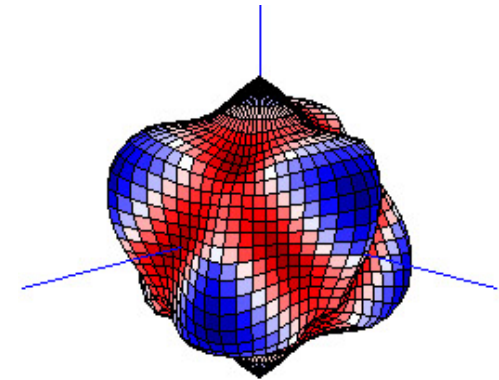
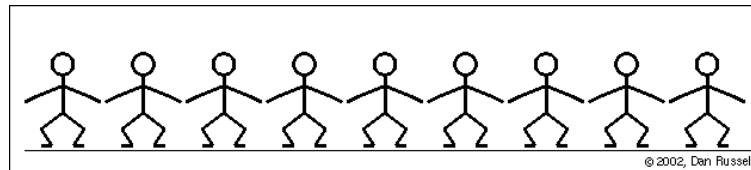


# Anisotropic physical properties of rocks

## Single to Polycrystals

### 2010



David Mainprice  
UMR Géosciences, Université Montpellier 2  
France



Number of students falling a sleep



# *Plan of this Talk*

*Introduction*

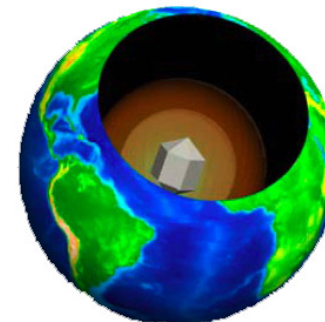
*2<sup>nd</sup> Rank Tensors - Thermal Conductivity, Thermal Diffusivity and Thermal expansion*

*4<sup>th</sup> Rank Tensors - Elasticity, Seismic Velocity*

# Introduction

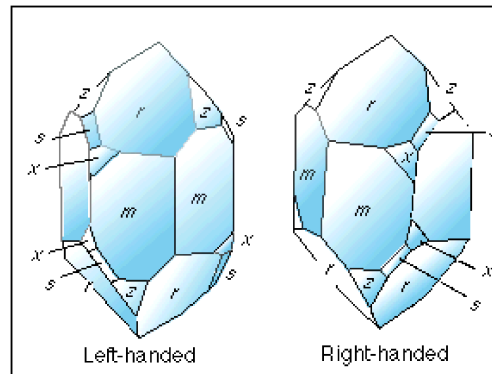
# Why are we interested in Single Crystals ?

- To understand the anisotropic physical properties of polycrystalline rocks caused by **crystal preferred orientation (CPO)** it is important to know the about the simplest case, the single crystal. CPO not **lattice** preferred orientation LPO !
- The single orientation (single crystal), has a perfectly defined ODF (orientation distribution function), PFs (pole figure) or IPFs (inverse pole figure).
- To understand the how crystal symmetry, sample symmetry, CPO and single crystal properties combined to produce anisotropic rock properties ...



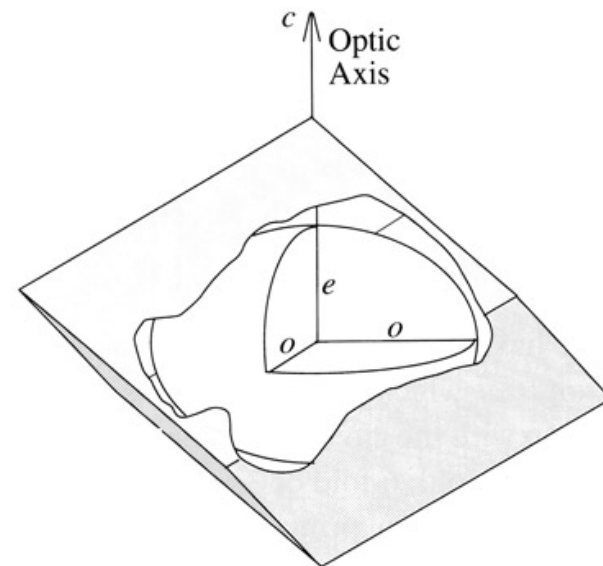
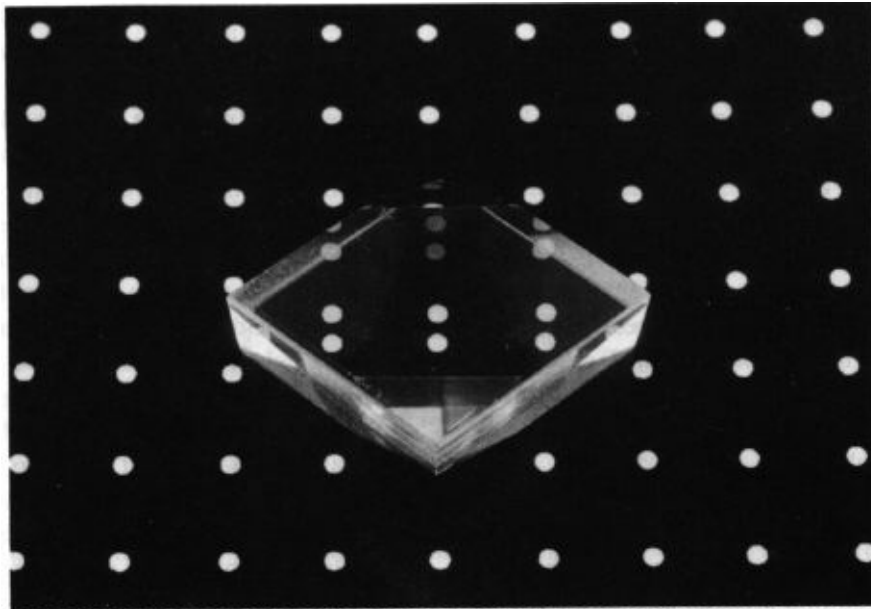
# Physical properties of crystals

- Thermal conductivity and diffusivity ( 2<sup>th</sup> rank tensor) → can be calculated from CPO
- Thermal expansion ( 2<sup>th</sup> rank tensor) → can be calculated from CPO
- Electrical conductivity, electrical polarization and dielectric properties) → can be calculated from CPO, BUT may not be relevant if conductivity controlled by high conductivity phases in the grain boundaries (e.g. water, graphite or carbon)
- Piezoelectricity ( 3<sup>rd</sup> rank tensor) → can be calculated from CPO, if we can determine the CPO of the Left- and Right-handed crystals...



- Elasticity ( 4<sup>th</sup> rank tensor) → seismic (elastic) properties, can be calculated from CPO

# Anisotropic Properties

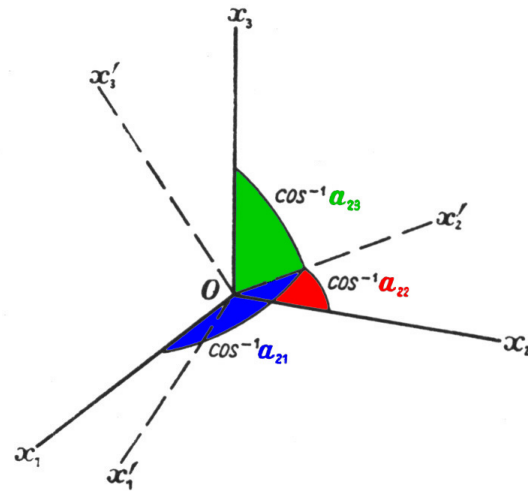


Calcite optical properties : 2<sup>nd</sup> Rank Tensor

# Tensor rank of physical properties

Physical Property (rank)	Driving Force (rank)	Response (rank)
Density (0)	Mass (0)	Volume (0)
Pyroelectricity (1)	Temperature (0)	Electric Field (1)
Electric conductivity (2)	Electric Field (1)	Electric Current Density (1)
Electric Permittivity (2)	Electric Field (1)	Dielectric Displacement (1)
Dielectric Susceptibility (2)	Electric Field (1)	Polarization (1)
Chemical Diffusivity (2)	Potential Gradient -ve (1)	Chemical Flux (1)
Thermal Conductivity (2)	Temperature Gradient -ve (1)	Heat Flux (1)
Thermal Expansion (2)	Temperature (0)	Strain (2)
Magnetic Susceptibility (2)	Magnetic Field (1)	Magnetisation Intensity (1)
Magnetic Permeability (2)	Magnetic Field (1)	Magnetic Induction (1)
Piezoelectricity (3)	Electric Field (1)	Strain (2)
Elastic Compliance (4)	Stress (2)	Strain (2)
Elastic Stiffness (4)	Strain (2)	Stress (2)

# Transformation of axes



Transformation of axes.

		"old" axes	
	x1	x2	x3
x'1	a11	a12	a13
"new" x'2	a21	a22	a23
x'3	a31	a32	a33

$a_{ij}$  is a 3 by 3 transformation matrix relating two orthogonal reference axes ( $x$  and  $x'$ ).

**A major concept in defining the properties of tensors.**

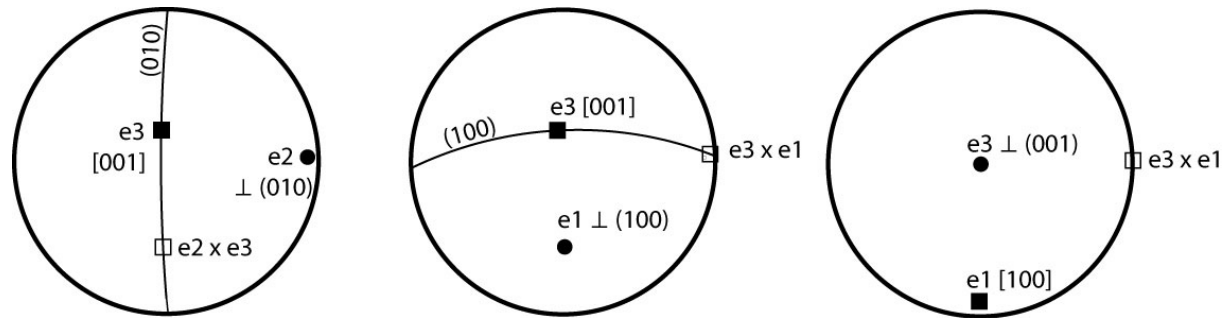


# Cartesian Reference Frame for Tensors I

- Measurement of anisotropic physical properties are reported in the literature as components of tensors, typically as tabulated values. The reference frame most commonly used is a **right-handed Cartesian** (also called orthonormal) system.
- For cubic, tetragonal and orthorhombic the obvious choice is to use the orthogonal lattice basis vectors  $\mathbf{a}[100]$ ,  $\mathbf{b}[010]$  and  $\mathbf{c}[001]$  of the crystal axes. However, for most general case of triclinic crystal symmetry where  $\mathbf{a}$ ,  $\mathbf{b}$ , and  $\mathbf{c}$  are not orthogonal, there are many possible choices and no general convention.
- The choice of a specific reference frame is often guided by presence of cleavage or well developed crystal faces that allow for an easy determination of the crystal orientation.

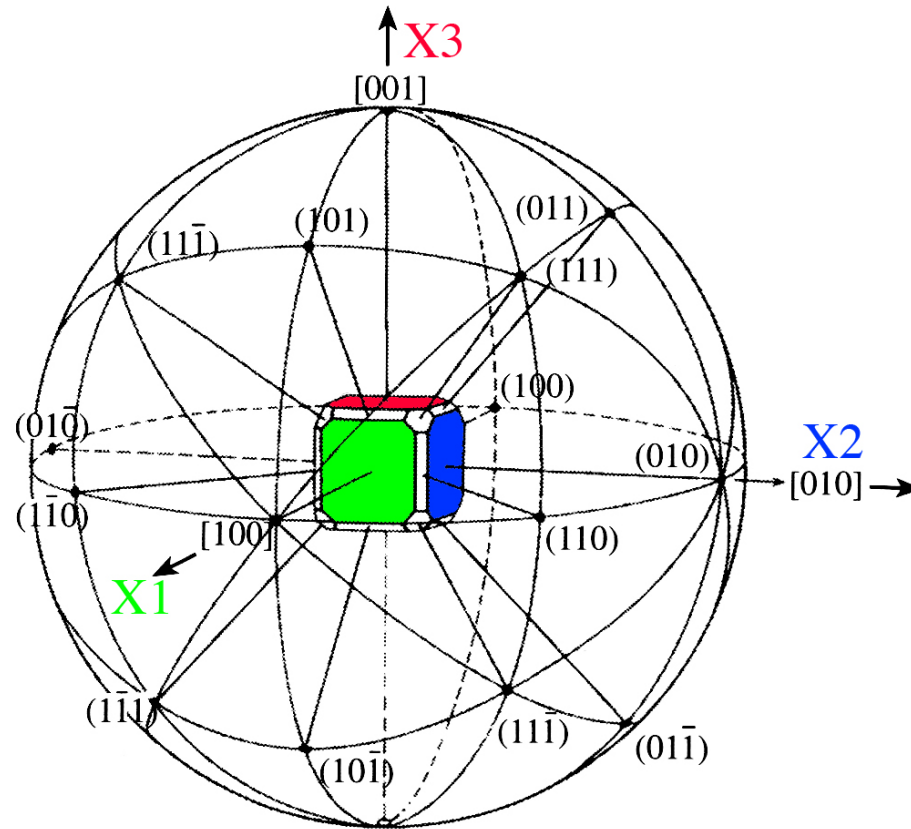
# Cartesian Reference Frame for Tensors II

- Here are 3 common choices, for the tensor Cartesian reference frame  $(X1, X2, X3)$  ;
- a)  $X3 = \mathbf{c}[001]$ ,  $X2 = \mathbf{b}^* \perp (010)$  and hence for a right-handed system  $X1 = X2 \times X3$
- b)  $X3 = \mathbf{c}[001]$  ,  $X1 = \mathbf{a}^* \perp (100)$  and  $X2 = X3 \times X1$
- c)  $X3 = \mathbf{c}^* \perp (001)$  ,  $X1 = \mathbf{a}[100]$  and  $X2 = X3 \times X1$



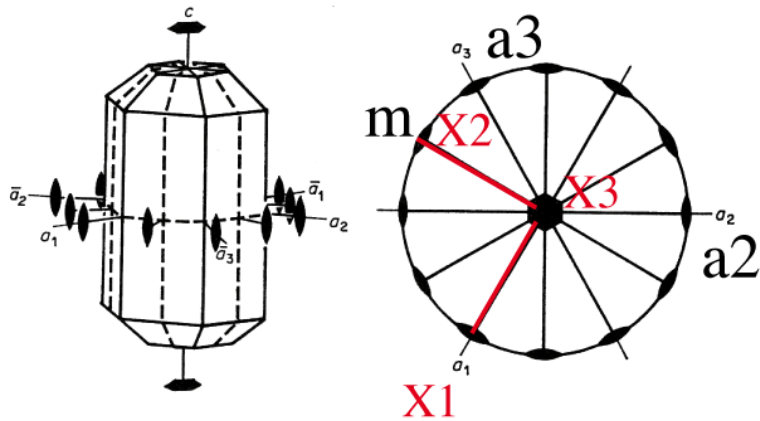
Reference Frames a), b) and c) for triclinic plagioclase Labradorite An66  
 (  $\mathbf{a}=0.817$  nm  $\mathbf{b}=1.287$  nm  $\mathbf{c}=1.420$  nm  $\alpha=93.46^\circ$   $\beta=116.09^\circ$   $\gamma=90.51^\circ$  )

# Cubic crystal

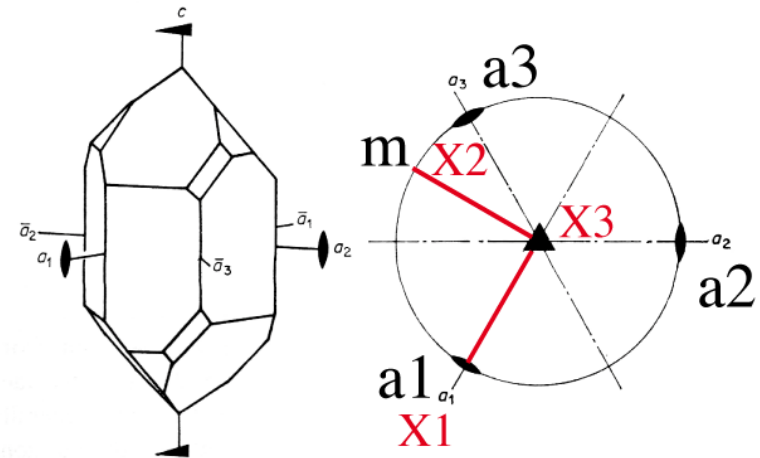


# Hexagonal and trigonal symmetries

$6/mmm$

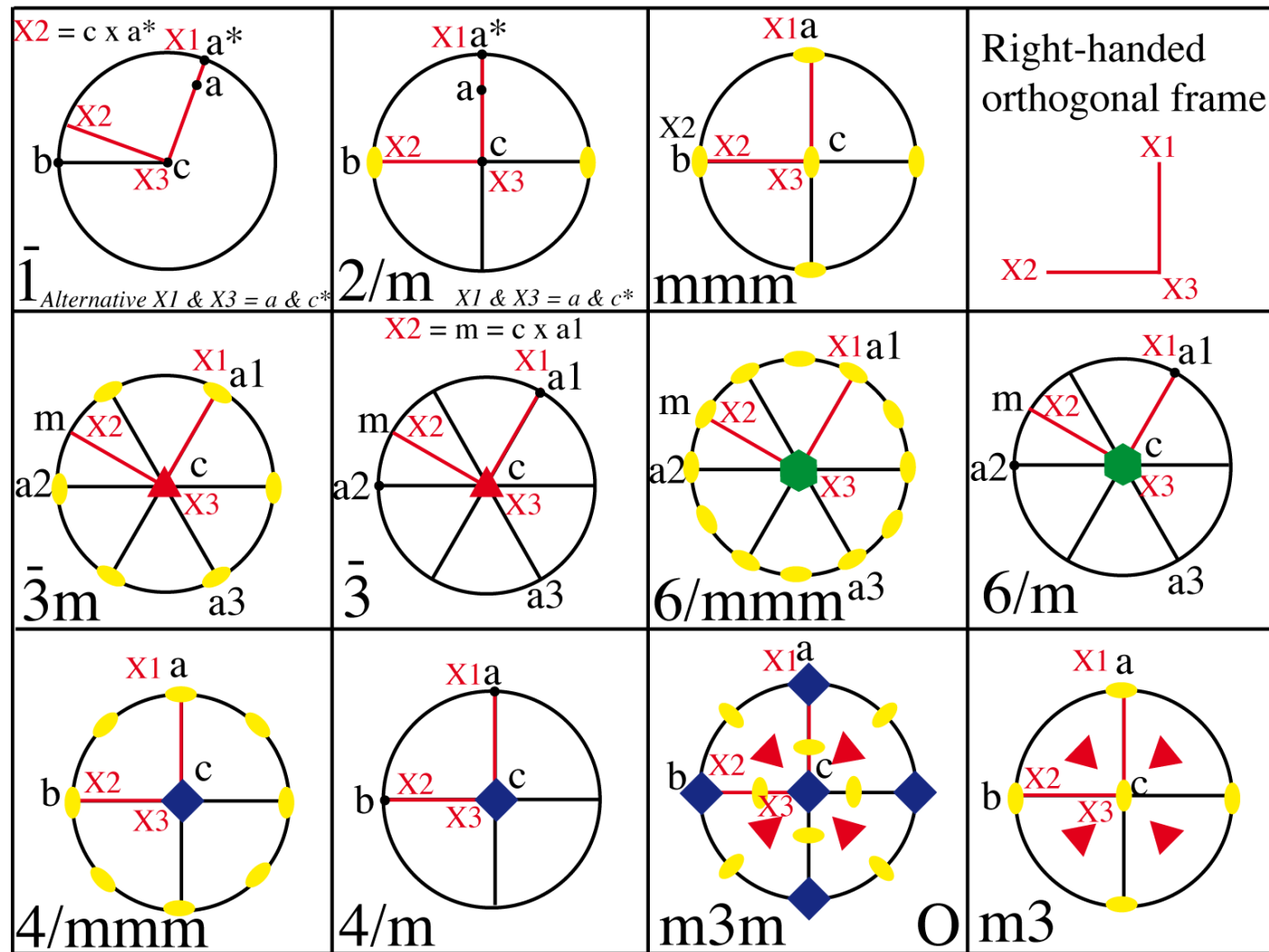


$\bar{3}m$



$$X2 = m = c \times a1$$

# Tensor reference frames for all Laue classes



$a = [100]$   $a^* = \perp(100)$  etc

# Cartesian Reference Frame for Tensors III

- Examples of reference frames for elastic constants, **but check for yourself !**  
a,b/b\*,c\* (monoclinic alkali feldspar)  
a\*,b/b\*,c (monoclinic diopside, augite, hornblende, muscovite, antigorite)  
a,(c\* x a),c\* (triclinic plagioclase, antigorite)

# Tensor and Euler frames

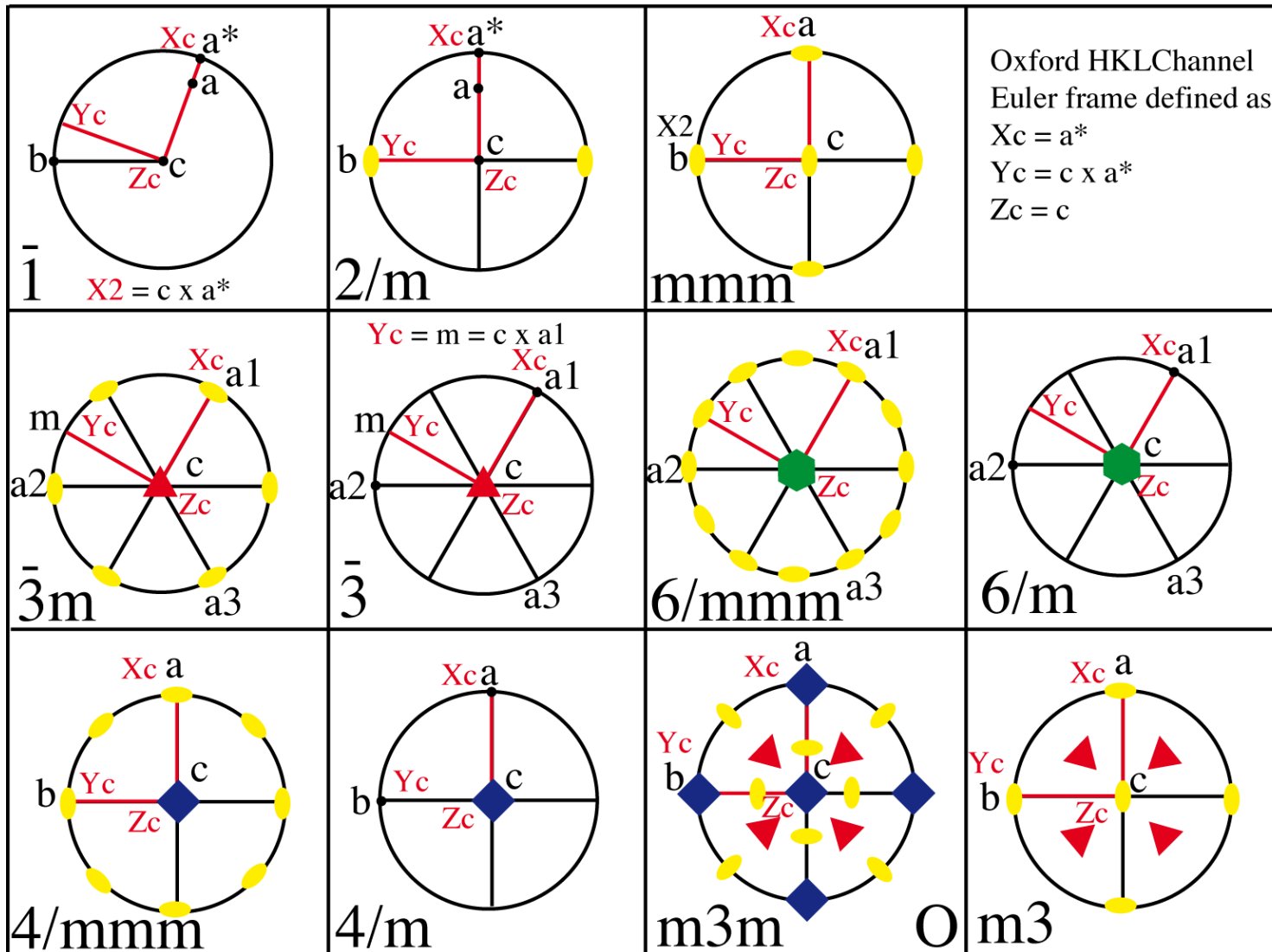
- You should also be aware that the tensor Cartesian reference frame ( $X_1, X_2, X_3$ ) used for a tensor may not be the same as the reference frame with respect to the crystal axes used for the Euler angles ( $X_c, Y_c, Z_c$ ) to define your orientation data.
- For example the commonly used BearTex Texture package uses  $Y_c = b^* \perp (010)$ ,  $Z_c = c[001]$  and  $X_c = Y_c \times Z_c$  and the EBSD software from Oxford HKL uses  $X_c = a^* \perp (100)$ ,  $Y_c = Z_c \times X_c$ ,  $Z_c = c[001]$ , and the TSL EBSD package uses different axes for each crystal symmetry !
- In general a rotation (or transformation) matrix ( $R_{ij}$ ) will be needed to bring the tensor reference frame into the correct orientation for a specific Euler angle frame. For example  $g'(i,j) = g(k,i) \cdot R(k,j)$  where  $g(k,i) = g(\phi_1 \phi_2)$  is Euler angle matrix,  $g'(i,j)$  is orientation matrix of the tensor and  $R(k,j)$  the rotation matrix.

Rules for selecting a crystallophysical coordinate system

Syngony	Orientation with respect to crystallographic axes	Orientation with respect to symmetry elements of crystal
Triclinic	$X_3 \parallel c$ , or $X_2 \parallel b$ , or $X_1 \parallel a$	
Monoclinic	$X_2 \parallel b$ and $X_1 \parallel a$ (or $X_3 \parallel c$ ) Sometimes $X_3 \parallel b$ and $X_1 \parallel a$ (or $X_2 \parallel c$ )	$X_2 \parallel$ to axis 2 or $\perp$ to plane $m$ . Sometimes $X_3 \parallel$ to axis 2 or $\perp$ to plane $m$
Orthorhombic	$X_3 \parallel c$ , $X_2 \parallel b$ , $X_1 \parallel a$	$X_3 \parallel$ to axis 2; $X_1, X_2 \parallel$ to other axes 2 or $\perp$ to planes $m$
Tetragonal	$X_3 \parallel c$ , $X_2 \parallel b$ , $X_1 \parallel a$	$X_3 \parallel$ to axis 4 (or $\bar{4}$ ); $X_1, X_2 \parallel$ to axes 2 or $\perp$ to planes $m$ (if they exist); for class $\bar{4}2m$ usually $X_1, X_2 \parallel$ to axes 2
Trigonal and hexagonal	$X_3 \parallel c$ , $X_1 \parallel a$ . For classes $3m$ and $\bar{6}m2$ sometimes $X_2 \parallel b$	$X_3 \parallel$ to axis 3, $\bar{3}$ , 6 or $\bar{6}$ ; $X_1 \parallel$ to axis 2 (if they exist, except class $\bar{6}m2$ ); for classes $3m$ and $\bar{6}m2$ usually $X_1 \perp$ to plane $m$ , but sometimes $X_2 \perp m$
Cubic	$X_3 \parallel c$ , $X_2 \parallel b$ , $X_1 \parallel a$	$X_1, X_2, X_3 \parallel$ to three mutually perpendicular axes 4 (or $\bar{4}$ ), and if they do not exist, then to similar axes 2



# Euler reference frames for all Laue classes



# Transformation laws for tensors

	Rank of		
Name	Tensor	New in terms of old	Old in terms of new
Scalar	0	$X' = X$	$X' = X$
Vector	1	$P'_i = A_{ij}P_j$	$P_i = A_{ji}P'_j$
2 <sup>nd</sup> Rank	2	$T'_{ij} = A_{ik}A_{jl}T_{kl}$	$T_{ij} = A_{ki}A_{lj}T'_{kl}$
3 <sup>rd</sup> Rank	3	$T'_{ijk} = A_{il}A_{jm}A_{kn}T_{lmn}$	$T_{ijk} = A_{li}A_{mj}A_{nk}T'_{lmn}$
4 <sup>th</sup> Rank	4	$T'_{ijkl} =$ $A_{im}A_{jn}A_{ko}A_{lp}T_{mnop}$	$T_{ijkl} =$ $A_{im}A_{nj}A_{ok}A_{pl}T'_{mnop}$

## Difference between a transformation matrix ( $A_{ij}$ ) and a 2<sup>nd</sup> rank tensor ( $T_{ij}$ )

$A_{ij}$  is a 3 by 3 matrix relating two (right-handed) reference frames [orthogonal matrix  $A^t.A = A.A^t = I$ ,  $A^t = A^{-1}$  and  $\text{Det}(A) = +1$  where the rows and columns are orthogonal (orthonormal) unit vectors.] (e.g. rotation or orientation matrix) N.B. when  $A_{ij}$  is relating right-handed to left-handed reference frames  $\text{Det}(A) = -1$ .

$T_{ij}$  is a physical quantity (e.g. 2<sup>nd</sup> rank tensor) that for a given set of reference axes is represented by 9 numbers (a 3 by 3 table).

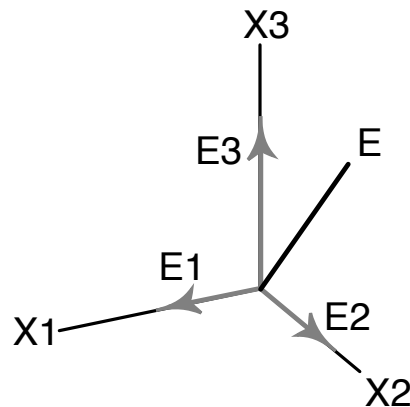
# Zero and First Rank Tensors

- **Zero Rank Tensor ( 1 component)**

Scalars – example density (kg/m<sup>3</sup>)

- **First Rank Tensor (3 components)**

Vectors – example electric field  $E = [E_1 \ E_2 \ E_3] = E_i$



# 2<sup>nd</sup> Rank Tensors

## 2<sup>nd</sup> Rank Tensors - important for geophysics

- Typically relates 2 vectors – for example thermal conductivity : applied vector (negative) temperature gradient and resulting vector heat flow density, exception thermal expansion relates temperature (0) and strain (2).

$$\mathbf{T}_{ij} = \begin{bmatrix} T_{11} & T_{12} & T_{13} \\ T_{21} & T_{22} & T_{23} \\ T_{31} & T_{32} & T_{33} \end{bmatrix}$$

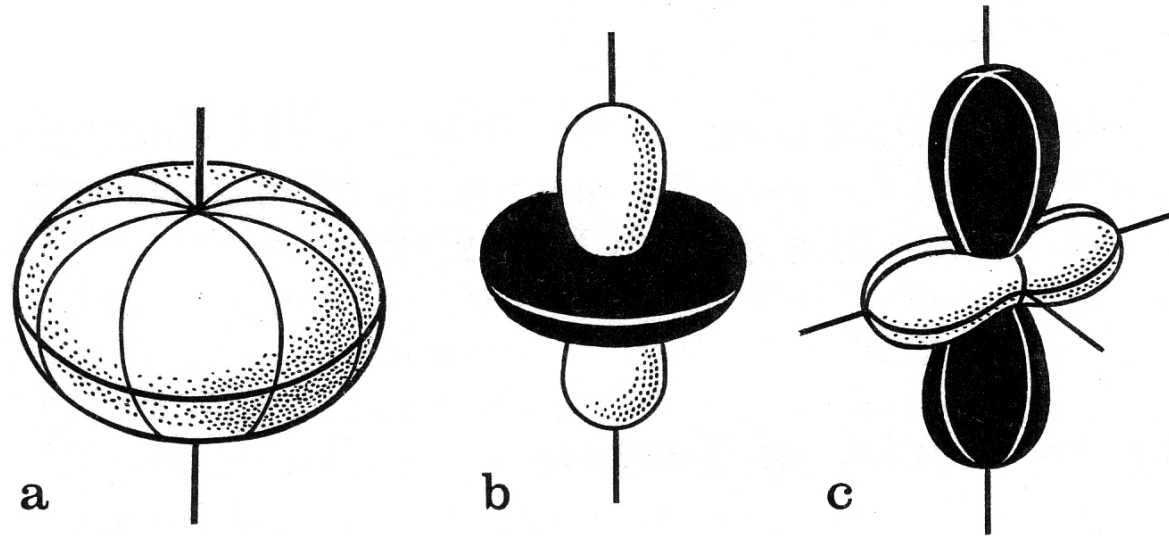
9 components

- The generic 2<sup>nd</sup> rank tensor  $\mathbf{T}$  is the relation between an applied vector  $\mathbf{p}$  and resultant vector  $\mathbf{q}$ . We can write relation between  $\mathbf{p}$  and  $\mathbf{q}$  as a tensor equation

$$\mathbf{p} = \mathbf{T} \mathbf{q} \quad \text{or} \quad p_i = T_{ij} q_j \quad (i=1,2,3 ; j=1,2,3)$$

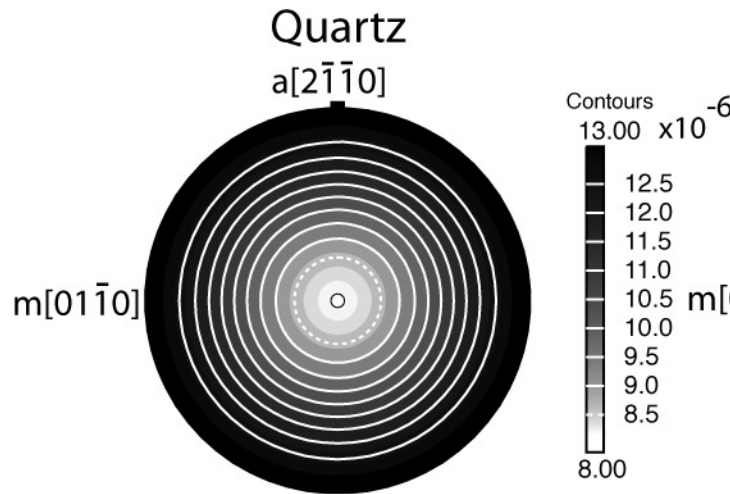
- In general the vectors  $\mathbf{p}$  and  $\mathbf{q}$  are not parallel.

# Thermal Expansion

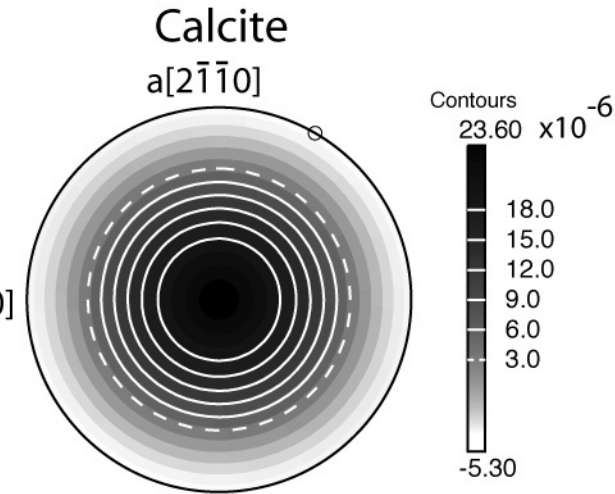


Examples of indicatrics surfaces of the tensor of thermal expansion  $\alpha_{ij}$  at various relationships among the signs of the main components of the tensor  $\alpha_1$ ,  $\alpha_2$ , and  $\alpha_3$ : (a) uniaxial crystals with  $\alpha_3 > 0$  and  $\alpha_1 > 0$ , surface symmetry  $\infty/mmm$ ; (b) uniaxial crystals with  $\alpha_3 > 0$  and  $\alpha_1 < 0$ , surface symmetry  $\infty/mmm$ ; (c) biaxial crystals with  $\alpha_1 \neq \alpha_2 \neq \alpha_3$ ;  $\alpha_1, \alpha_2 > 0$ , while  $\alpha_3 < 0$ ; surface symmetry  $mmm$ . The regions corresponding to  $\alpha > 0$  are white, and those corresponding to  $\alpha < 0$  are black

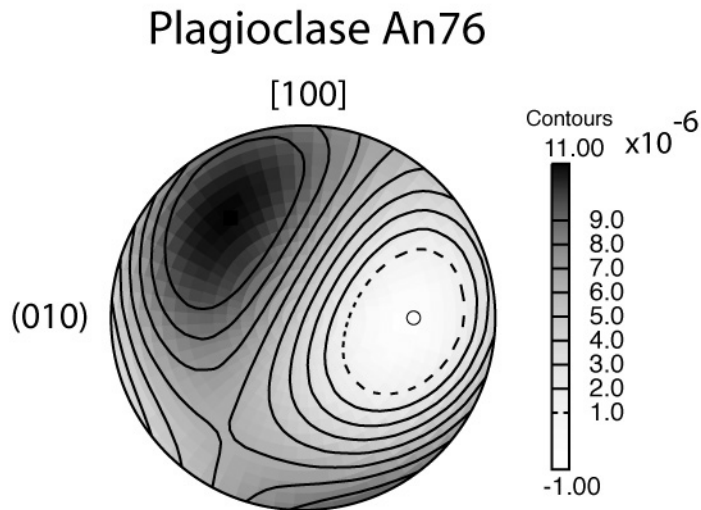
# Single Crystal Thermal Expansion



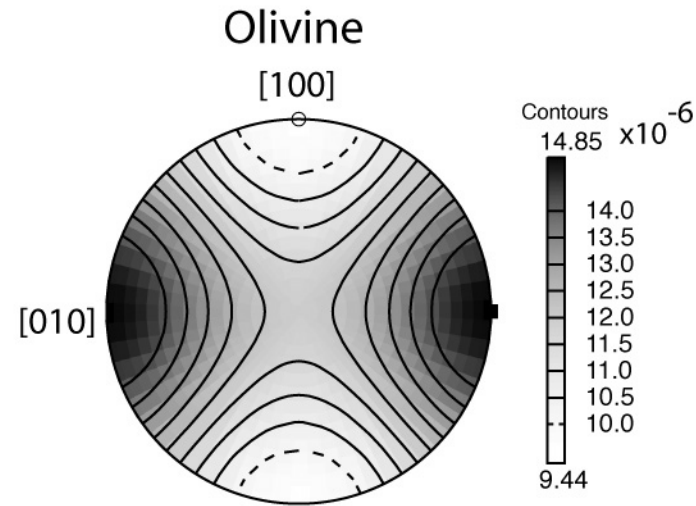
■ Max. Value = 13.00    ○ Min. Value = 8.00  
Anisotropy = 47.6 %  
Eigenvalues 13.0000 13.0000 8.0000



■ Max. Value = 23.60    ○ Min. Value = -5.30  
Anisotropy = 315.8%  
Eigenvalues 23.6000 -5.3000 -5.3000

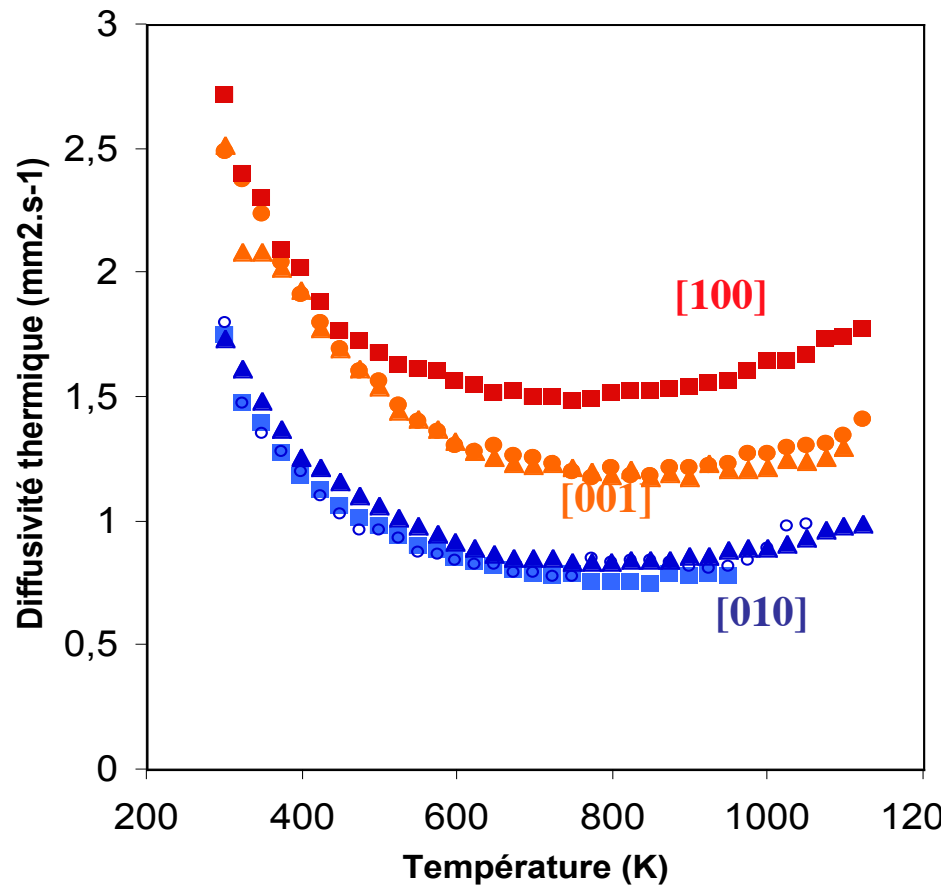


■ Max. Value = 11.00    ○ Min. Value = -1.00  
Anisotropy = 239.9%  
Eigenvalues 10.9970 6.0000 -0.9967



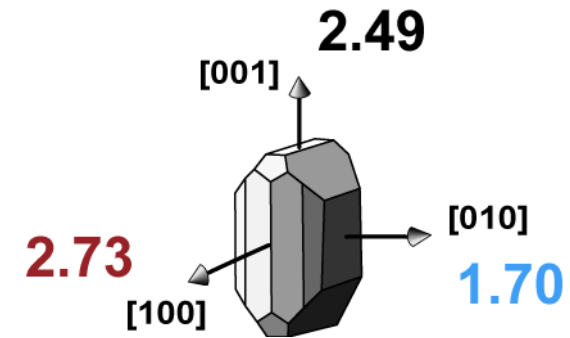
■ Max. Value = 14.85    ○ Min. Value = 9.44  
Anisotropy = 44.6 %  
Eigenvalues 9.4350 11.8080 14.8460

# Thermal diffusivity of Olivine



2<sup>nd</sup> order Tensor

$$\begin{vmatrix} 2.73 & 0 & 0 \\ 0 & 1.70 & 0 \\ 0 & 0 & 2.49 \end{vmatrix}$$



Gibert et al, *GRL*, 2003



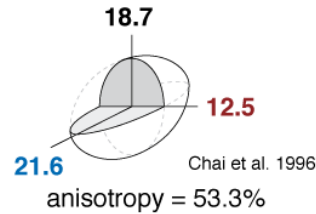
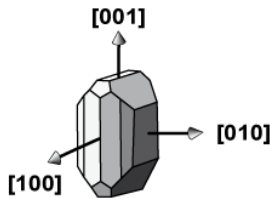
# Anisotropie des propriétés thermiques dans le manteau supérieur

**Modélisation pétrophysique et mesure en laboratoire de l'anisotropie de diffusivité thermique des roches mantelliques**

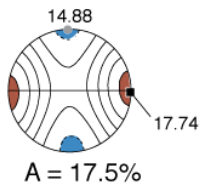
Thèse B. Gibert (2001-2003)

Prix Haüy-Lacroix 2004 SFMC  
coll. U. Seipold & F. Schilling, GFZ Potsdam

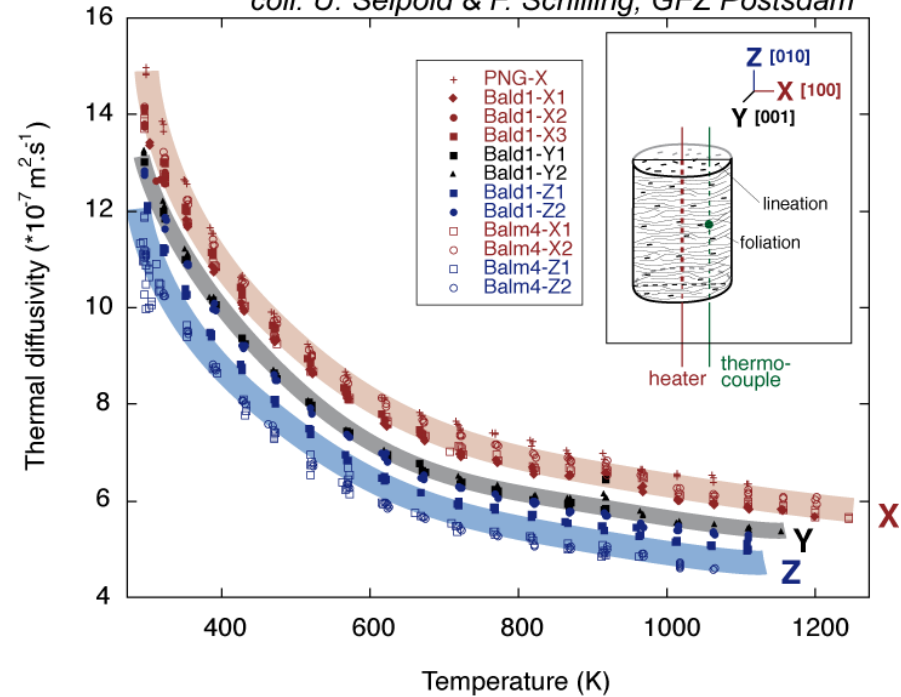
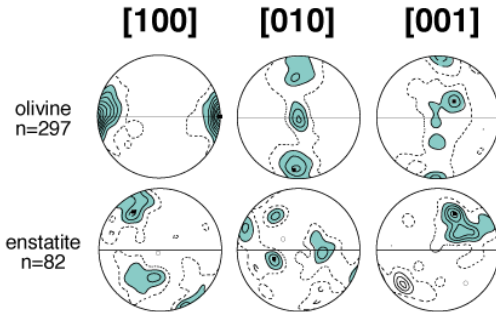
diffusivité thermique du cristal d'olivine



diffusivité thermique ( $\cdot 10^{-7} \text{ m}^2/\text{s}$ )



échantillon BALD1

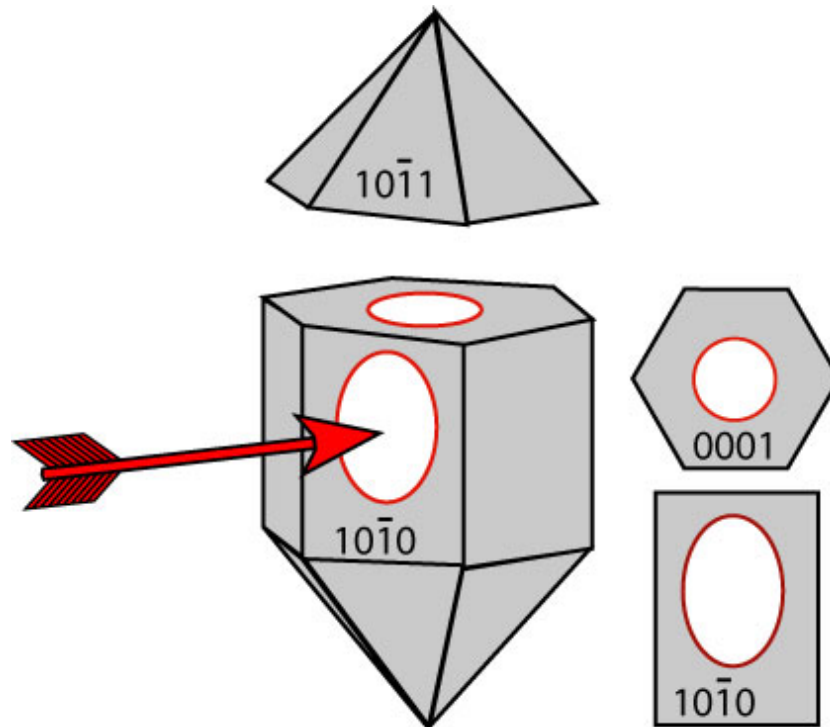


Tommasi, Gibert, Seipold & Mainprice, Nature 2001  
Gibert, Seipold, Tommasi & Mainprice, JGR 2003  
Gibert, Schilling, Tommasi & Mainprice, GRL 2003  
Gibert, Schilling, Gratz & Tommasi, PEPI 2005

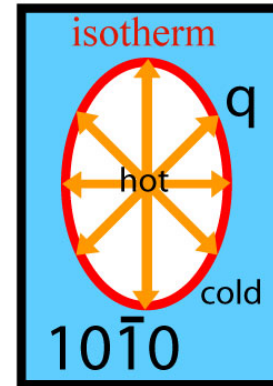
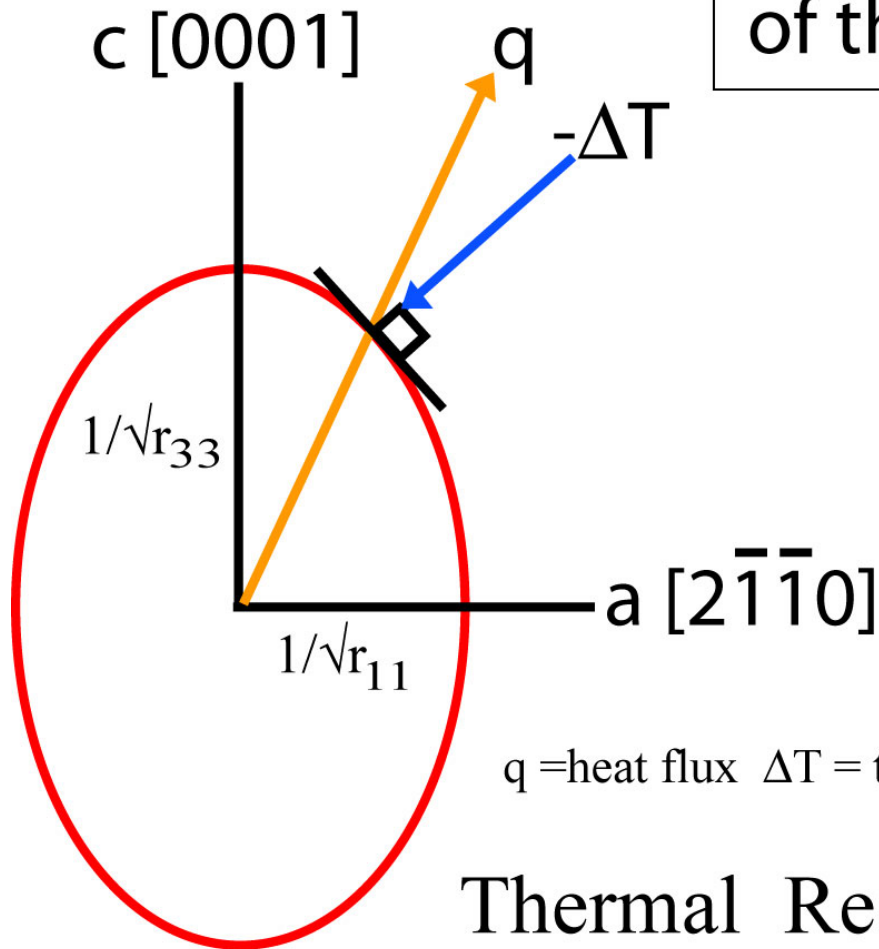


# Radius-normal property and melting wax experiment

- Consider a red hot arrow touching the second order prism plane of a single crystal of quartz and creating a point heat source. A (negative) thermal gradient will result in heat flow away from the hot heat source (arrow tip) to the colder regions of the crystal. As the thermal resistivity (reciprocal of thermal conductivity) varies with direction the heat flow will not be equal in all radial directions, i.e. heat flow is anisotropic.



# Radius-normal property of the representation quadric



Radial heat flux ( $q$ )  
(controlled by crystal properties not boundary conditions)

$q$  = heat flux  $\Delta T$  = thermal gradient

## Thermal Resistivity ( $r$ )

$$\Delta T_i = - r_{ij} q_j$$

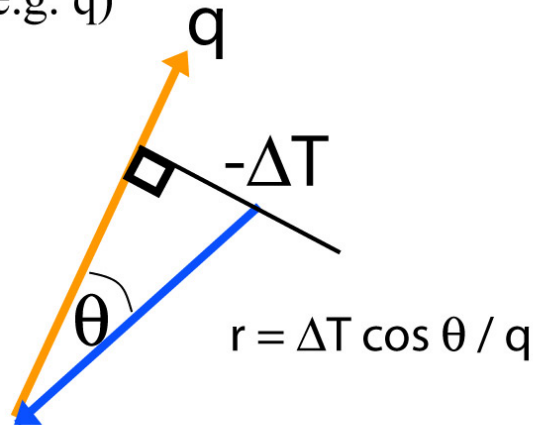
where  $r_{11} = 1/k_{11}$  etc

## Thermal Conductivity ( $k$ )

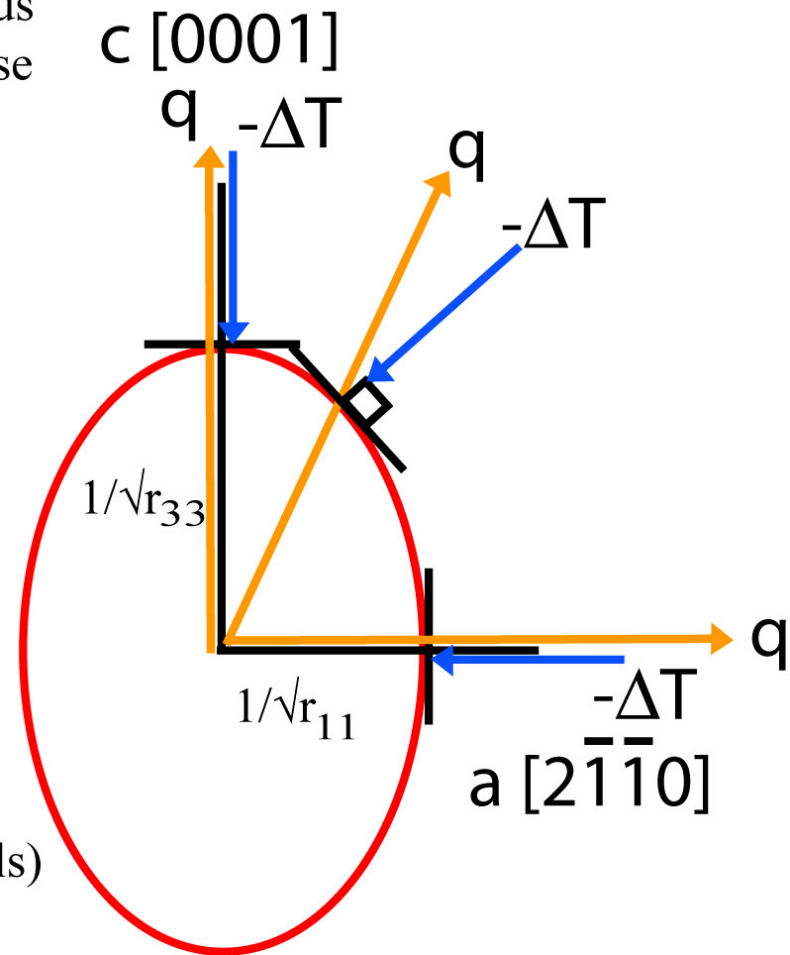
$$q_i = - k_{ij} \Delta T_j$$

# The stimulus vector and response vectors for 2<sup>nd</sup> rank tensors

In general in anisotropic crystals the stimulus vector (e.g.  $\Delta T$ ) is not parallel to the response vector (e.g.  $q$ )



In certain directions (e.g. principal axes) the stimulus vectors and the response vectors may be parallel in anisotropic crystals. It is always the case for isotropic crystals (e.g. 2<sup>nd</sup> rank tensors of cubic crystals)



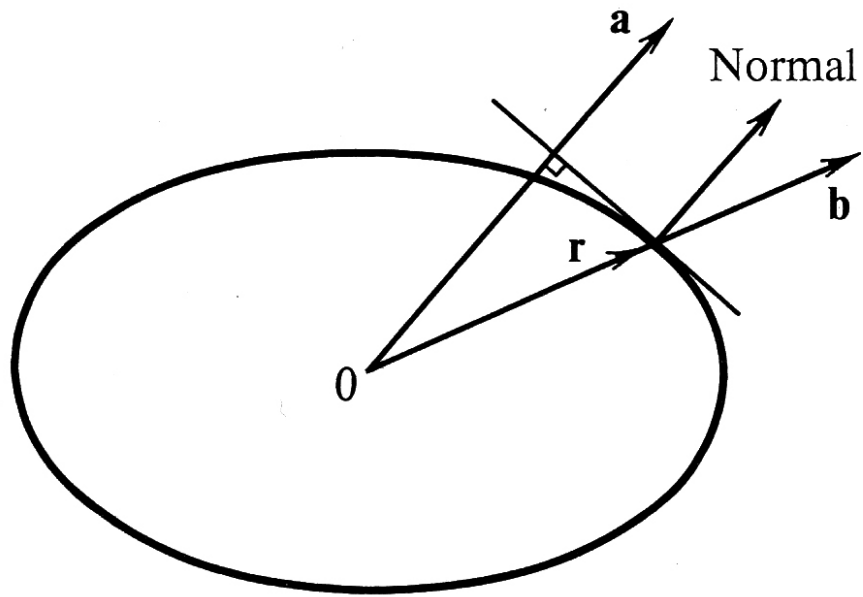
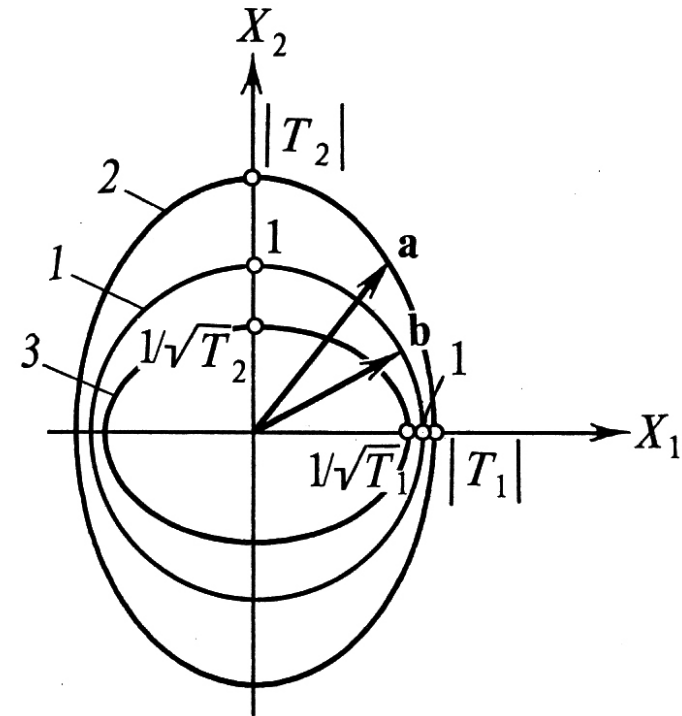


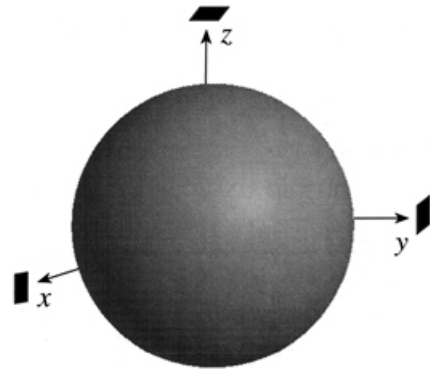
Diagram explaining the properties of a radius vector and normal of the characteristic surface of a symmetric second-rank tensor



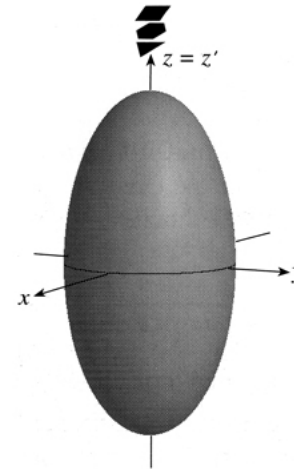
Central sections of a sphere (1) described by a unit vector  $\mathbf{b}$ , the ellipsoid of the values of a symmetric tensor (2) given by the equation  $a_i = T_{ij} b_j$  and of the characteristic surface of this tensor (3)

# The effect of symmetry on symmetric 2<sup>nd</sup> rank tensors

Cubic  
(Sphere)



Tetragonal, Hexagonal, Trigonal  
(Uniaxial Ellipsoid)

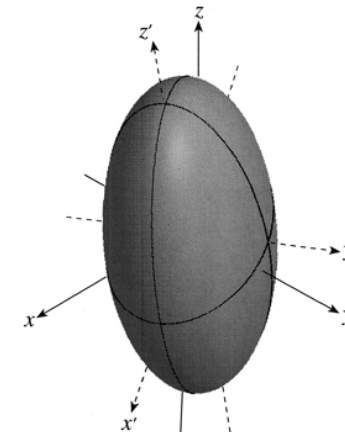
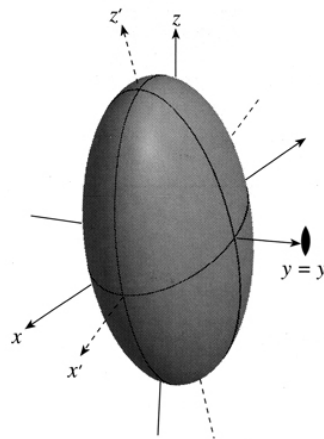
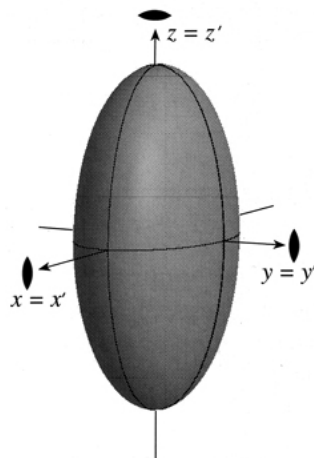


Orthorhombic

Monoclinic

Triclinic

←----- Triaxial Ellipsoids ----->



# Effect of symmetry on Physical Properties : Neumann's Principle

F.E. Neumann's principle (1885) states that "symmetry elements of any physical property of a crystal must include ALL the symmetry elements of the point group of the crystal". This implies that a given physical property may possess a **higher** symmetry than that possessed by the crystal and it cannot be of a **lower** symmetry than that of the crystal. Some physical properties are inherently centrosymmetric (all symmetric second order tensors and elasticity) which will add a center of symmetry in many minerals (e.g. quartz) and result in a **higher** symmetry than the possessed by the crystal.

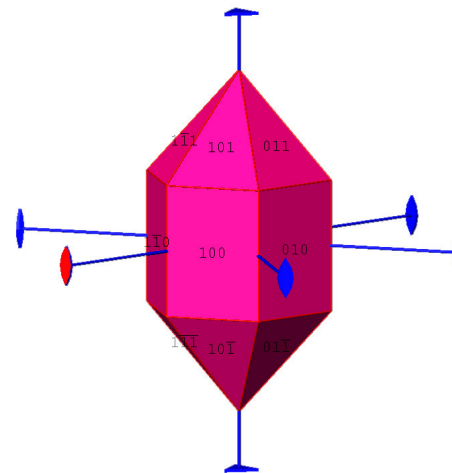
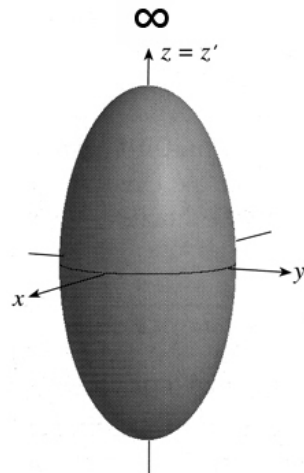


Table 4.2 Number of independent components of physical properties represented by second rank (order) tensors

Crystal system	Orientation of principal axes with respect to the crystal axes	Form of tensor	Number of independent components
Cubic	Any. Representation quadric is a sphere	$\begin{bmatrix} S & 0 & 0 \\ 0 & S & 0 \\ 0 & 0 & S \end{bmatrix}$	1
Tetragonal Hexagonal Trigonal*	$x_3$ parallel to 4, 6, 3 or $\bar{3}$	$\begin{bmatrix} S_1 & 0 & 0 \\ 0 & S_1 & 0 \\ 0 & 0 & S_3 \end{bmatrix}$	2
Orthorhombic	$x_1, x_2, x_3$ parallel to the diads along $x$ -, $y$ -, $z$ -axes	$\begin{bmatrix} S_1 & 0 & 0 \\ 0 & S_2 & 0 \\ 0 & 0 & S_3 \end{bmatrix}$	3
Monoclinic	$x_2$ parallel to the diad along $y$ -axis	$\begin{bmatrix} S_{11} & 0 & S_{13} \\ 0 & S_{22} & 0 \\ S_{13} & 0 & S_{33} \end{bmatrix}$	4
Triclinic	Not fixed	$\begin{bmatrix} S_{11} & S_{12} & S_{13} \\ S_{12} & S_{22} & S_{23} \\ S_{13} & S_{23} & S_{33} \end{bmatrix}$	6

\* A hexagonal cell is used.



# The magnitude of a 2<sup>nd</sup> rank property in a given direction

- Let a given direction ( $\mathbf{x}$ ) be specified by the direction cosines ( $x_1, x_2, x_3$ ). Referred to general axes

$$T(\mathbf{x}) = T_{ij} x_i x_j$$

- From the magnitude  $T(\mathbf{x})$  in a given direction one can calculate the radius vector  $r(\mathbf{x})$  of the representation quadric. When  $T(\mathbf{x})$  is negative, which occurs for example for thermal expansion in calcite, radius vector  $1/\sqrt{T(\mathbf{x})}$ , is an imaginary value.

$$R(\mathbf{x}) = 1/\sqrt{T(\mathbf{x})}$$

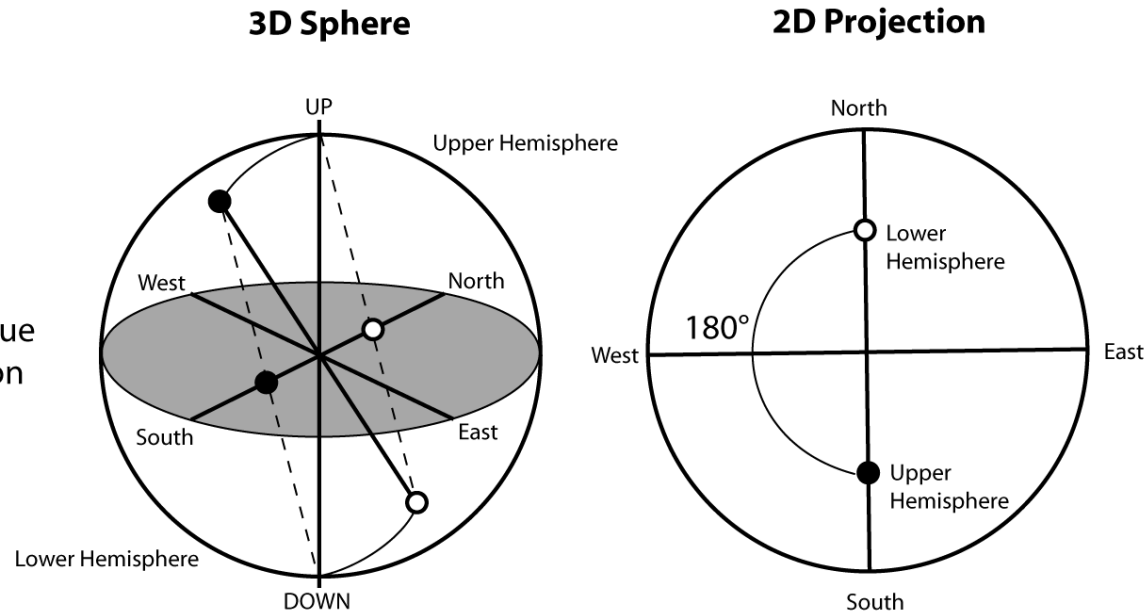
- Although the quadric representation is a hyperboloid when one or more principal axes are negative, the representation is quite straight forward as a contoured equal area projection (pole figure).

# Non-polar and Polar properties

## Non-polar case:

centro-symmetric  
 symmetry  
 property the same value  
 in +ve and -ve direction

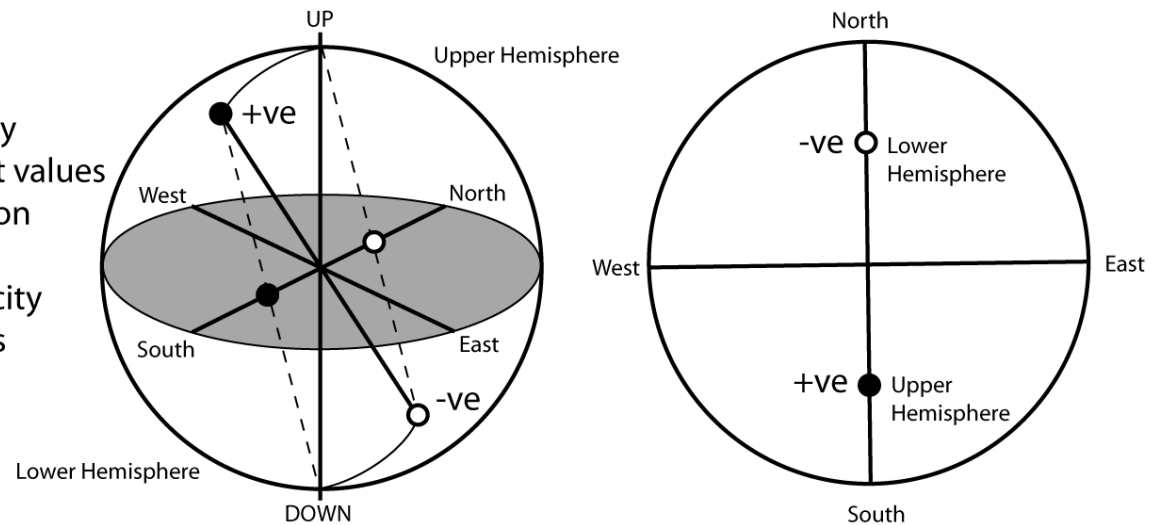
Example elasticity



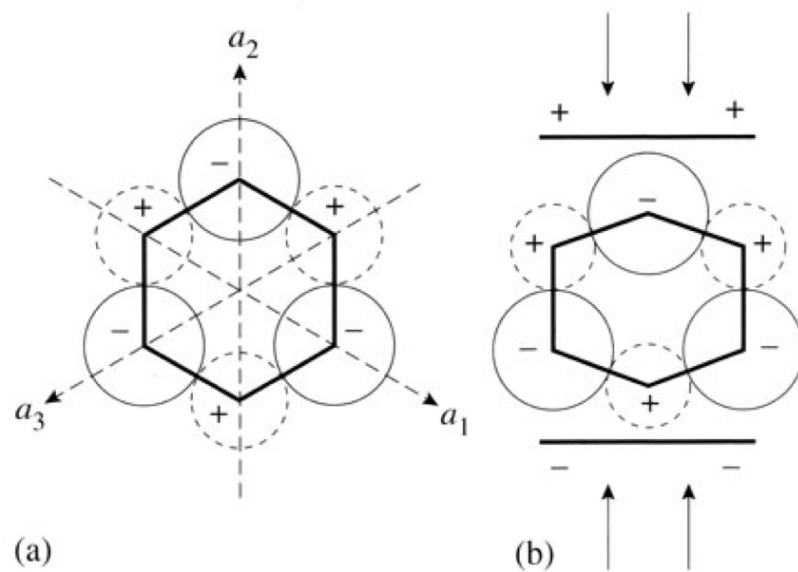
## Polar case:

no centre of symmetry  
 property has different values  
 in +ve and -ve direction

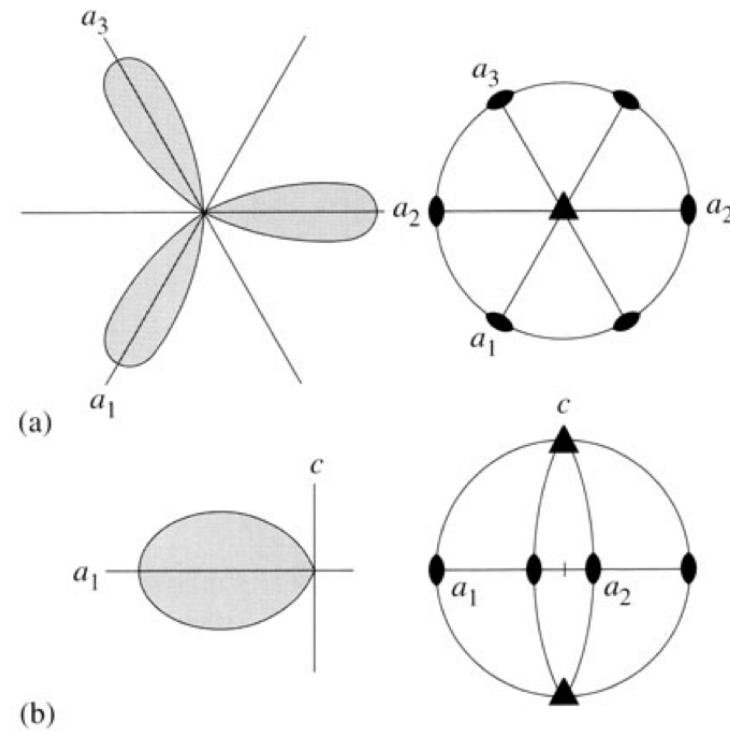
Example piezoelectricity  
 in quartz along a-axes



# Piezoelectricity – 3rd Order Tensor



**Fig. 8.16** Piezoelectricity in quartz. (a) In an undeformed quartz crystal charges of cations ( $\text{Si}^{4+}$ ) and anions ( $\text{O}^{2-}$ ) are balanced. (b) Compression parallel or perpendicular to an  $a$ -axis produces a shift of charges and induces an electric field.

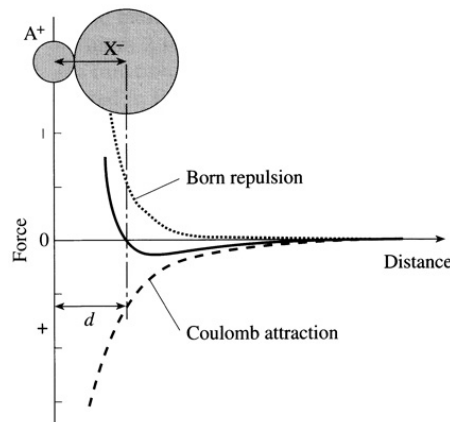


**Fig. 8.17** The representation quadric of the piezoelectric tensor of quartz consists of three lobes that extend parallel to the positive  $a$ -axes.

# 4<sup>th</sup> Rank Tensors

# 4<sup>th</sup> Rank Tensors - important for geophysics

- In any crystalline material there is balance between Coulomb attractive forces between oppositely charge ions and Born repulsive forces due to the overlap of electron shells. At any given thermodynamic state the crystal will tend toward an equilibrium structure.



- For a change in hydrostatic or non-hydrostatic stress, the crystal structure will adjust at the atomic level to the new thermodynamic state.
- The fundamental nature of atomic forces in the determination of elastic properties has been illustrated by the emergence of first principles atomic modeling to predict single crystal elastic tensors of geophysical importance at lower mantle PT conditions.

# Hooke's Law

$$\sigma = c \varepsilon \quad \text{and} \quad \varepsilon = s \sigma$$

where  $c$  = stiffness coefficients (dimensions of stress)

$s$  = compliance coefficients (dimensions of 1/stress)

$\sigma$  = stress tensor (2<sup>nd</sup> order tensor)

$\varepsilon$  = deformation tensor (2<sup>nd</sup> order tensor)

or 
$$\sigma_{ij} = C_{ijkl} \varepsilon_{kl}$$

$$\varepsilon_{ij} = S_{ijkl} \sigma_{kl}$$

$i, j, k, l$  can have the values 1, 2 or 3

so  $3 \times 3 \times 3 \times 3 = 3^4 = 81$  coefficients.

But due to the symmetry of the deformation and stress tensors the 81 coefficients are not independent. In Voigt notation we can write the  $C_{ijkl}$  tensor as 6 by 6 symmetric tensor  $C_{nm}$  with 21 independent values for a triclinic crystal where  $n, m$  can have the values 1, 2, 3, 4, 5 or 6.

# 32 Crystal symmetry classes and their Elastic tensors

Cubic (3) 23,  $m\bar{3}$ , 432,  $\bar{4}3m$ ,  $m\bar{3}m$

$$\begin{bmatrix} c_{11} & c_{12} & c_{12} & 0 & 0 & 0 \\ c_{12} & c_{11} & c_{12} & 0 & 0 & 0 \\ c_{12} & c_{12} & c_{11} & 0 & 0 & 0 \\ 0 & 0 & 0 & c_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & c_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & c_{44} \end{bmatrix}$$

Hexagonal (5) 6,  $\bar{6}$ , 6/m, 622, 6mmm,  $\bar{6}2m$ , 6/mmm

$$\begin{bmatrix} c_{11} & c_{12} & c_{13} & 0 & 0 & 0 \\ c_{12} & c_{11} & c_{13} & 0 & 0 & 0 \\ c_{13} & c_{13} & c_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & c_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & c_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & \frac{1}{2}(c_{11}-c_{12}) \end{bmatrix}$$

Orthorhombic (9) 222,  $mm2$ ,  $mmm$

$$\begin{bmatrix} c_{11} & c_{12} & c_{13} & 0 & 0 & 0 \\ c_{12} & c_{22} & c_{23} & 0 & 0 & 0 \\ c_{13} & c_{23} & c_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & c_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & c_{55} & 0 \\ 0 & 0 & 0 & 0 & 0 & c_{66} \end{bmatrix}$$

Monoclinic (13) 2,  $m$ , 2/m

$$\begin{bmatrix} c_{11} & c_{12} & c_{13} & 0 & c_{15} & 0 \\ c_{12} & c_{22} & c_{23} & 0 & c_{25} & 0 \\ c_{13} & c_{23} & c_{33} & 0 & c_{35} & 0 \\ 0 & 0 & 0 & c_{44} & 0 & c_{46} \\ c_{14} & c_{25} & c_{35} & 0 & c_{55} & 0 \\ 0 & 0 & 0 & c_{46} & 0 & c_{66} \end{bmatrix}$$

Trigonal (6) 32,  $3m$ ,  $\bar{3}m$

$$\begin{bmatrix} c_{11} & c_{12} & c_{13} & c_{14} & 0 & 0 \\ c_{12} & c_{11} & c_{13} & -c_{14} & 0 & 0 \\ c_{13} & c_{13} & c_{33} & 0 & 0 & 0 \\ c_{14} & -c_{14} & 0 & c_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & c_{44} & c_{14} \\ 0 & 0 & 0 & 0 & c_{14} & \frac{1}{2}(c_{11}-c_{12}) \end{bmatrix}$$

Triclinic (21)  $1, \bar{1}$

$$\begin{bmatrix} c_{11} & c_{12} & c_{13} & c_{14} & c_{15} & c_{16} \\ c_{12} & c_{22} & c_{23} & c_{24} & c_{25} & c_{26} \\ c_{13} & c_{23} & c_{33} & c_{34} & c_{35} & c_{36} \\ c_{14} & c_{24} & c_{34} & c_{44} & c_{45} & c_{46} \\ c_{15} & c_{25} & c_{35} & c_{45} & c_{55} & c_{56} \\ c_{16} & c_{26} & c_{36} & c_{46} & c_{56} & c_{66} \end{bmatrix}$$

Trigonal (7)  $3, \bar{3}$

$$\begin{bmatrix} c_{11} & c_{12} & c_{13} & c_{14} & -c_{25} & 0 \\ c_{12} & c_{11} & c_{13} & -c_{14} & c_{25} & 0 \\ c_{13} & c_{13} & c_{33} & 0 & 0 & 0 \\ c_{14} & -c_{14} & 0 & c_{44} & 0 & c_{25} \\ -c_{25} & c_{25} & 0 & 0 & c_{44} & c_{14} \\ 0 & 0 & 0 & c_{25} & c_{14} & \frac{1}{2}(c_{11}-c_{12}) \end{bmatrix}$$

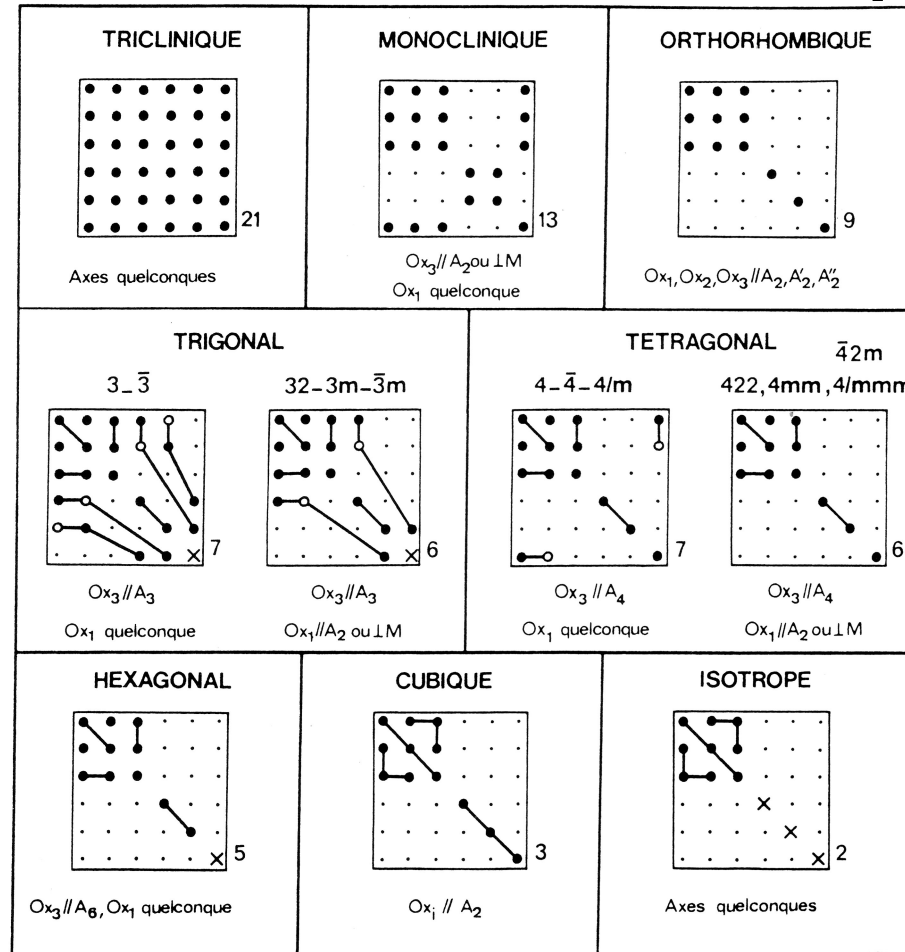
Tetragonal (6) 422,  $4mm$ ,  $\bar{4}2m$ ,  $4/mmm$

$$\begin{bmatrix} c_{11} & c_{12} & c_{13} & 0 & 0 & 0 \\ c_{12} & c_{11} & c_{13} & 0 & 0 & 0 \\ c_{13} & c_{13} & c_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & c_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & c_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & c_{66} \end{bmatrix}$$

Tetragonal (7),  $4, \bar{4}$  4/m

$$\begin{bmatrix} c_{11} & c_{12} & c_{13} & 0 & 0 & c_{16} \\ c_{12} & c_{11} & c_{13} & 0 & 0 & -c_{16} \\ c_{13} & c_{13} & c_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & c_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & c_{44} & 0 \\ c_{16} & -c_{16} & 0 & 0 & 0 & c_{66} \end{bmatrix}$$

# Constants Elastique



Composantes  $c_{\alpha\beta}$  du tenseurs des rigidités élastiques suivant les systèmes de symétrie avec les axes de référence de la figure 2. 22.

composante - non nulle:  $\bullet$   $\circ$  - nulle:  $\cdot$   
 composantes - égales:  $\bullet\bullet$  - opposées:  $\bullet\circ$   
 - égales à  $(c_{11} - c_{12})/2$ :  $\times$

La symétrie par rapport à la diagonale principale n'est pas mentionnée. Le nombre de composantes indépendantes est indiqué, en bas, à droite de chaque ensemble.



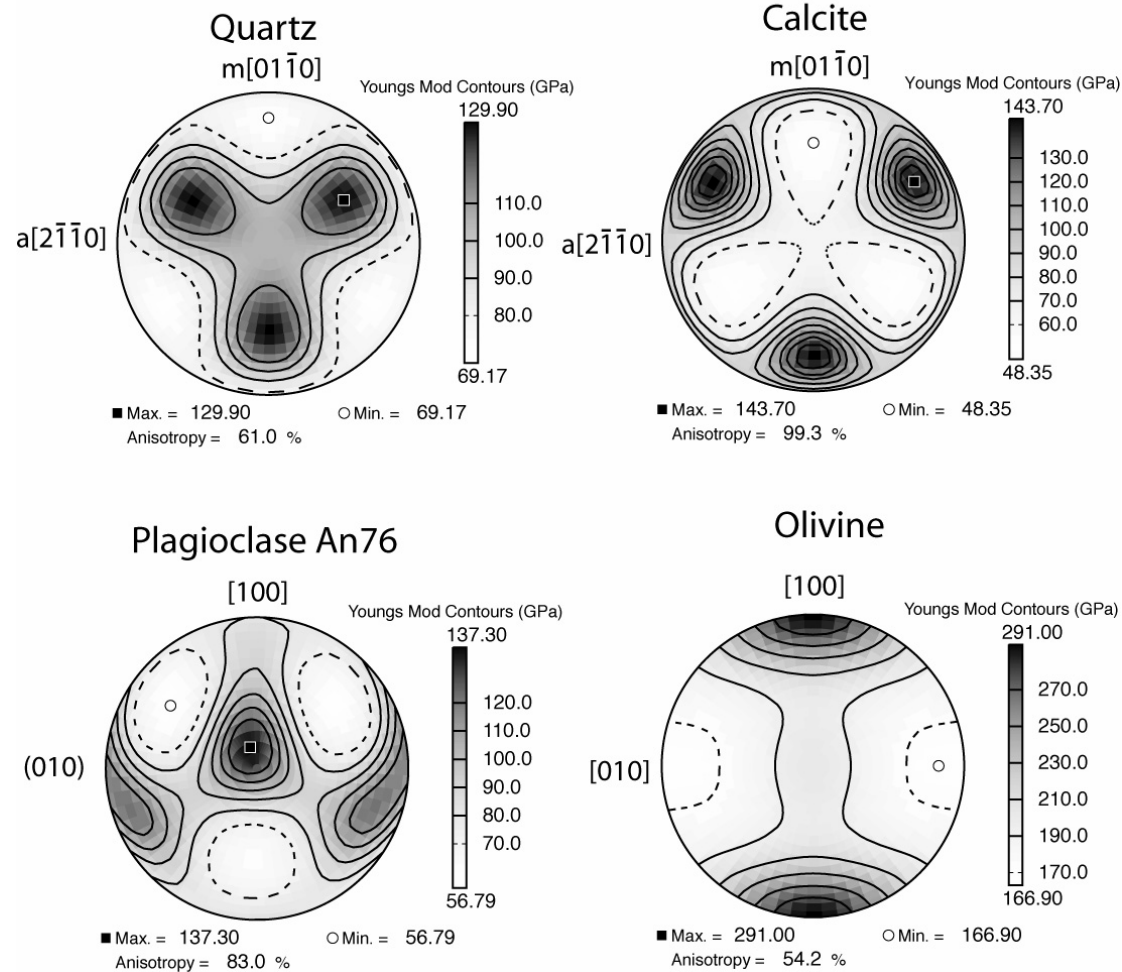
# What can you do with $C_{ij}$ ?

- Elastic wave velocity -  $V_p, V_s$ , (slowness -  $1/V$ ) polarization, wavefront velocity
- Moduli - Young's modulus
- Linear compressibility in any direction
- Volume compressibility or Bulk modulus
- Mechanical instability analysis - softening of acoustic modes
- Isotropic moduli and velocities
- Debye temperature & mean sound (phono) velocity
- Elastic energy

# Young's Modulus

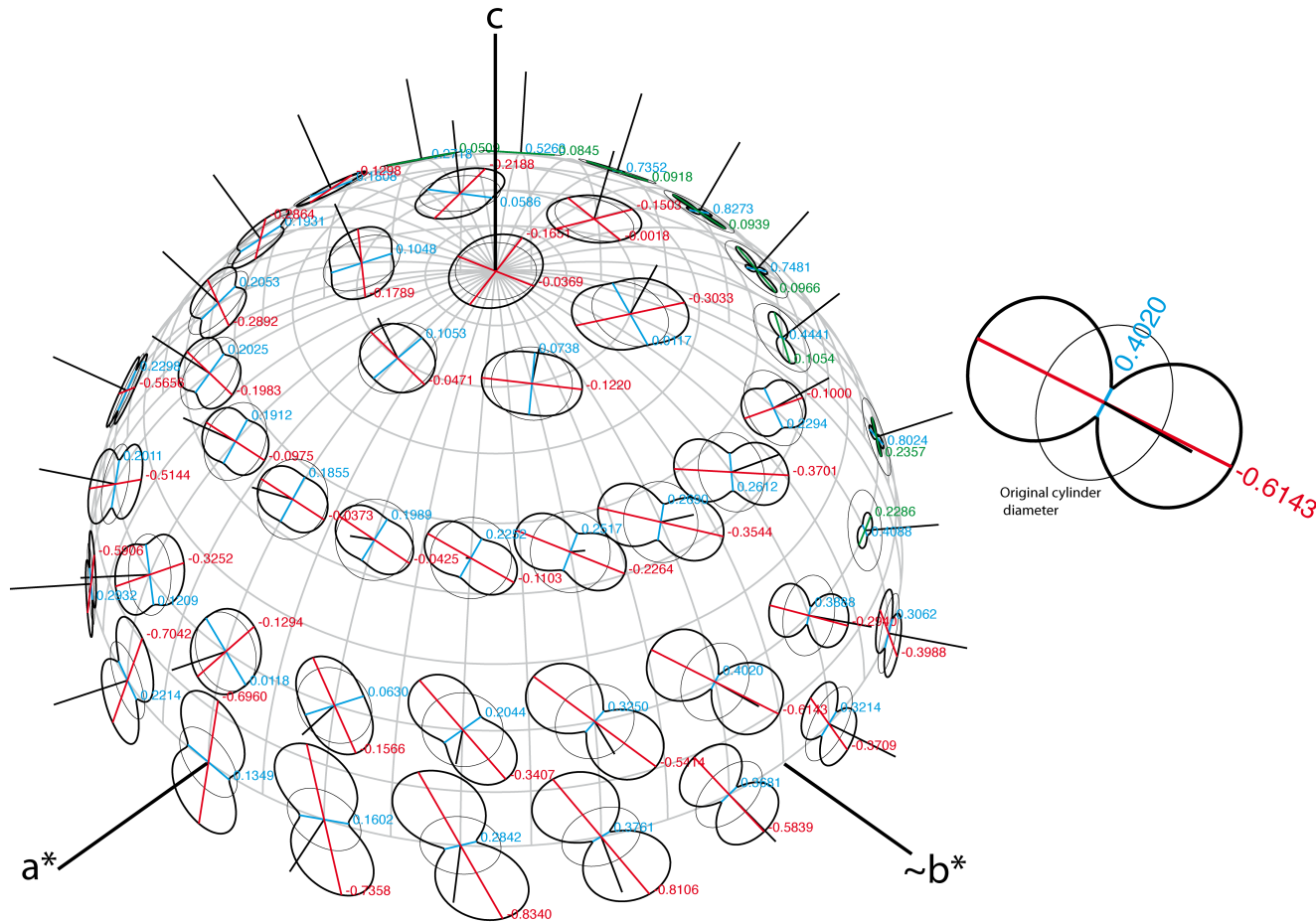
No single surface can represent the elastic behaviour of a single crystal. The surface of the Young's modulus is often used. Young's modulus for a given direction is defined as the ratio of the longitudinal stress to the longitudinal elastic strain.

## Single Crystal Young's Modulus



# Talc – Poisson's ratio

Talc P= 0.0 GPa



# Isotropic moduli

$$K_R = [S_{11} + S_{22} + S_{33} + 2(S_{12} + S_{13} + S_{23})]^{-1} = \frac{1}{\beta},$$

$$G_R = 15[4(S_{11} + S_{22} + S_{33}) - 4(S_{12} + S_{13} + S_{23}) + 3(S_{44} + S_{55} + S_{66})]^{-1}$$

$$K_V = \frac{1}{9} [C_{11} + C_{22} + C_{33} + 2(C_{12} + C_{13} + C_{23})],$$

$$G_V = \frac{1}{15} [C_{11} + C_{22} + C_{33} - (C_{12} + C_{13} + C_{23}) + 3(C_{44} + C_{55} + C_{66})]$$

$$K_{VRH} = \frac{1}{2} (K_V + K_R); \quad G_{VRH} = \frac{1}{2} (G_V + G_R)$$

# Typical Cij data set

Table 5. Elastic Moduli of Tetragonal Crystals (6 Moduli) at Room P & T

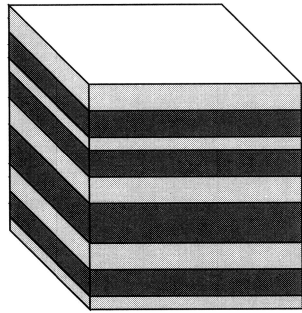
Material	$\rho$ Mg/m <sup>3</sup>	Subscript ij in modulus $c_{ij}$ (GPa)						$K_S$ GPa	$G$ GPa	Refer- ences
		11	33	44	66	12	13			
<i>Rutile-Structured</i>										
SiO <sub>2</sub> , Stishovite	4.290	453	776	252	302	211	203	316	220	143
SiO <sub>2</sub> , $\alpha$ -Cristobalite	2.335	59.4	42.4	67.2	25.7	3.8	-4.4	16.4	39.1	151
SnO <sub>2</sub> , Cassiterite	6.975	261.7	449.6	103.1	207.4	177.2	155.5	212.3	101.8	22
TeO <sub>2</sub> , Paratellurite	6.02	55.7	105.8	26.5	65.9	51.2	21.8	45.0	20.4	93
TiO <sub>2</sub> , Rutile	5.99	53.2	108.5	24.4	55.2	48.6	21.2	43.7	19.0	122
GeO <sub>2</sub>	4.260	269	480	124	192	177	146	215.5	112.4	47
	6.279	337.2	599.4	161.5	258.4	188.2	187.4	257.6	150.8	131
<i>Other Minerals</i>										
Ba <sub>2</sub> Si <sub>2</sub> TiO <sub>8</sub> , Fresnoite ( $c^B$ )		140	83	33	59	36	24	56.9	42.1	46
Scapolite, (Na,Ca,K) <sub>4</sub> Al <sub>3</sub> (Al,Si) <sub>3</sub>		166	100	31.7	69.4	58	44	77.6	43.3	46
Si <sub>6</sub> O <sub>24</sub> (Cl,SO <sub>4</sub> ,CO <sub>3</sub> )		99	113	15.6	22.9	35.1	35.4	58.0	23.1	47
Vesuvianite Ca <sub>10</sub> Mg <sub>2</sub> Al <sub>4</sub> (SiO <sub>4</sub> ) <sub>5</sub> (Si <sub>2</sub> O <sub>7</sub> ) <sub>2</sub> (OH) <sub>4</sub>		102	140	23.0	30.4	38.9	43.3	65.3	29.1	47
ZrSiO <sub>4</sub> <sup>a</sup> , Zircon		102	140	23.0	30.4	38.9	43.3	65.3	29.1	47
		153	166	55.8	54.0	48	44	82.6	55.5	47
	4.675	424.3	489.3	131.1	48.3	69.7	149	227.9	109.0	88
	4.70	256	372	73.5	116	175	214	223.9	66.6	47

<sup>a</sup> nonmetamict.

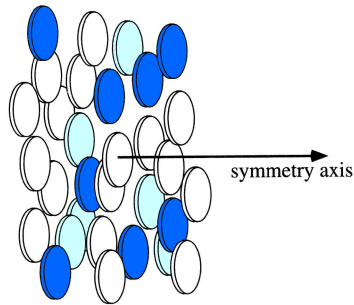
From J.D.Bass (1995) Elasticity of Minerals, Glasses and Melts  
Mineral Physics and Crystallography A Handbook of Physical Constants  
AGU Reference Shelf 2 pp 45-63.

<http://www.agu.org/reference/minphys.html>

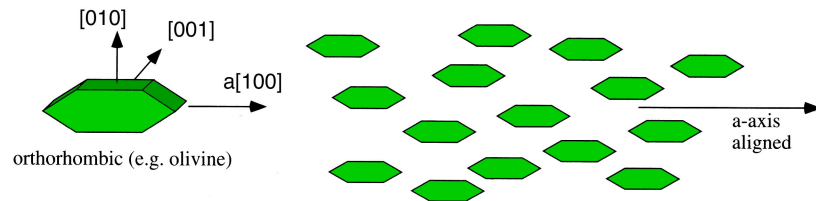
# The origin of seismic anisotropy



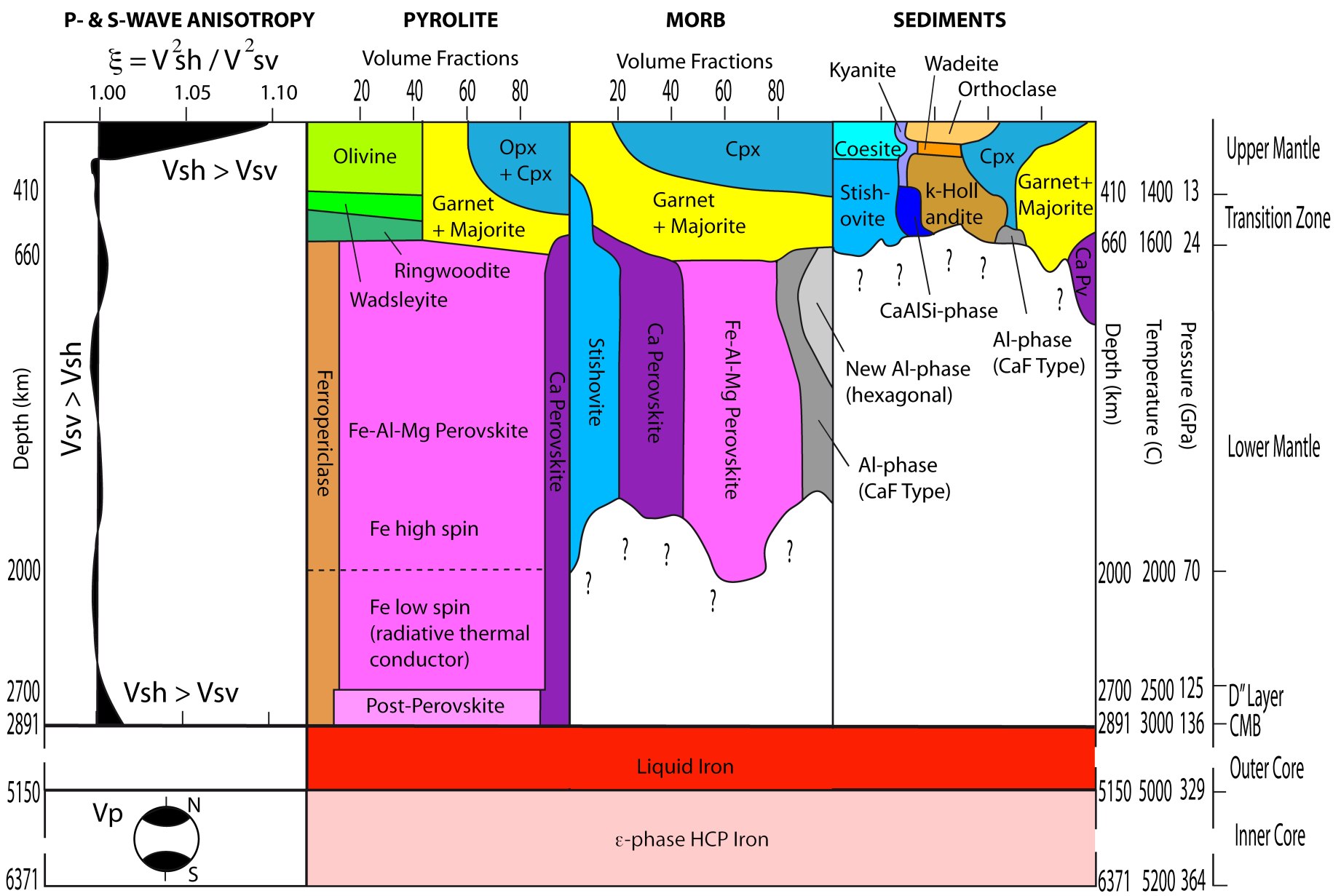
Horizontal layering - sediments, metamorphic layering..  
Upper and lower crust, transition zone, D''



Vertically aligned cracks in the crust.....  
filled with gas, liquid or solid



CPO in lower crust, upper mantle  
& inner core



# Christoffel Tensor $T_{ik}$

The calculation of the three seismic phase velocities from the Christoffel tensor

$$T_{ik} = C_{ijkl} n_j n_l$$

where  $\mathbf{n}$  is the plane wave propagation direction and  $C_{ijkl}$  the elastic constants of the crystal. The three eigenvalues  $E_i$  of the symmetric Christoffel tensor ( $T_{ik}$ ) are related to three seismic phase velocities  $V_i$  (qP,qS1,qS2) by  $V_i = (E_i/\rho)^{1/2}$ ,

where  $\rho$  is the density and  $qS1 > qS2$ .

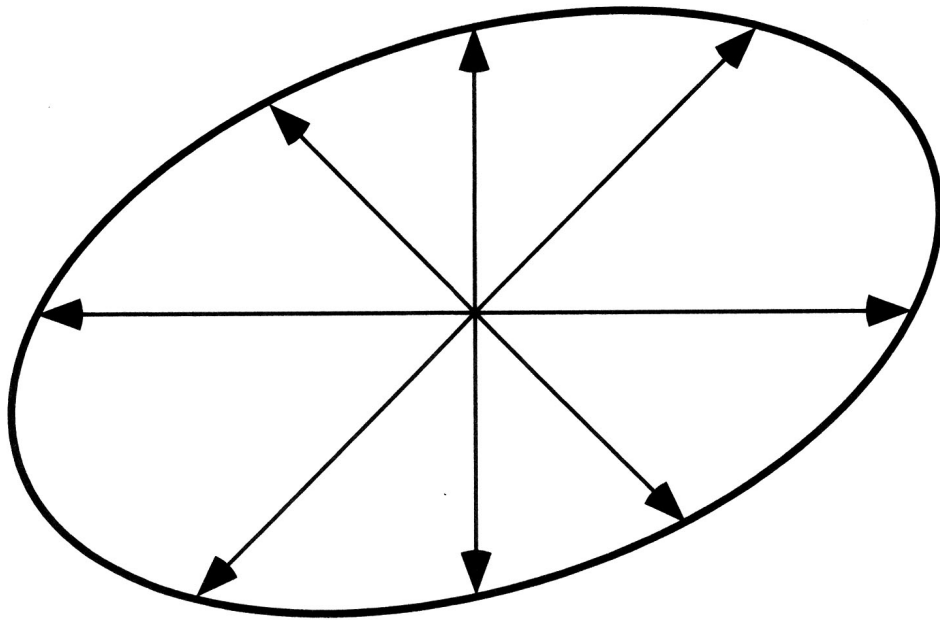
The three eigenvectors of Christoffel tensor are the particle motion vectors of qP,qS1,qS2.

Why qP and not  $V_p$  ?

For a triclinic elastic body the plane wave propagation direction ( $\mathbf{n}$ ) is not *parallel* to the particle motion vector for P, hence it is called quasi-P or qP. Similarly the  $\mathbf{n}$  is not *perpendicular* to particle motion vector for S1 and S2, hence qS1,qS2.

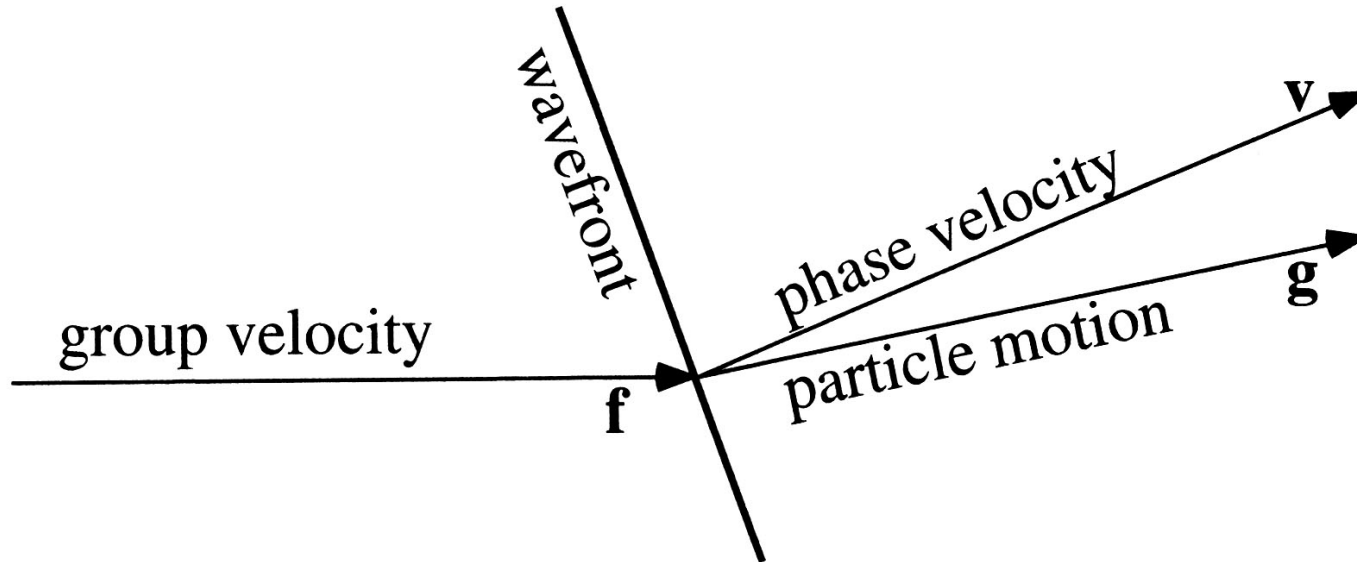


# Wavefronts



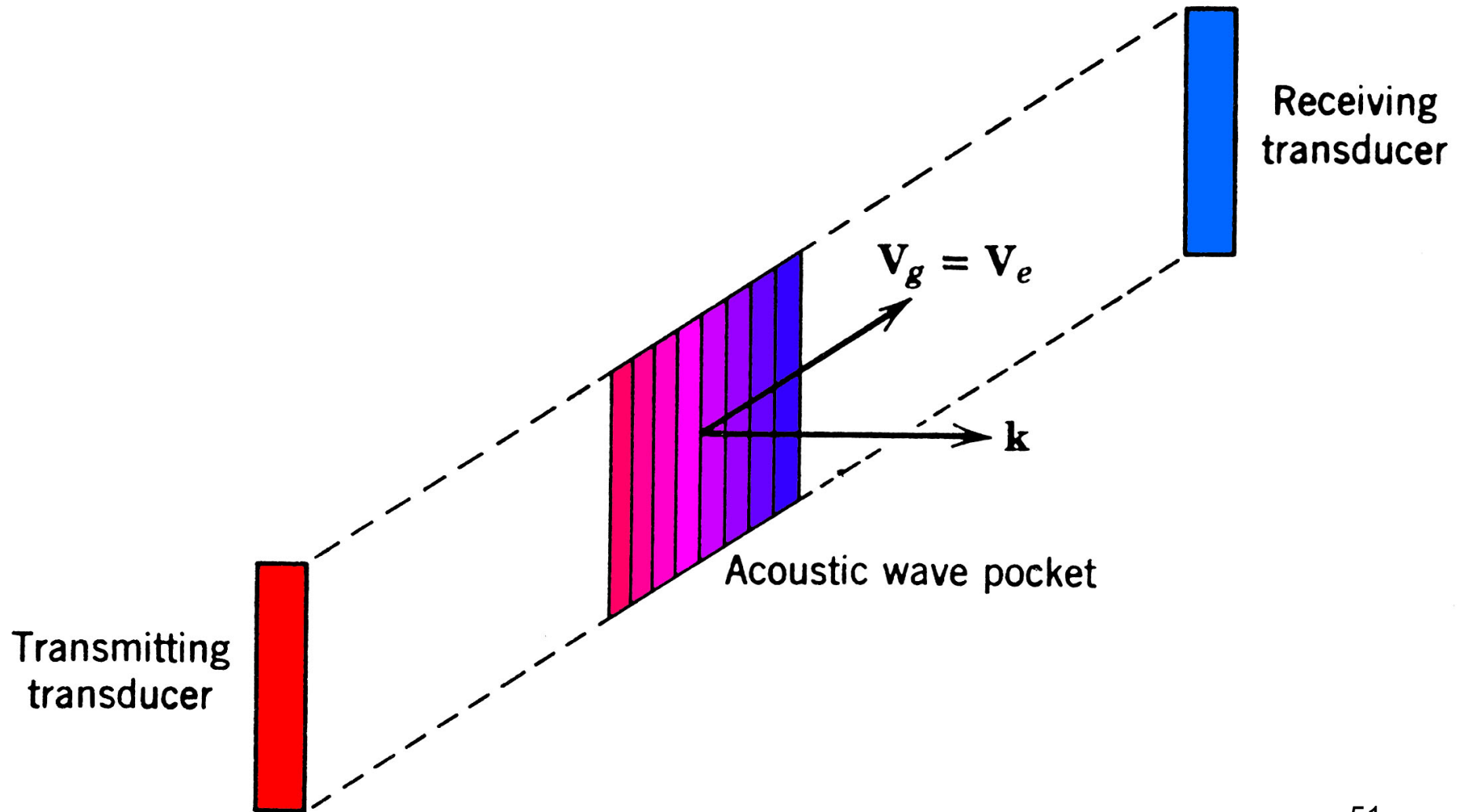
The wavefront generated by a point source in an anisotropic material.

# Group and phase velocity



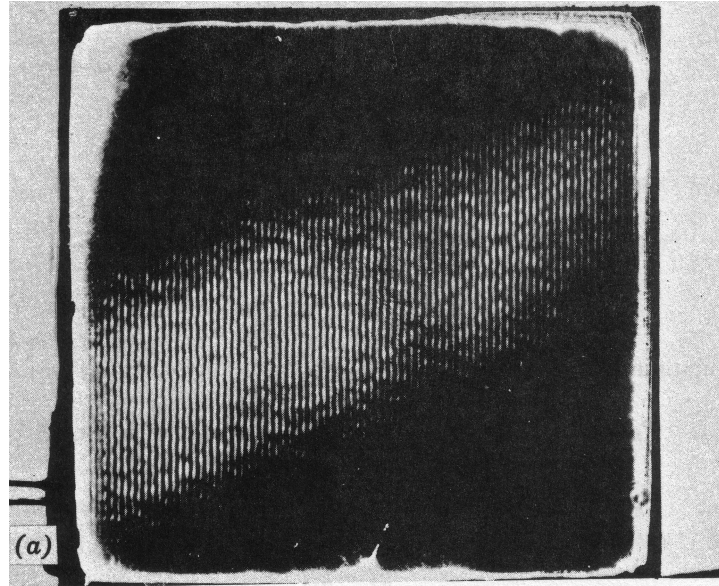
The phase velocity, group velocity, and particle motion vectors point in different directions for a quasi-compressional wave propagating in anisotropic material (except along a symmetry plane of the anisotropy where they will coincide). The phase velocity vector  $\mathbf{v}$  points in the same direction as the slowness vector  $\mathbf{s}$

# Deflected beam trajectory in an anisotropic solid

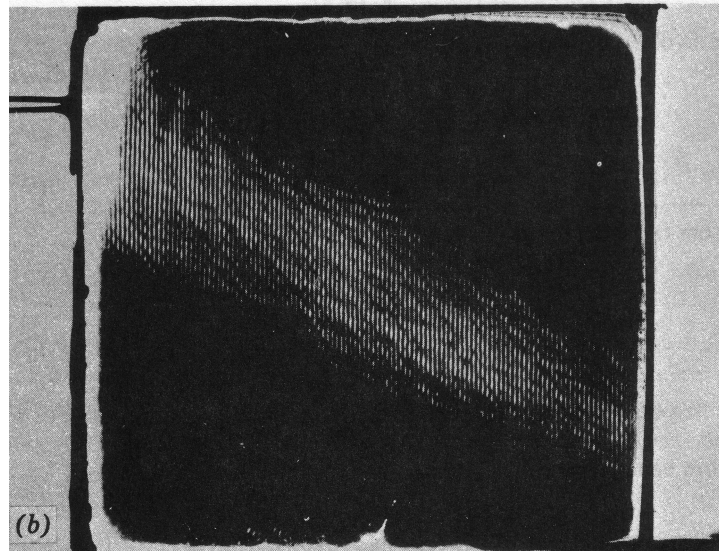


# Deflected trajectory in Quartz crystal

Quasi-Vp

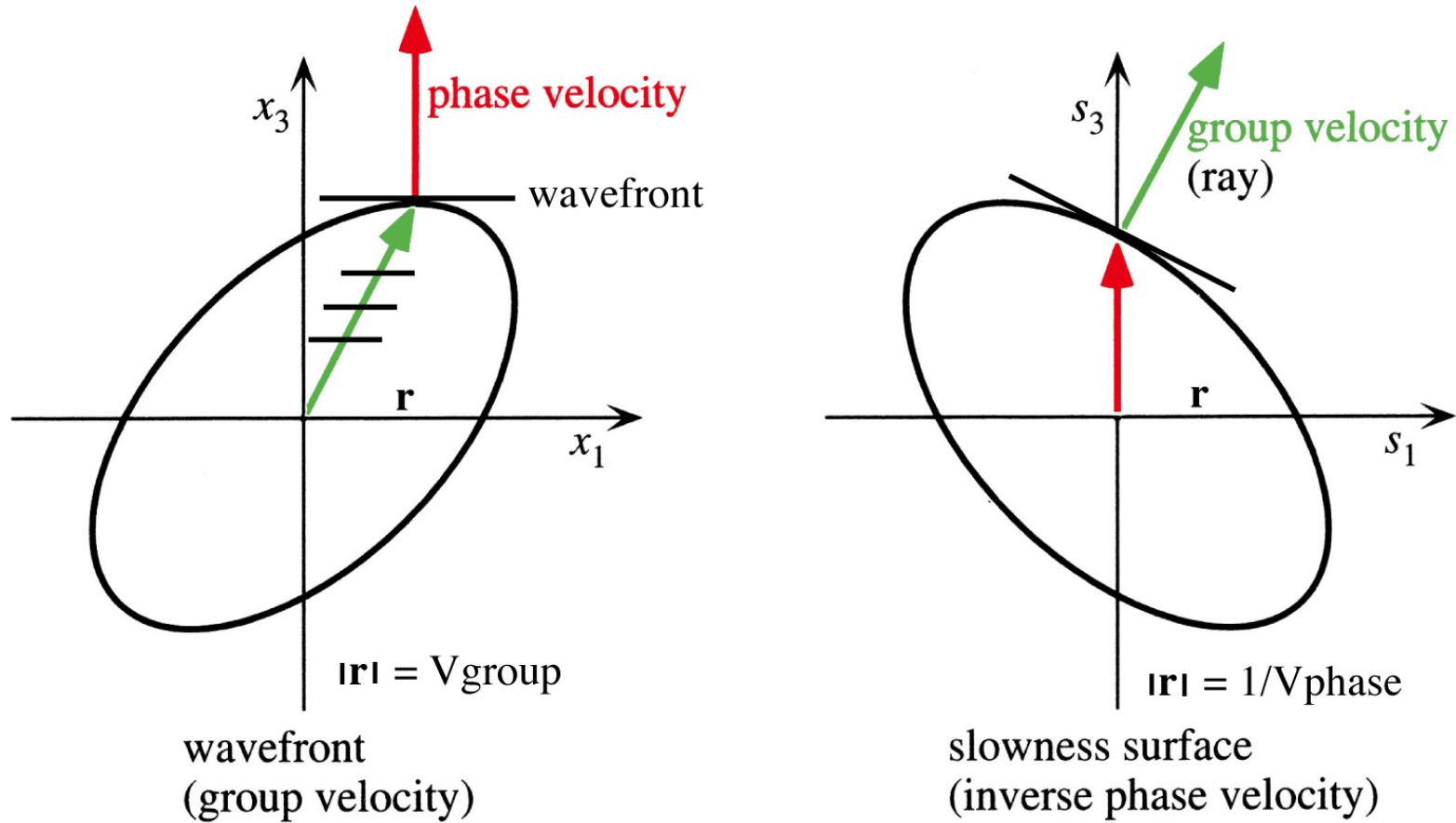


Quasi-Vs



Staudt & Cook (1967)

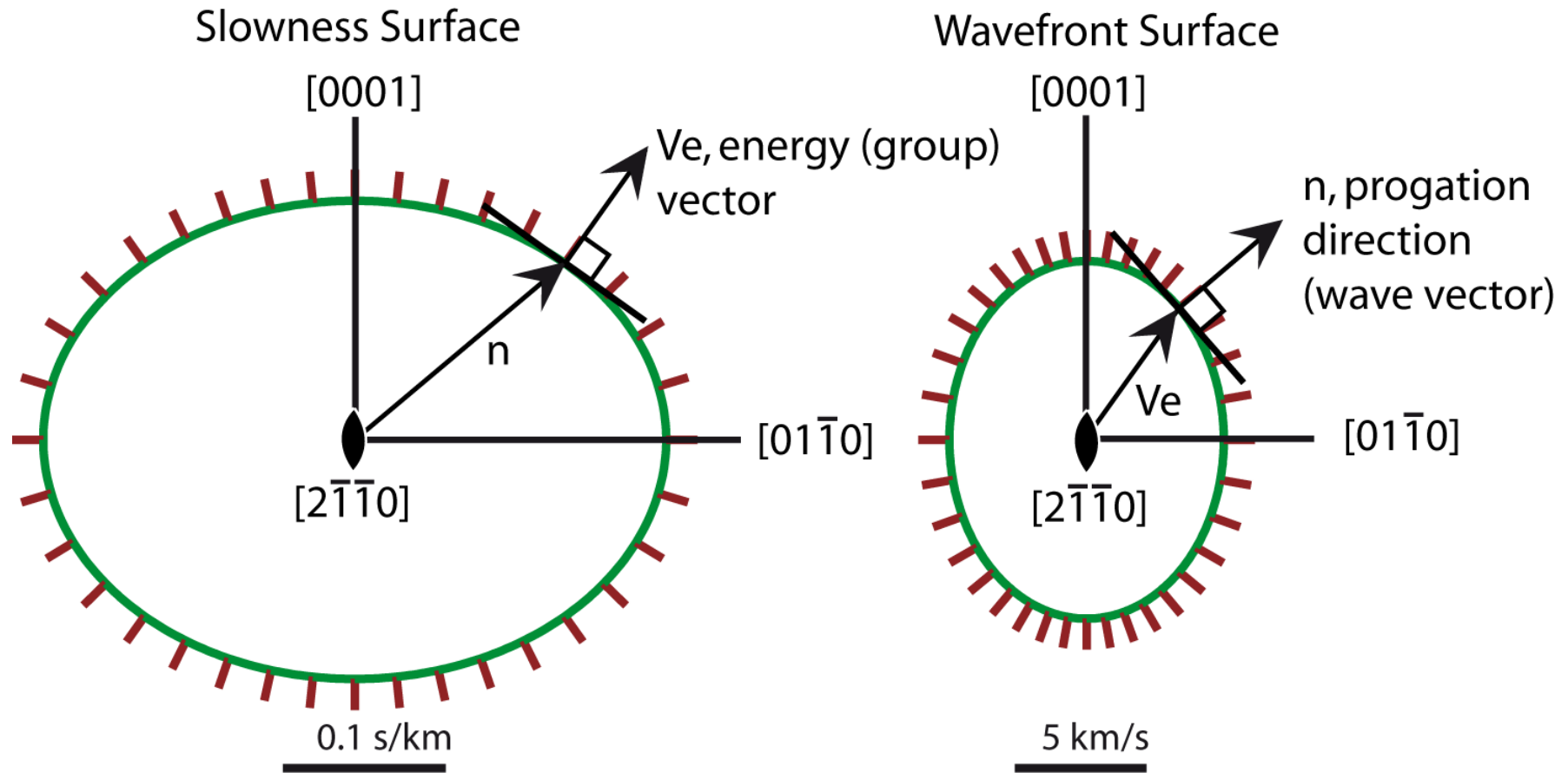
# Group velocity and slowness surfaces



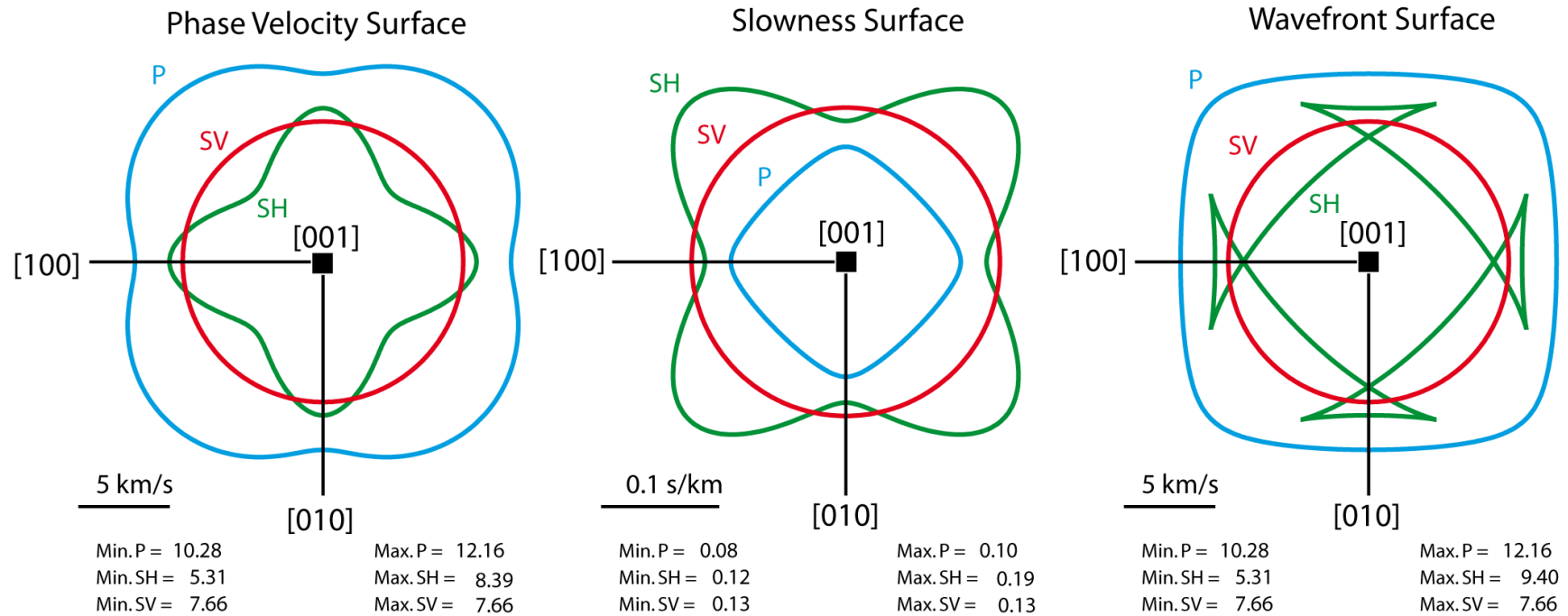
The relationship between the wavefront and slowness surface for anisotropic material.

N.B. These surfaces are not generally an ellipse !

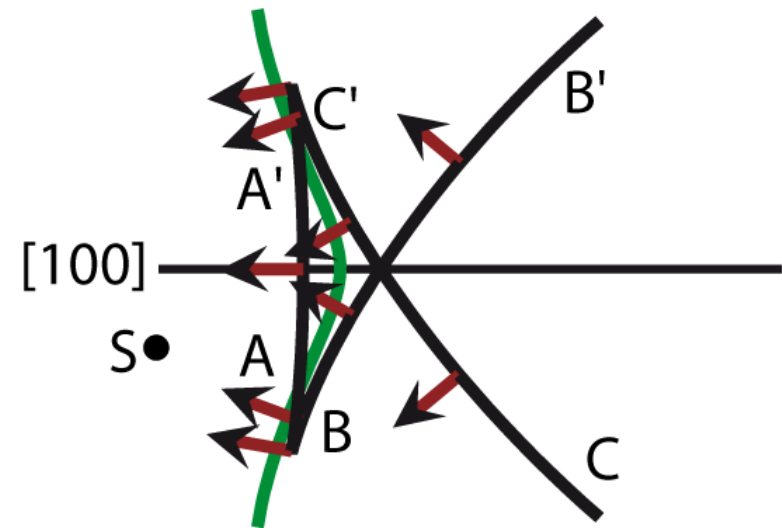
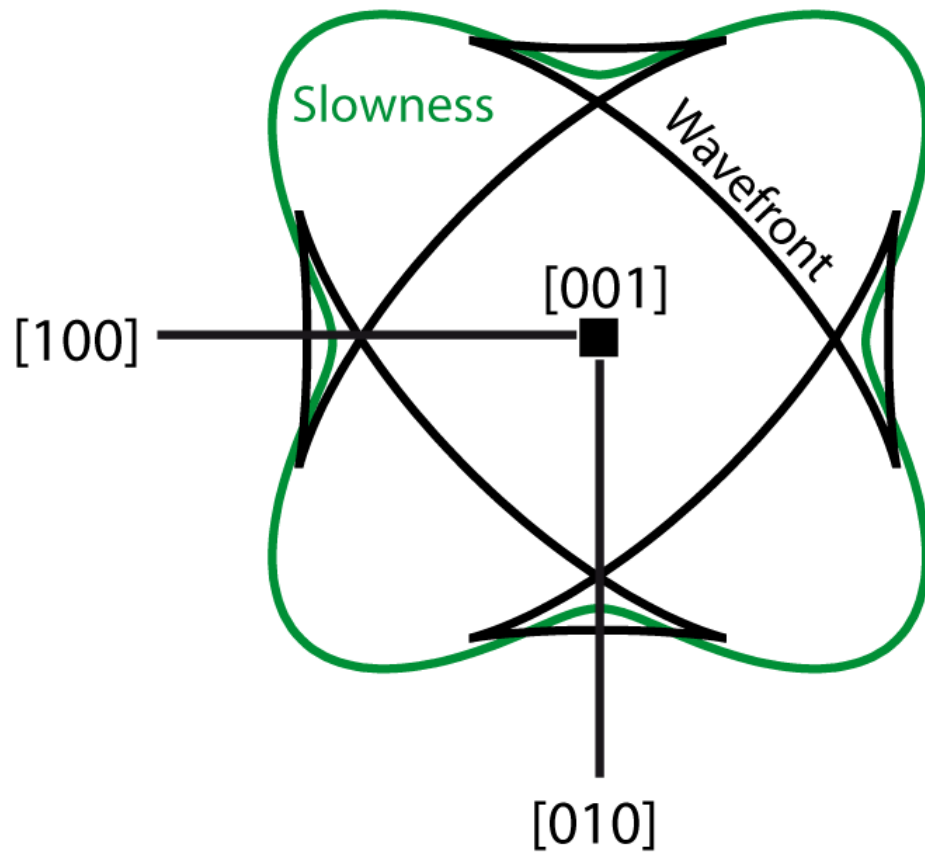
# SH-wave surfaces of $\epsilon$ -phase iron at a pressure of 211 GPa



## Stishovite surfaces in the (001) plane

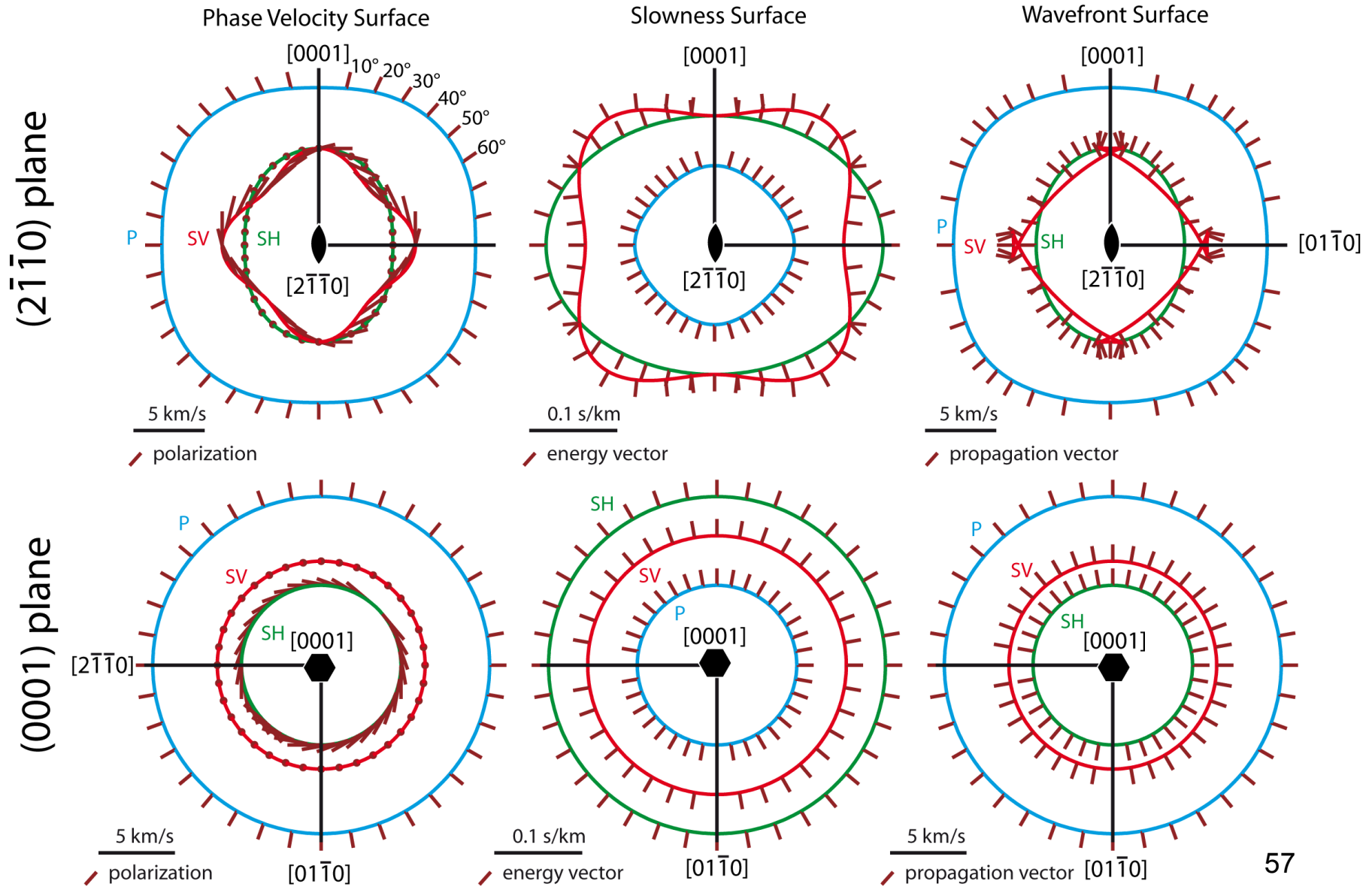


# Wavefront cusps on SH-waves in Stishovite in the (001) plane

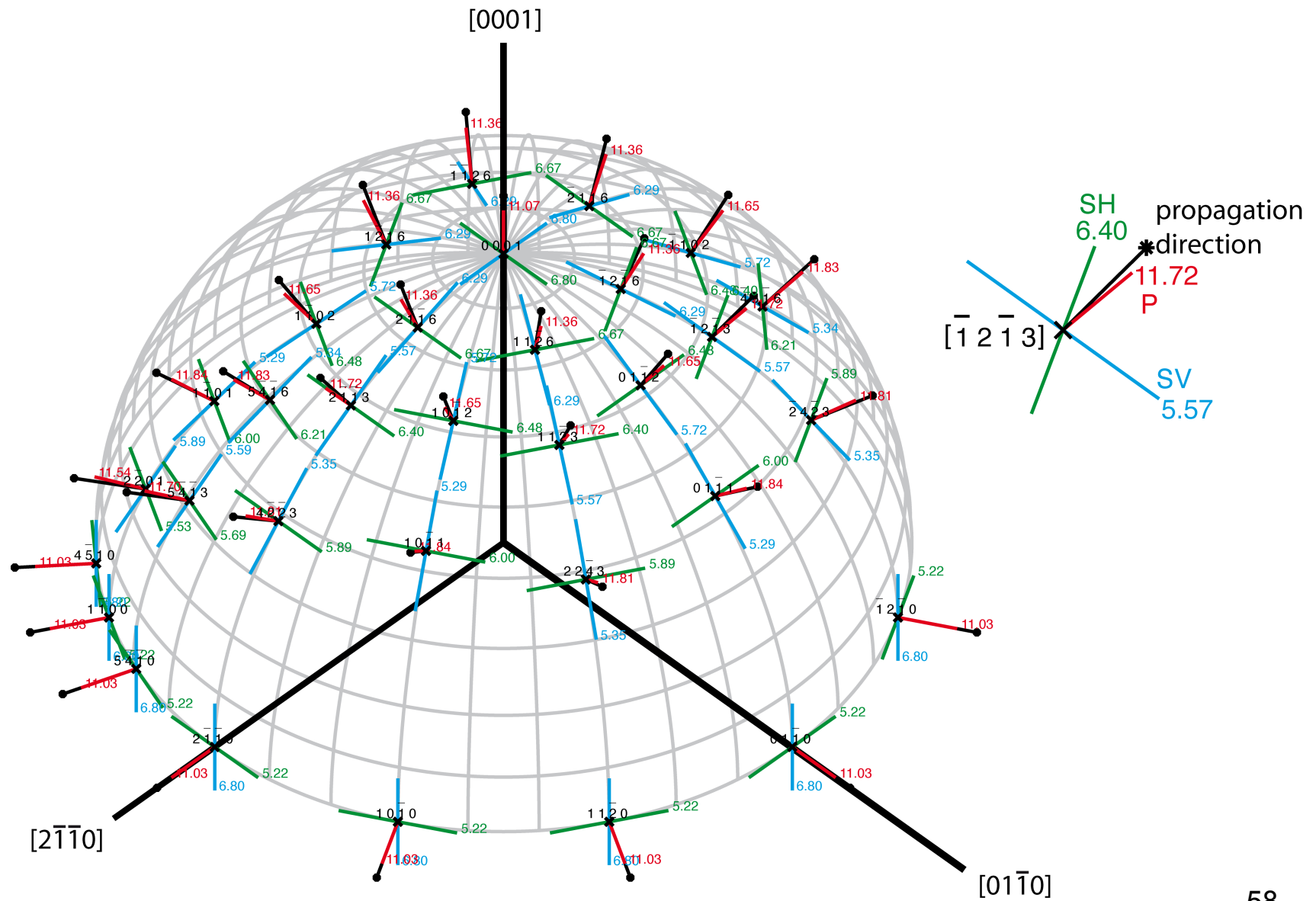




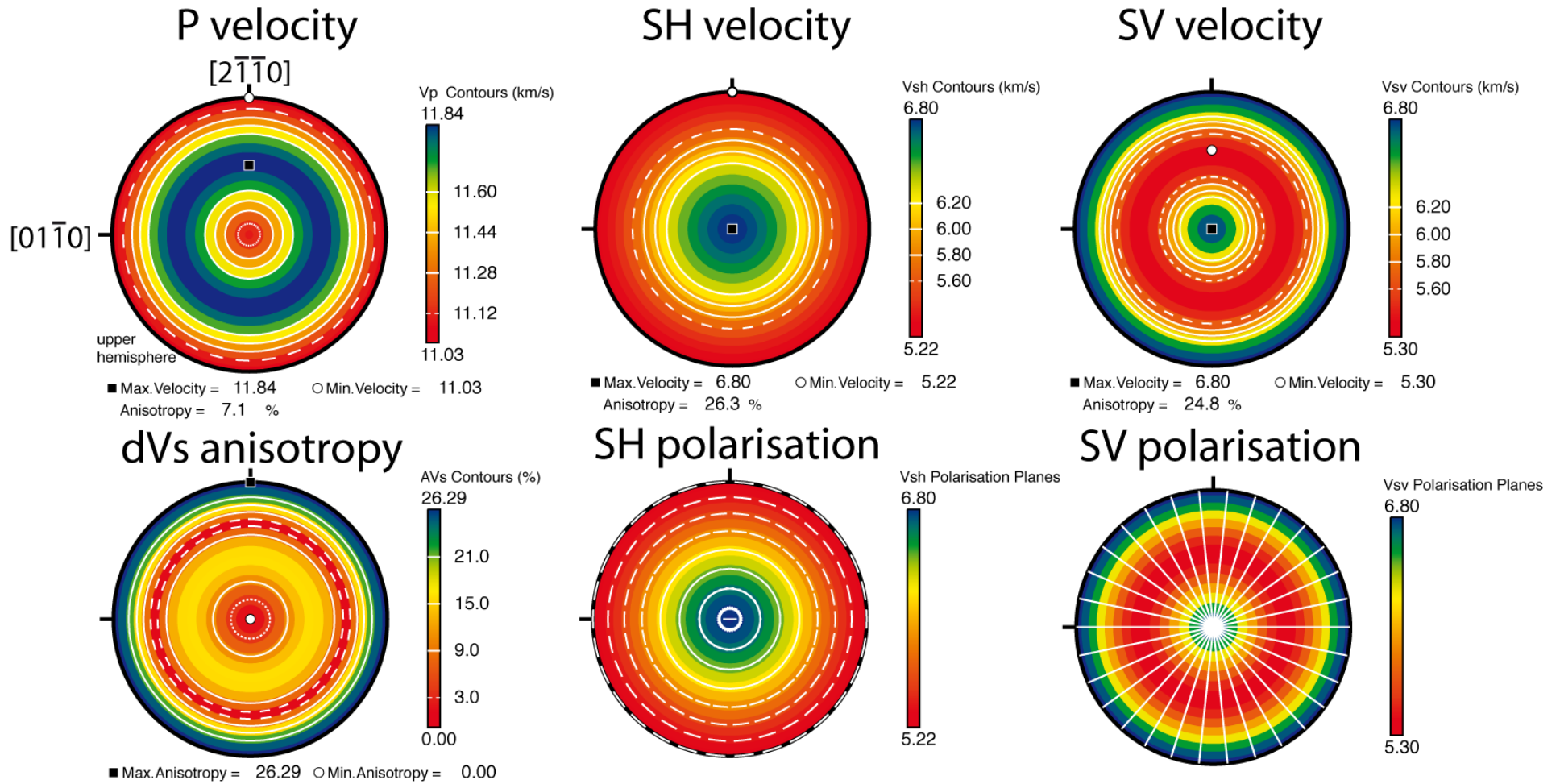
# $\epsilon$ -phase iron (hexagonal) surfaces in the (100) and (001) planes



# $\epsilon$ -phase iron (hexagonal)



# $\epsilon$ -phase iron (hexagonal)

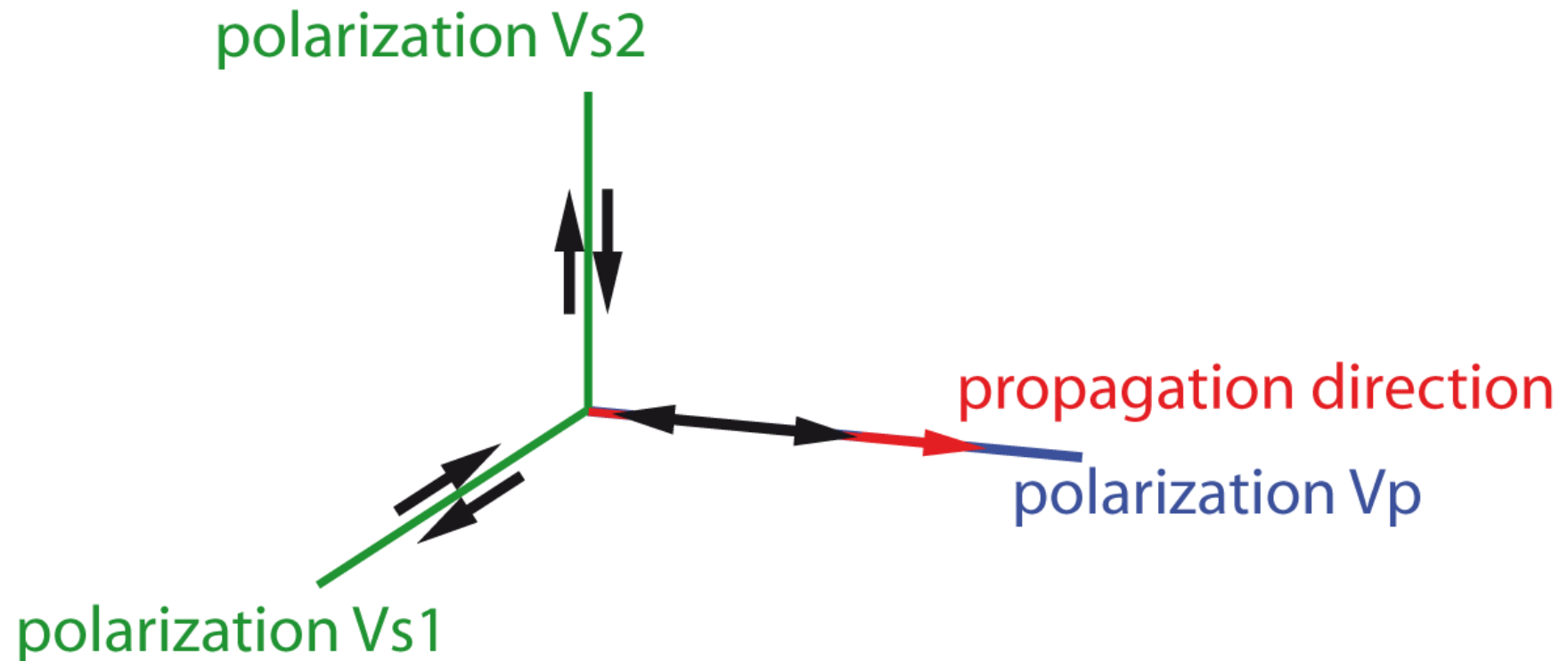


## Isotropic case :

propagation direction // polarization  $V_p$

in general  $V_p > V_{s1} = V_{s2}$

polarization  $V_{s1}$  &  $V_{s2} \perp$  propagation direction

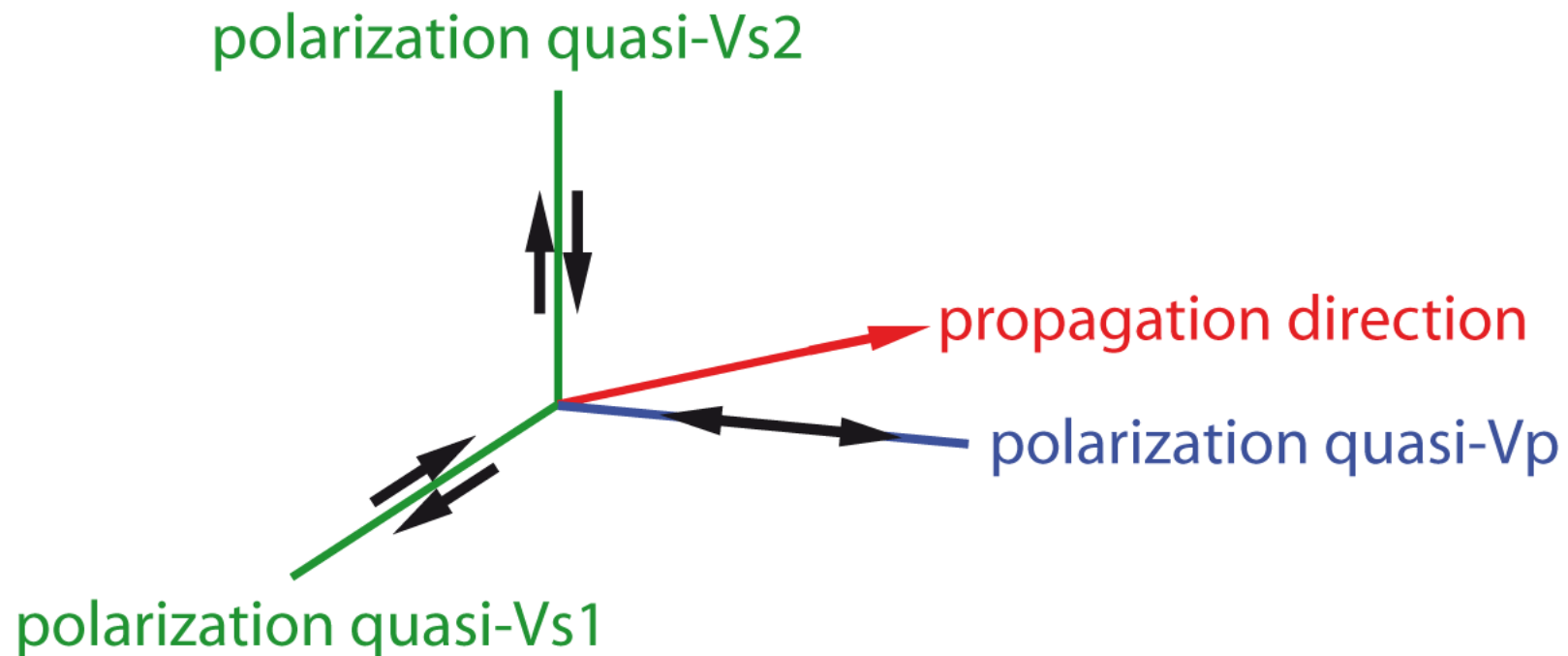


## Anisotropic case :

polarisation of quasi-Vp closest to propagation direction

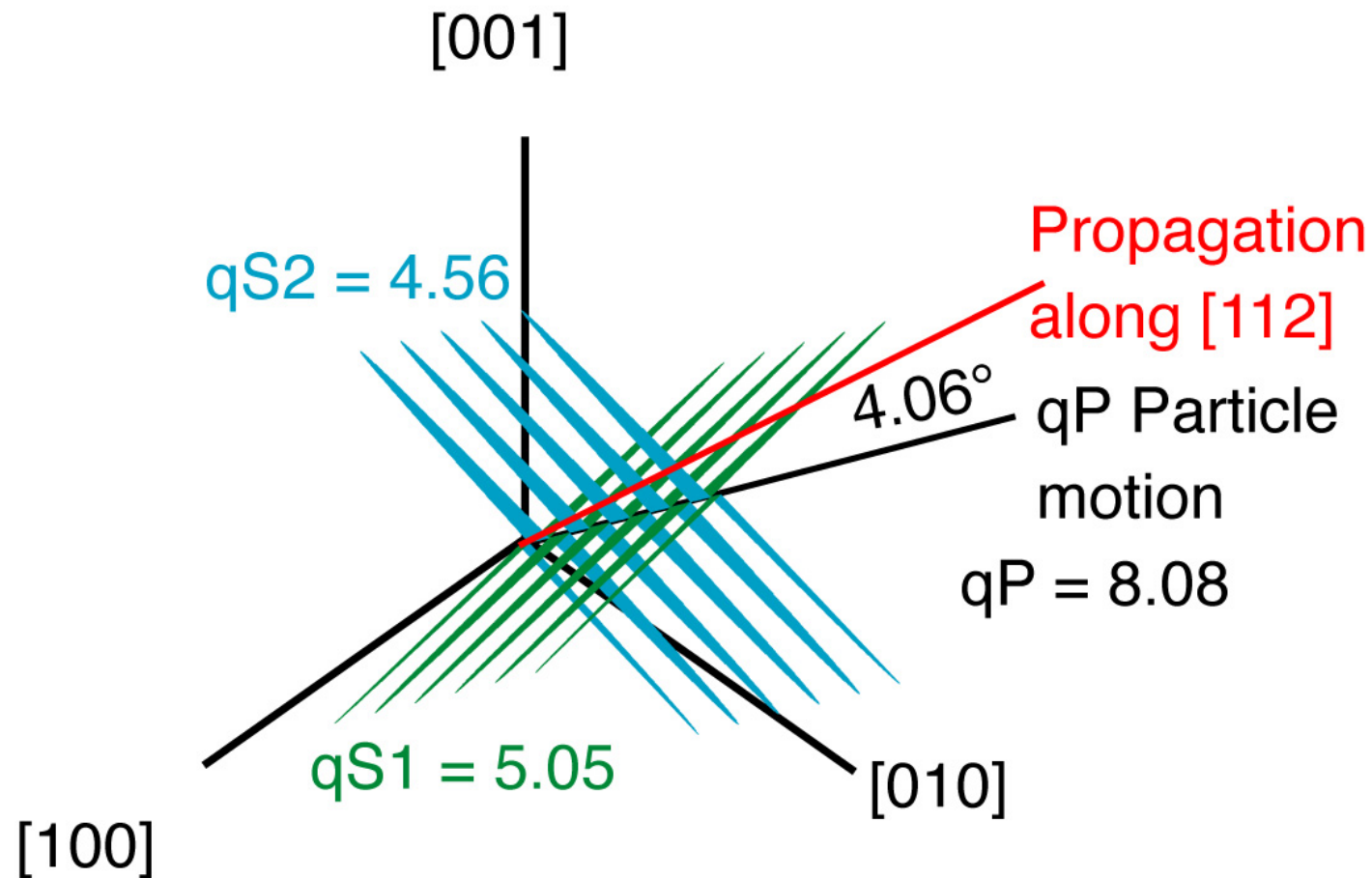
in general  $\text{quasi-Vp} > \text{quasi-Vs1} > \text{quasi-Vs2}$

polarizations are mutually perpendicular

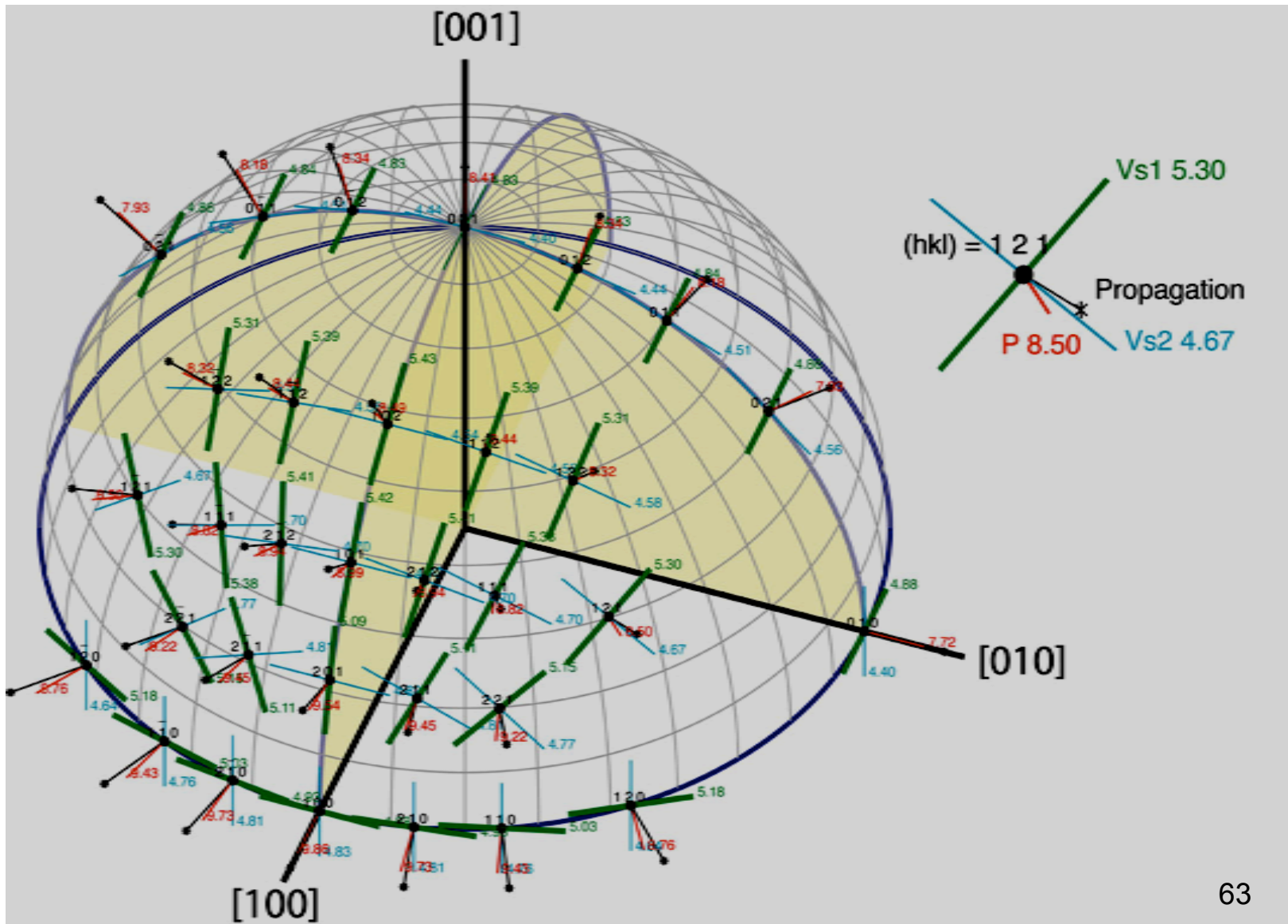


# Olivine Single Crystal - quasi-P, S1 & S2

General case : particale motion not parallel to propagation direction long [112]

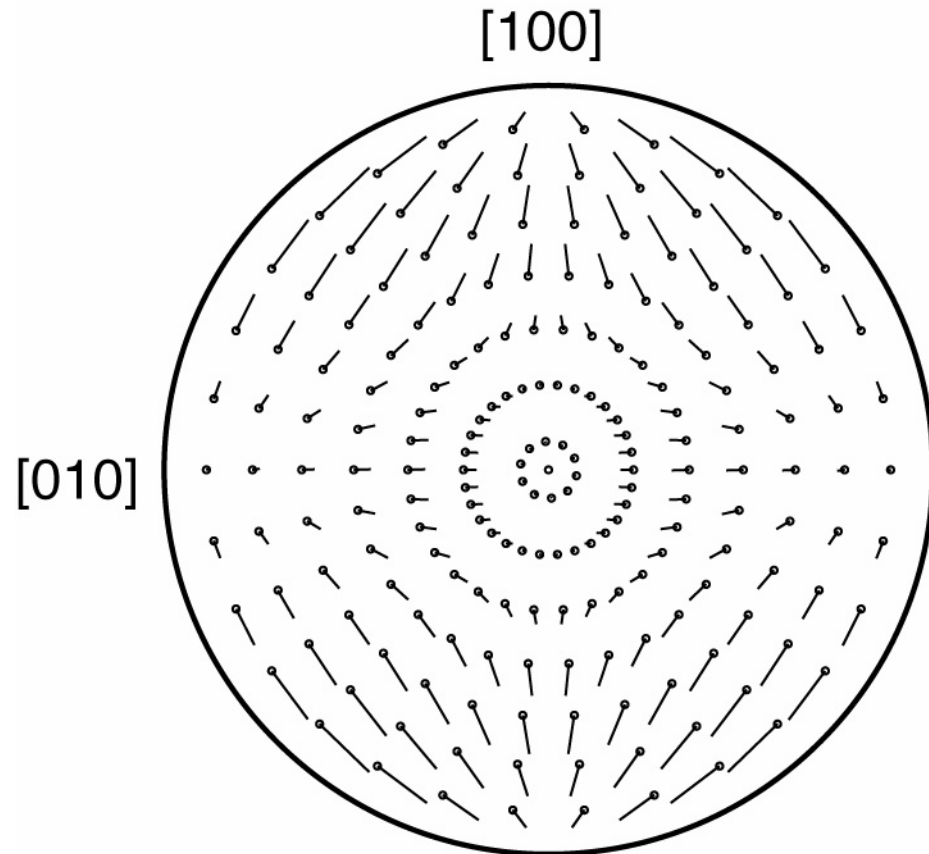


# Olivine Single Crystal

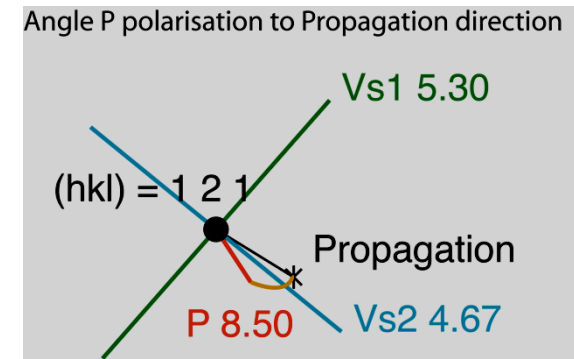


# Olivine Single Crystal

qP Propagation to Polarisation

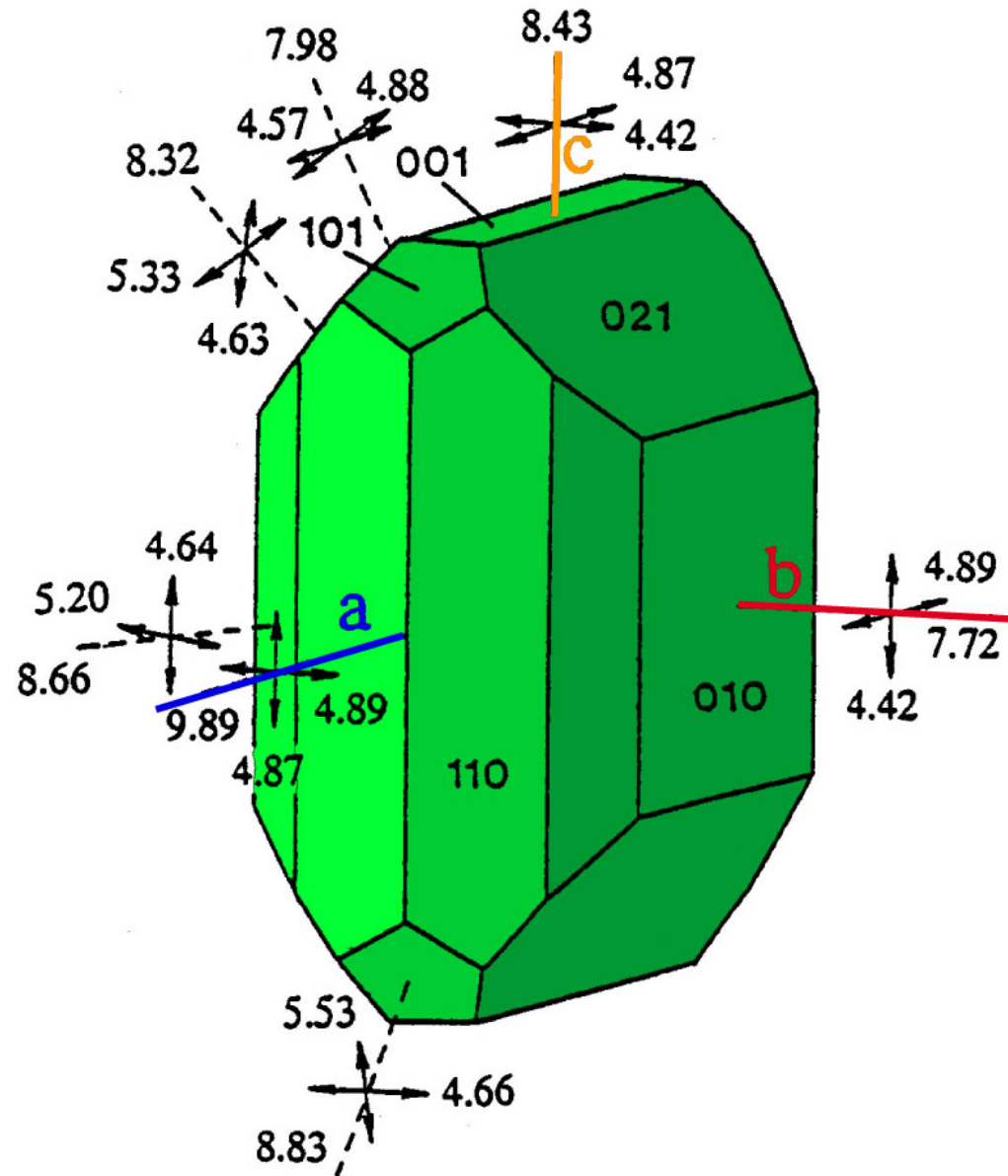


- Propagation vector upper hemisphere

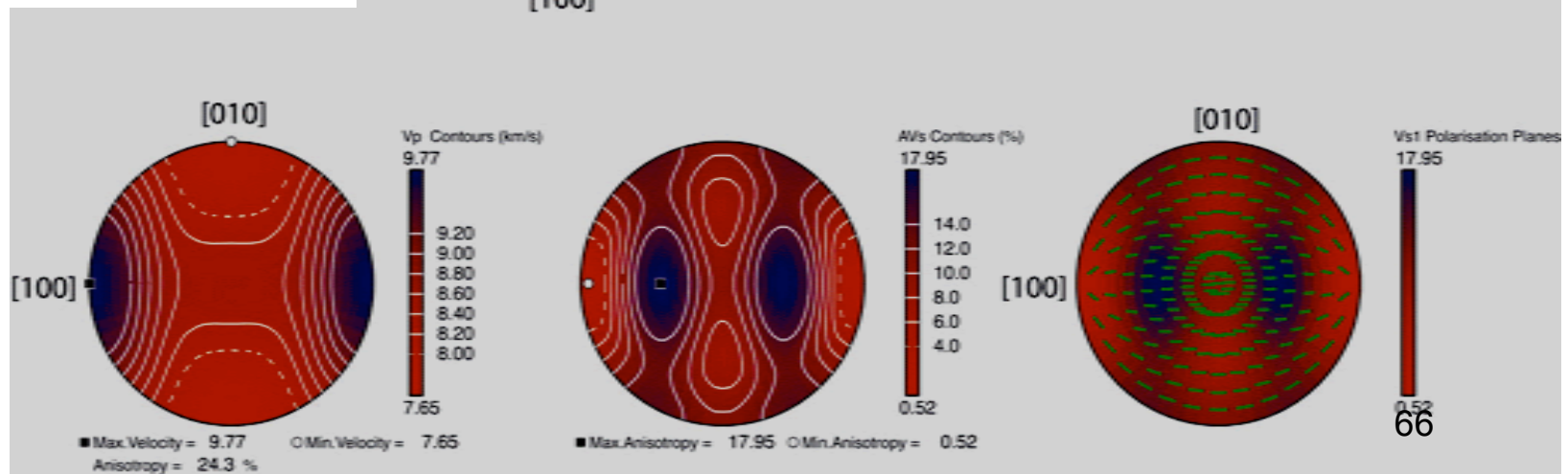
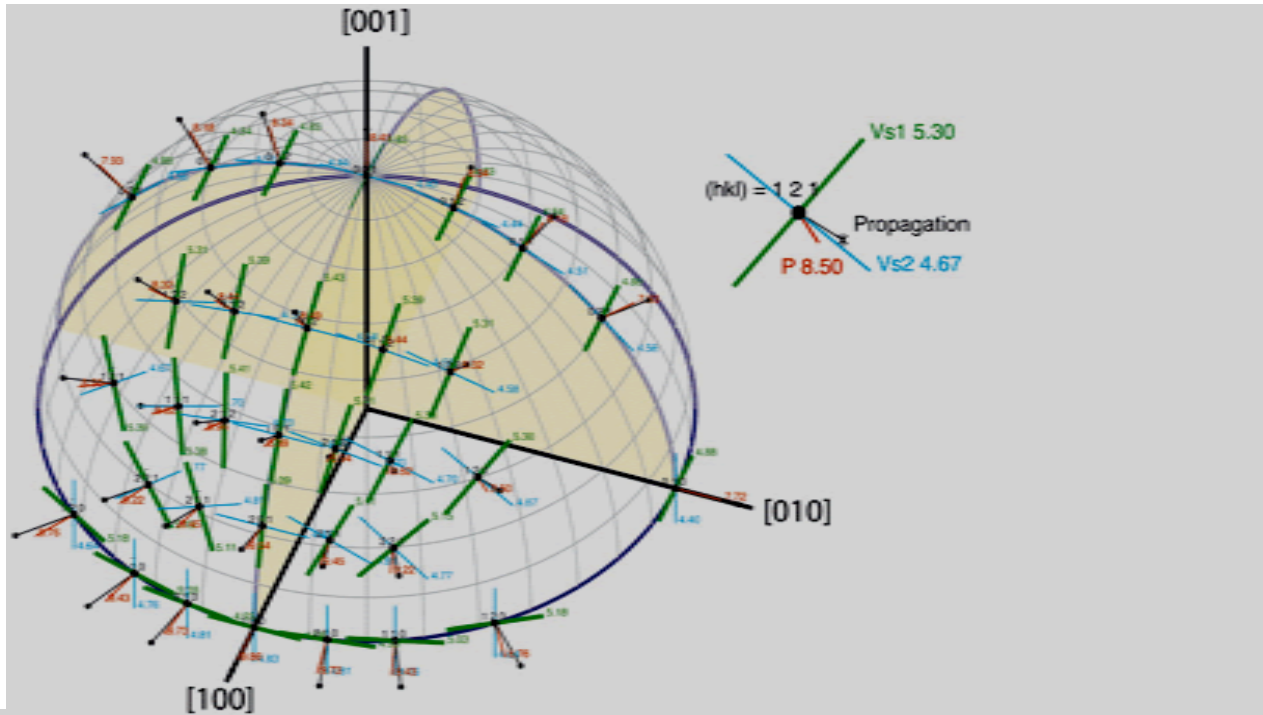
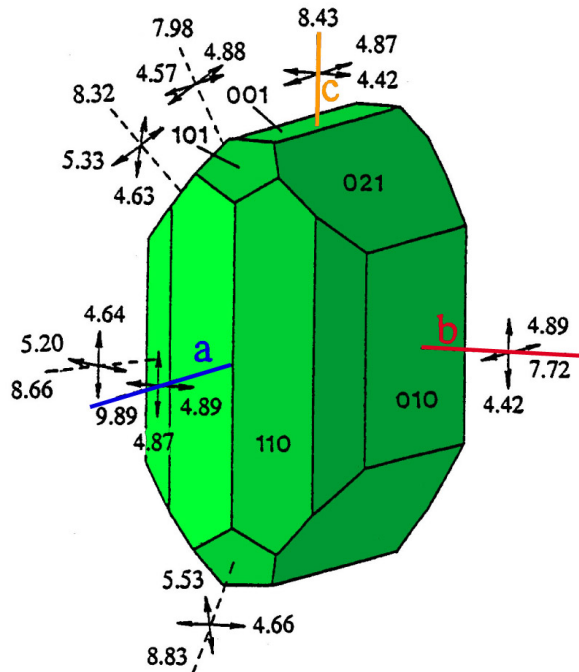




# Olivine

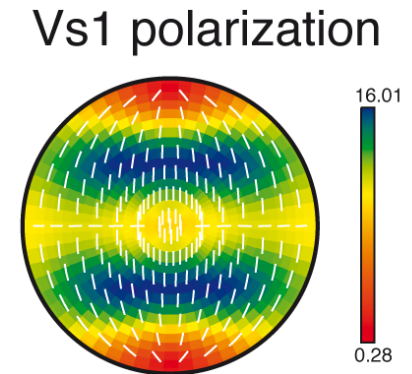
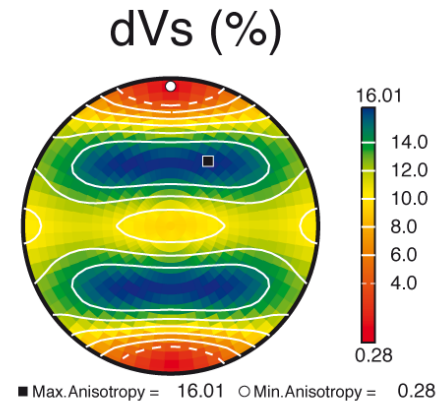
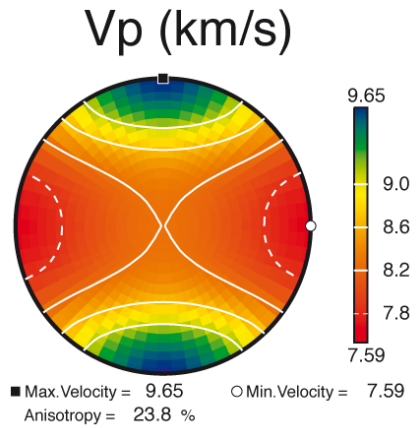
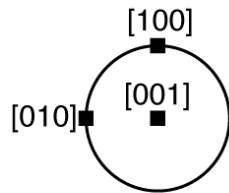


# Olivine 3D to 2D projection

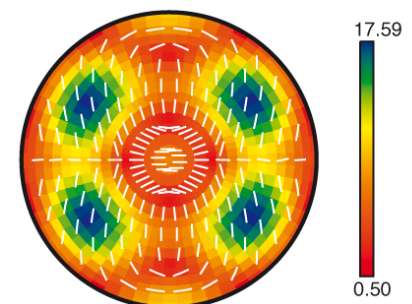
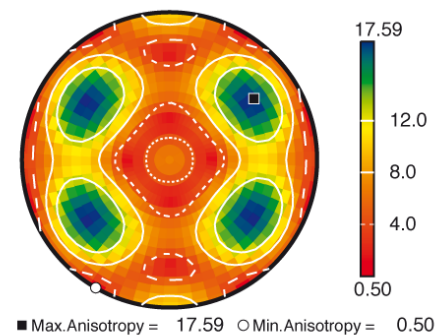
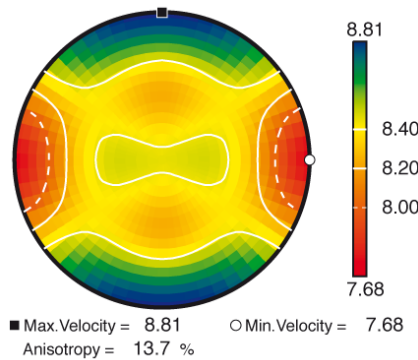


# Upper Mantle (0- 410 km)

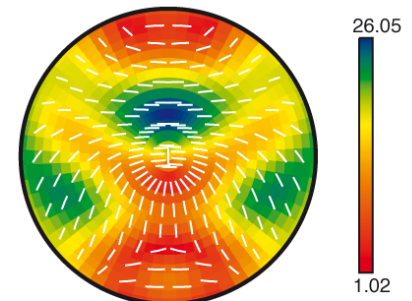
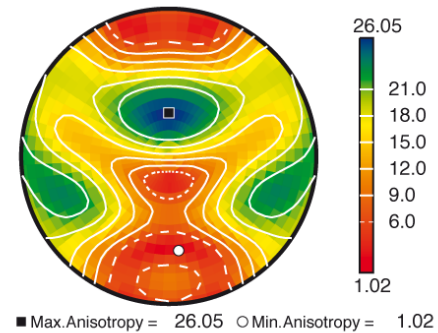
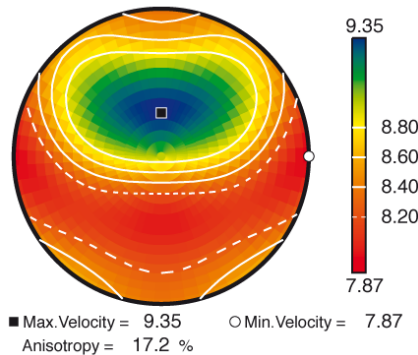
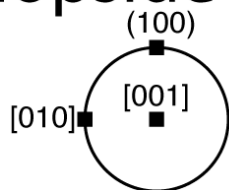
Olivine



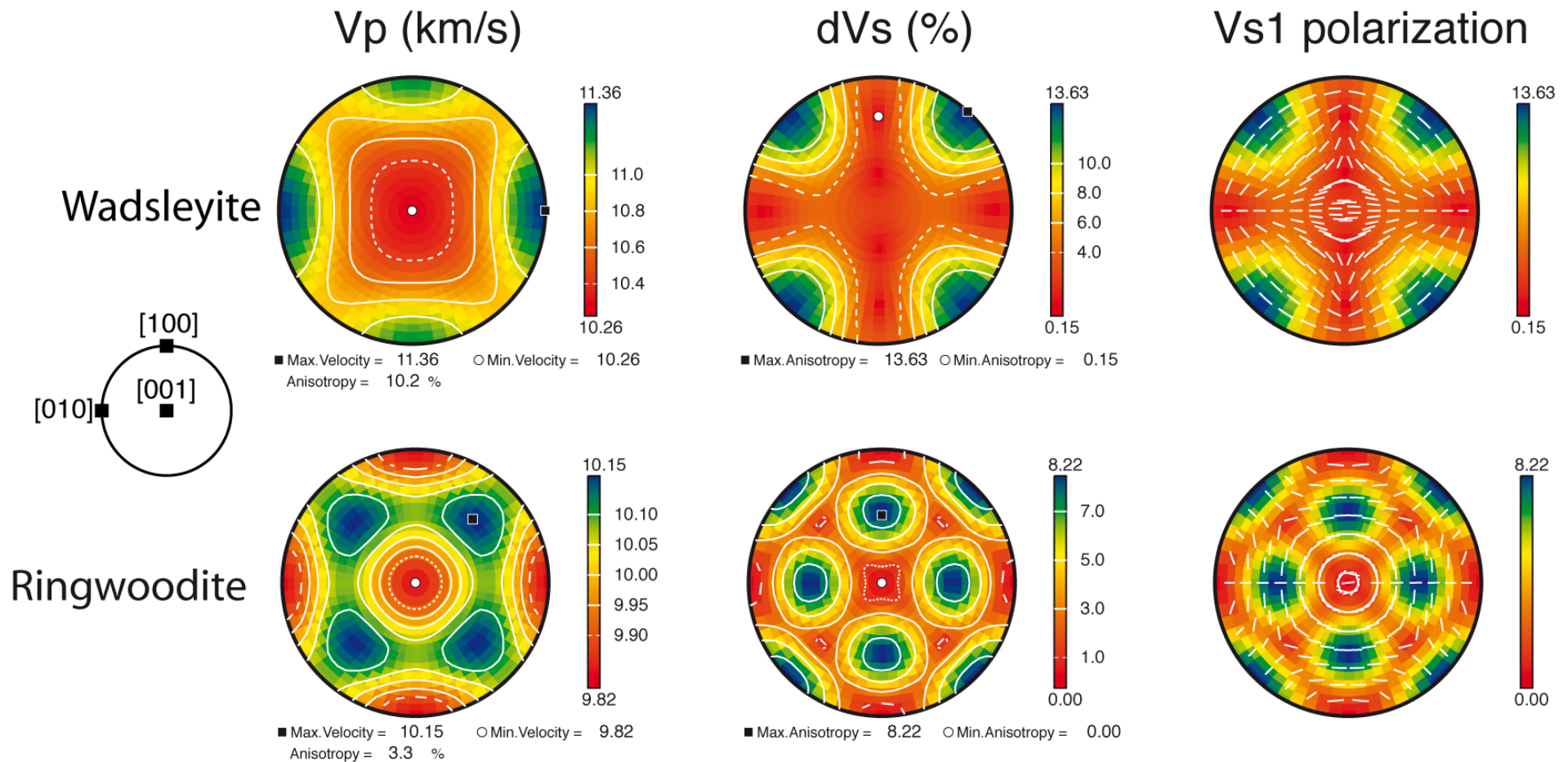
Enstatite



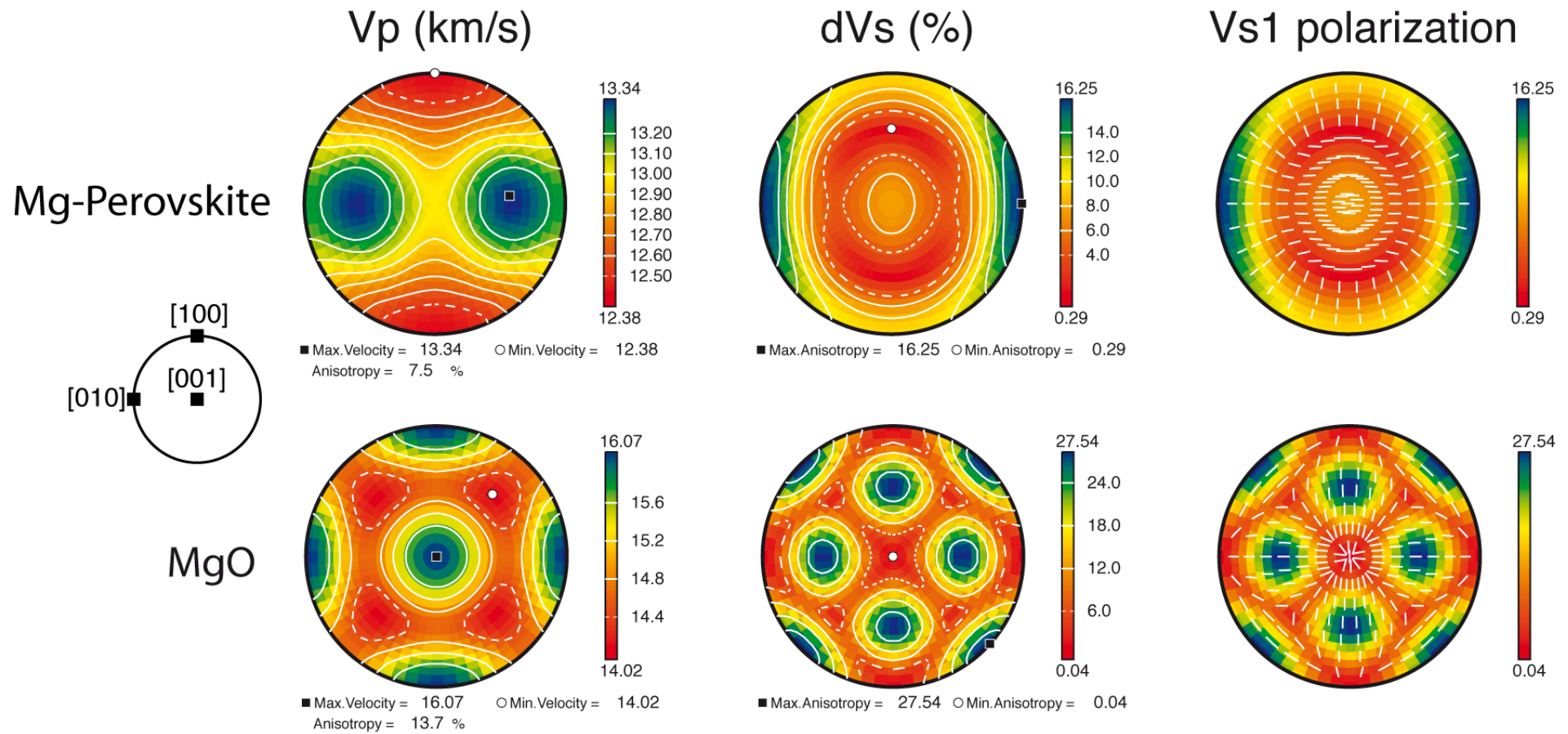
Diopside



# Transition zone (410-670 km)

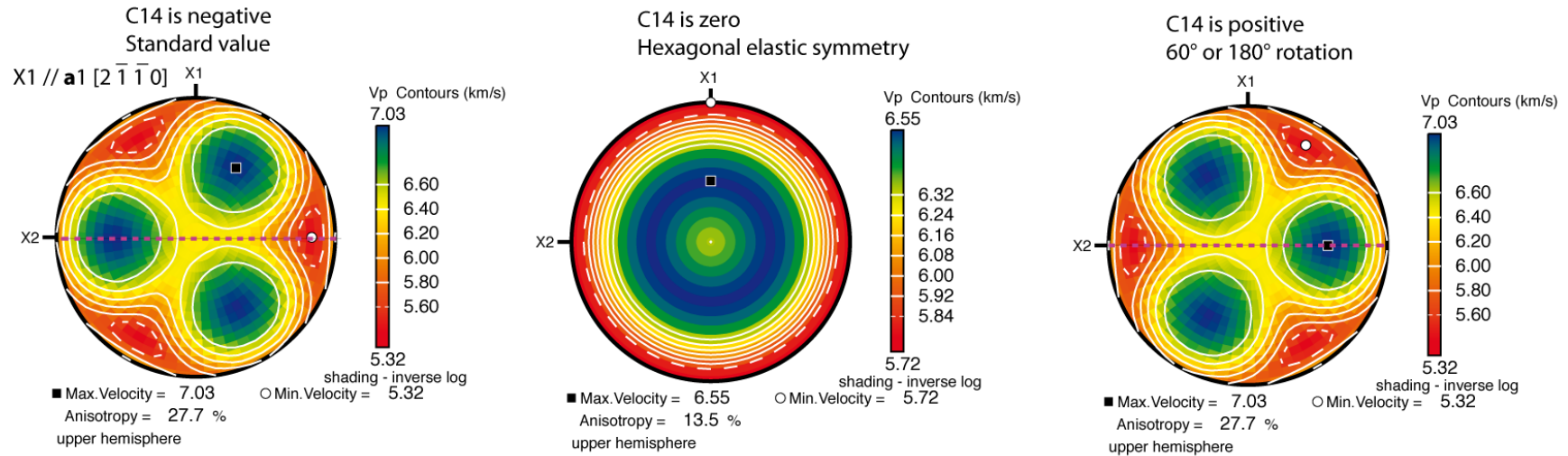


# Lower Mantle (670-2800 km)

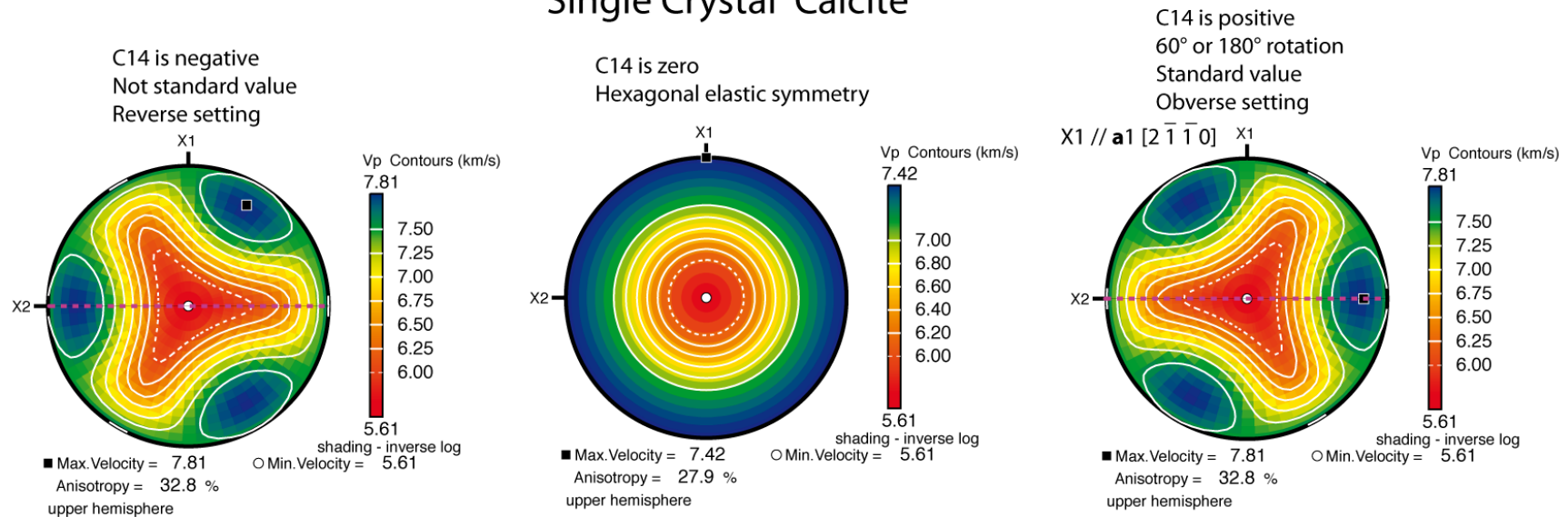


# Trigonal/Rhombohedral C14

## Single Crystal $\alpha$ -Quartz



## Single Crystal Calcite

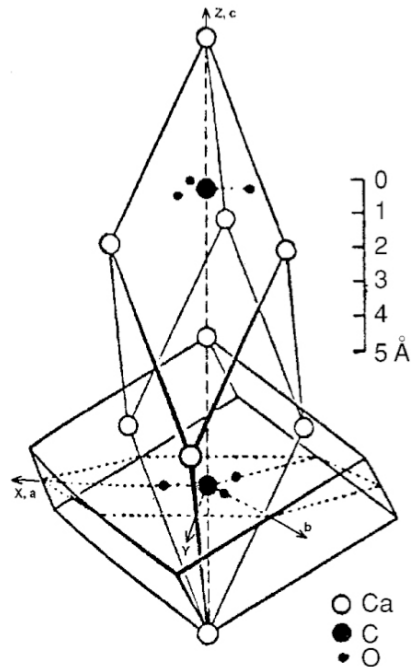


# Background for calcite 1

Table

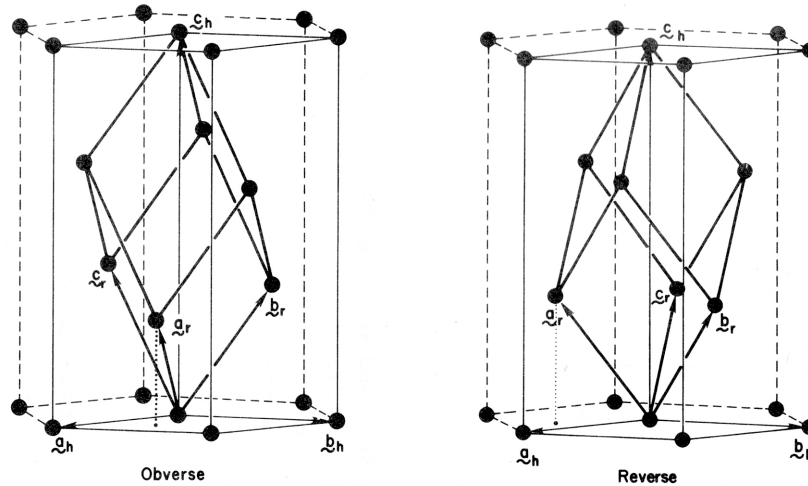
Computed and experimental stiffness coefficients for calcite. See Section 3.2 for the explanation of the different signs of  $C_{14}$ .

$C_{11}$	$C_{12}$	$C_{13}$	$C_{14}$	$C_{33}$	$C_{44}$	(GPa)
159	70	63	+20	98	39	calculated ab-initio with data in Table 3
144	54	51	-21	84	34	exptl. from Landoldt-Börnstein [16]

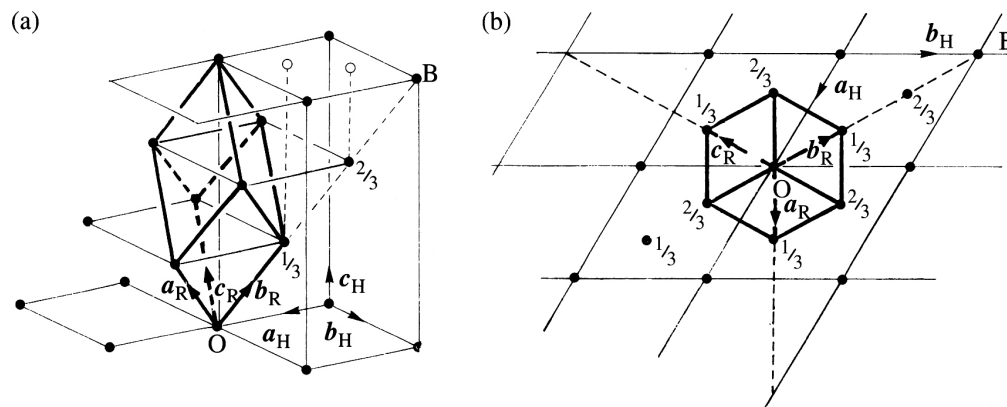


Composite of Fig. 1 from Kaga [18] and Fig. 140 p. 292 in [26]. In the axes shown on the Figure, the slender primitive rhombohedron indexes as form  $\{01\bar{1}1\}$  rather than  $\{10\bar{1}1\}$  as it should for an obverse setting of the rhombohedral lattice in the hexagonal axes. Kaga [18] (and therefore all other post-1949 authors who report a negative value for  $C_{14}$ ) then used the reverse setting of the rhombohedral lattice

# Background for calcite 2



Obverse setting to Reverse is  $180^\circ$  rotation about the c-axis



Rhombohedral lattice. The basis of the rhombohedral cell is labelled  $a_R$ ,  $b_R$ ,  $c_R$ , the basis of the hexagonal centred cell is labelled  $a_H$ ,  $b_H$ ,  $c_H$  (numerical fractions are calculated in terms of the  $c_H$  axis). (a) Obverse setting; (b) the same figure as in (a) projected along  $c_H$ .



## References – Tensors and Crystal Physics

- Bhagavantam, S., Crystal Symmetry and Physical Properties , London, New York, Academic Press (1966)
- Cady, W. G. Piezoelectricity: An Introduction to the Theory and Applications of Electromechanical Phenomena in Crystals, New rev. ed., 2 vols. New York: Dover, 1964.
- Curie, P., Oeuvres , pp. 118, Paris, Société Française de Physique (1908)
- Hartmann,E. 1984, An Introduction to Crystal Physics, International Union of Crystallography Teaching Pamphlet 18
- <http://www.iucr.org/iucr-top/comm/cteach/pamphlets.html>
- D.R.Lovett Tensor properties of Crystals, Institute of Physics Publishing, Bristol and Philadelphia, 2nd Edition 1999
- **D.McKie and C.McKie Essentials of Crystallography, Blackwell Scientific Punlishing, 1986**
- **Mainprice, D. 2007 Seismic anisotropy of the deep Earth from a mineral and rock physics perspective. Schubert, G. ‘Treatise in Geophysics’ Volume 2 pp 437-492. Oxford: Elsevier.**
- **Nye, J. F., Physical Properties of Crystals , Oxford, Clarendon Press (1957)**
- **A.Putnis, 1992. Introduction to Mineral Sciences, Cambridge University Press (Chapter 2)**
- **Wenk, H.-R. and Bulakh,A., 2004. Minerals Their Constitution and Origin, Cambridge University Press. (Chapter 8)**

# Useful web pages

- **AGU Bookshelf Reference Books**

<http://www.agu.org/reference/minphys.html>

- **Bilbao Crystallographic server (interactive)**

<http://www.cryst.ehu.es/>

- **American Mineralogist Table of Contents (old reference are free!)**

<http://www.minsocam.org/MSA/AmMin/TOC/>

- **American Mineralogist Crystal Structure Database (PDFs of publications are often associated with data files \*.cif etc)**

<http://rruff.geo.arizona.edu/AMS/amcsd.php>

- **American Mineralogist Mineral PDFs**

<http://rruff.geo.arizona.edu/doclib/hom/>

- **Crystallography Open Database**

<http://cod.crystallography.net/>

- **Russian Crystallographic & chemical Database**

<http://database.iem.ac.ru/mincryst/search.php?>

- **Dictionary of Crystallography**

[http://reference.iucr.org/dictionary/Main\\_Page](http://reference.iucr.org/dictionary/Main_Page)

# CAREWARE



- Presently available [Unicef](#) CAREWARE programs
- For [Mac](#) and [Windows](#) :
- link [ftp://www.gm.univ-montp2.fr/mainprice//CareWare\\_Unicef\\_Programs/](ftp://www.gm.univ-montp2.fr/mainprice//CareWare_Unicef_Programs/)
- <http://www.gm.univ-montp2.fr/PERSO/mainprice/>
- Pole Figures
- Inverse Pole Figures
- Anisotropic Elastic and Seismic properties
- Second Order Symmetric Tensor Properties
- Field work (az/inc) pole figures
- All programs have a simple question and answer interface (no graphical interface with pull down menus) and are intended to produced publication or presentation ready graphics in [Adobe Illustrator](#) postscript.

# Pole figure programs for data in Euler angle triplets

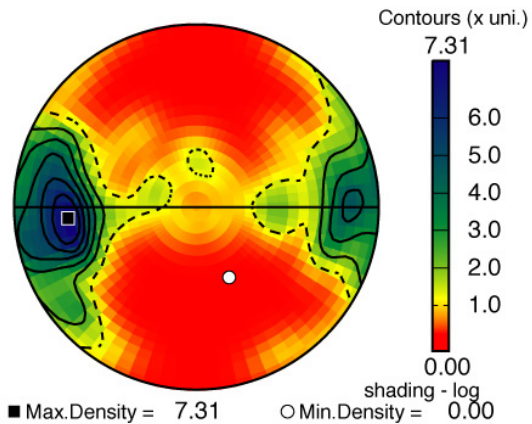
PF - general purpose program for Euler angle triplets

PFctf - for Channel 5 ASCII text export files (\*.ctf)

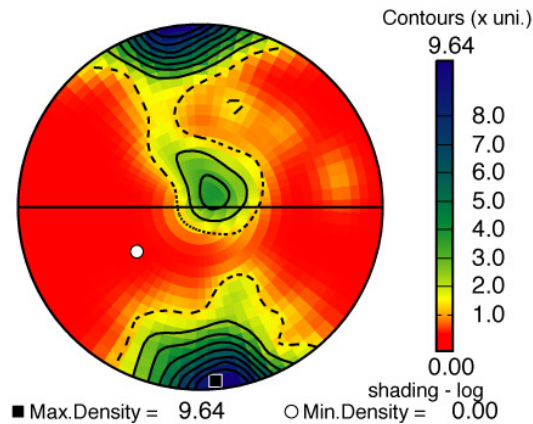
## Olivine pole figures

900A65A

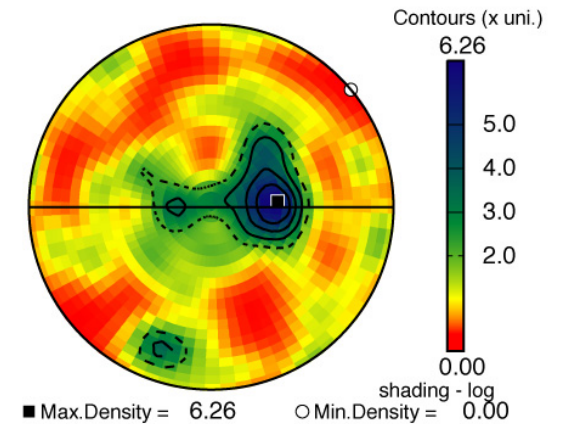
[100]



[010]



[001]

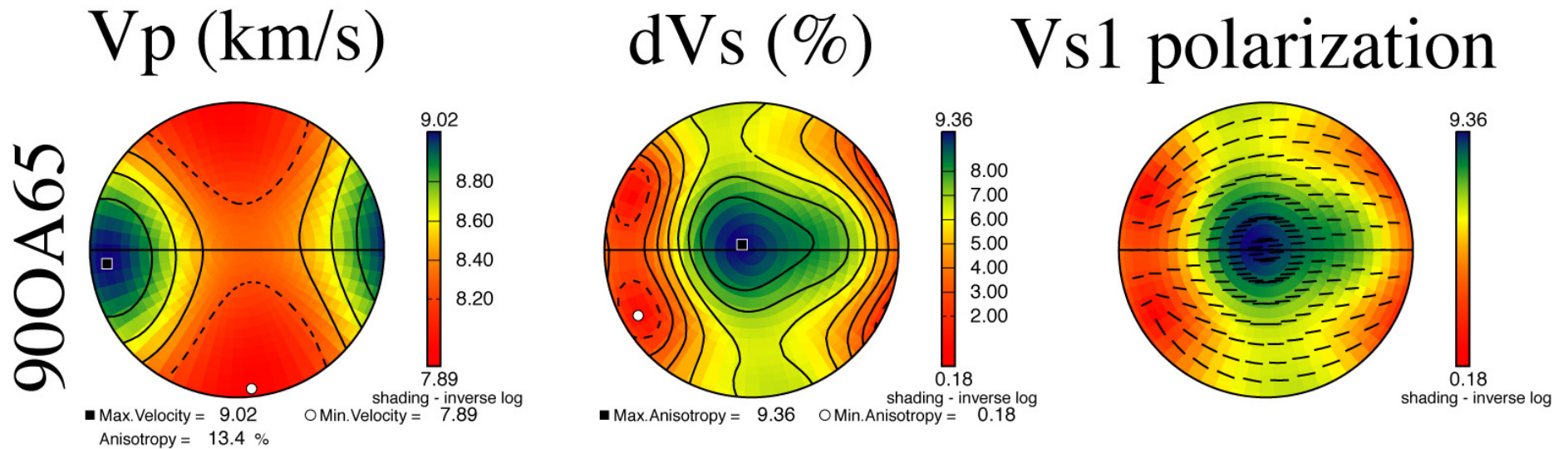


# Seismic and elastic tensor properties for data in

## Euler angle triplets

Anis2k - data in Euler angle triplets

Anisctf - for Channel 5 ASCII text export file (\*.ctf)



# Anisotropic physical properties of rocks

## Part II : Estimation of the anisotropic physical properties of rocks

### 2010

*Introduction*

*Effective medium and anisotropy*

*Quantitative texture analysis and calculation of physical properties*

*Example : Olivine to Antigorite transformation*

# Introduction

- The calculation of the physical properties from microstructural information (crystal orientation, volume fraction, grain shape etc.) is important for rocks because it gives insight into the role of microstructure in determining the bulk properties.
- A calculation can be made for the in-situ state at high temperature and pressure for samples where the microstructure has been changed by subsequent chemical alteration (e.g. the transformation olivine to serpentine) or mechanically induced changes (e.g. fractures created by decompression). The in-situ temperatures and pressures can be simulated using the appropriate single crystal derivatives.
- Additional features not necessarily preserved in the recovered microstructure, such as the presence of fluids (e.g. magma) can be modeled.
- The effect of phase change on the physical properties can also be modeled using these methods.
- Modeling is essential for anisotropic properties as experimental measurements in the many directions necessary to fully characterize anisotropy is not currently feasible for the majority of the temperature and pressure conditions found in the deep Earth.

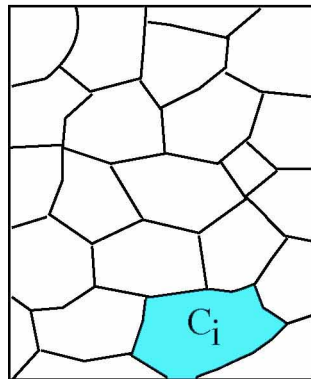
# Effective medium and Anisotropy



# Microscopic and Macroscopic

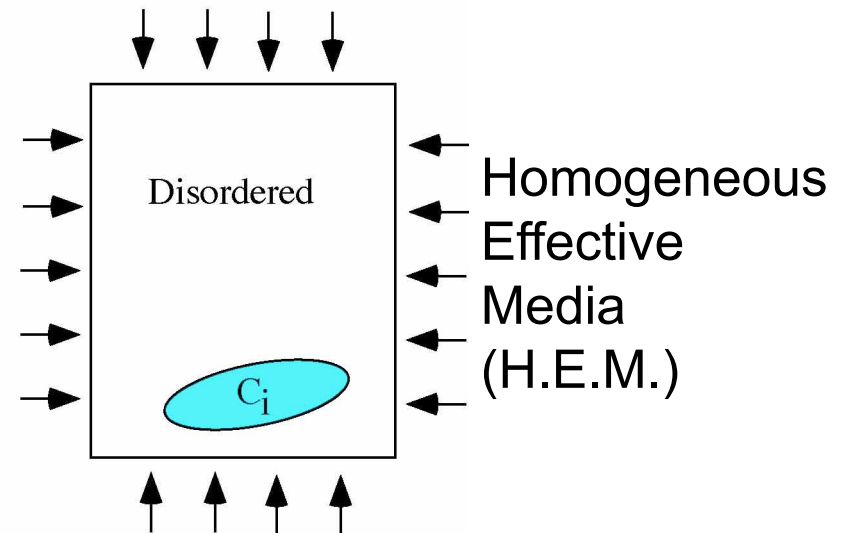
Microscopic

$$C_i = \sigma_i e_i^{-1}$$



Macroscopic

$$C^* = \langle \sigma \rangle \langle e \rangle^{-1}$$

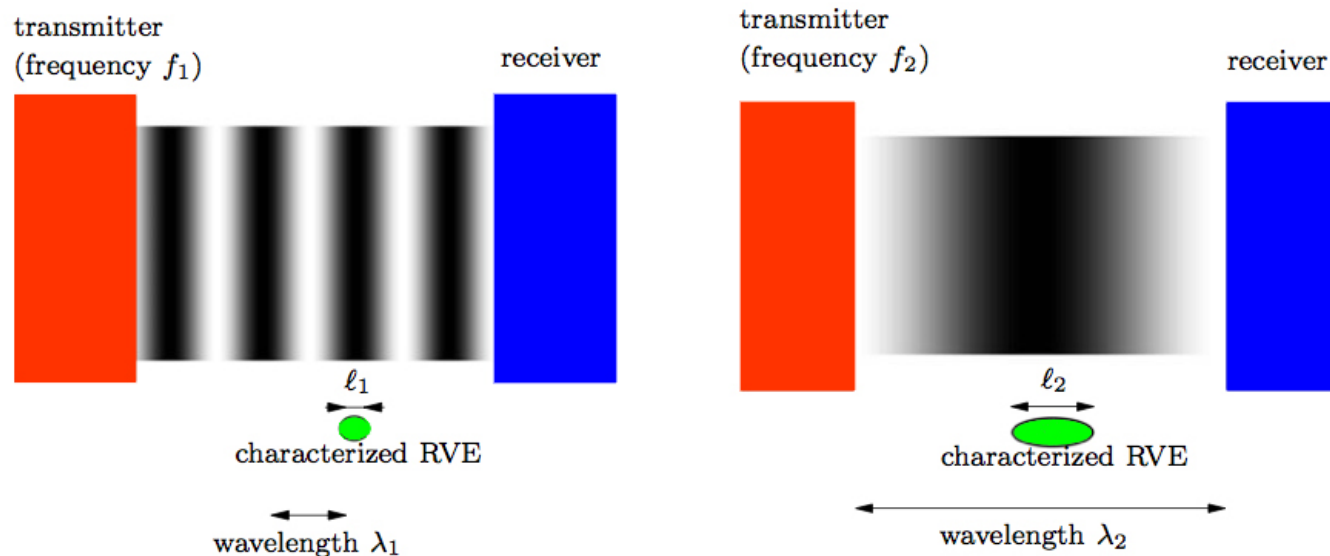


$C^*$  = effective medium elastic constants

$$\langle \sigma \rangle = \text{volume averaged stress} = \sum v_i \sigma_i = \sum v_i (C_i e_i)$$

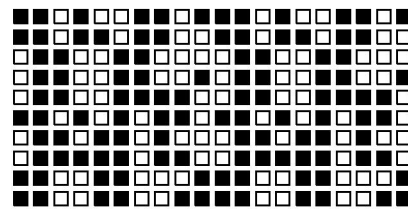
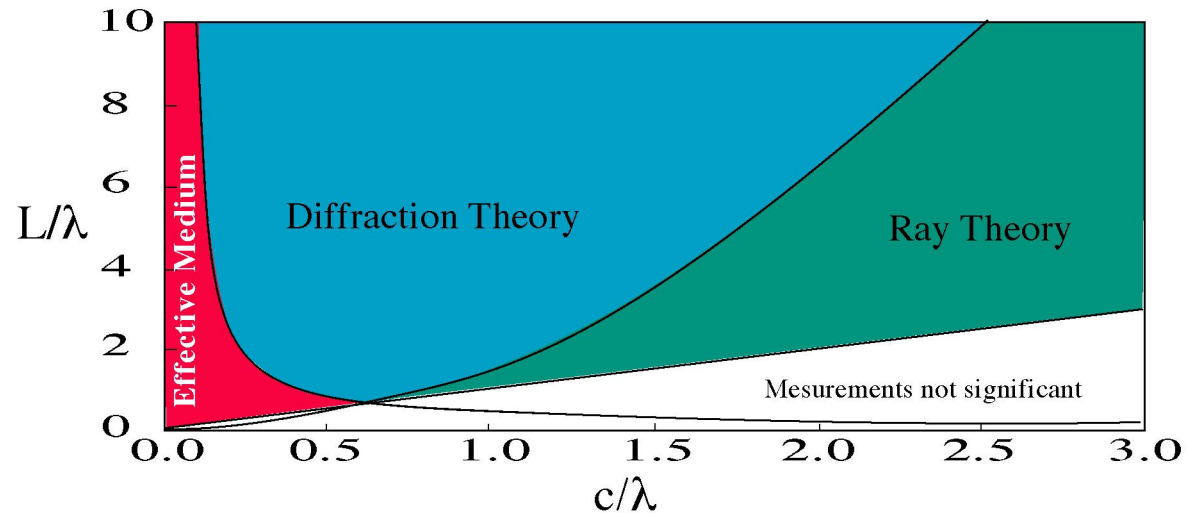
$$\langle e \rangle = \text{volume averaged strain} = \sum v_i e_i$$

# Frequency, Wavelength and Representative Volume Element (RVE)



Schematic, grey-scale based illustration of stress magnitude in specimens tested ultrasonically with different frequencies ( $f_1 > f_2$ ) (Fritsch and Hellmich 2007)

# Length scale and effective medium

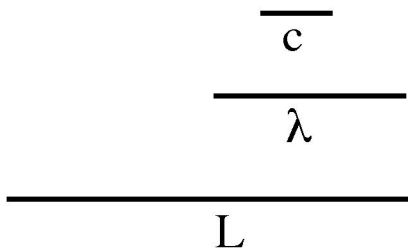


Effective medium  $L > \lambda > c$

correlation length (motif)

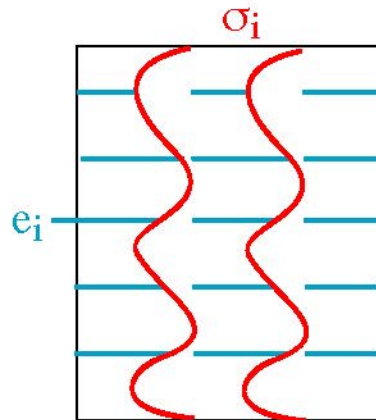
dimension of physical phenomenon  
(e.g. seismic wavelength)

dimension of specimen  
(propagation distance)



# Simple volume averages

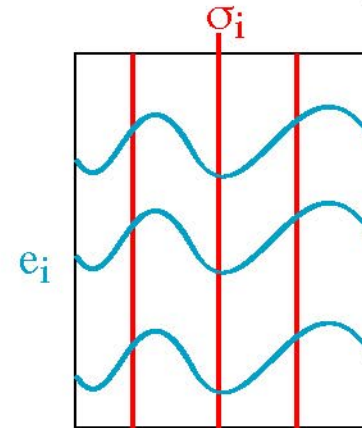
Voigt Average



$$e_i = \text{constant}: C^* \langle \sigma \rangle = \sum V_i (C_i e_i)$$

$$C^* \approx C^{\text{Voigt}} = \sum V_i C_i$$

Reuss Average



$$\sigma_i = \text{constant}: C^* \langle e \rangle^{-1} = (\sum V_i e_i)^{-1} = (\sum V_i (S_i \sigma_i))^{-1}$$

$$C^* \approx C^{\text{Reuss}} = (\sum V_i S_i)^{-1}$$

- easy to calculate
- widely separated bounds for strongly anisotropic minerals
- cannot introduce microstructure e.g. shape
- very poor bounds for mixtures with very different stiffnesses e.g. solids, liquids and voids

## Relationships between Voigt and Reuss Averages

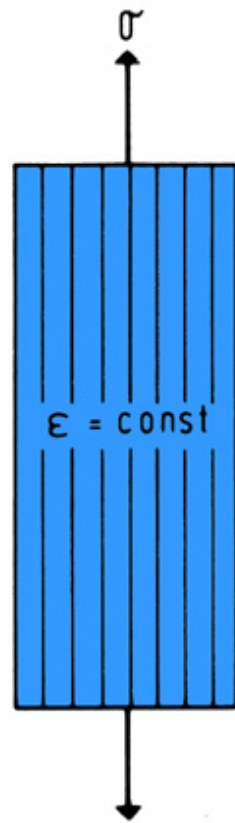
$$C^* \approx C^{\text{Voigt}} = \left[ \sum_i V_i C(\mathbf{g}_i) \right]$$

$$C^* \approx C^{\text{Reuss}} = \left[ \sum_i V_i S(\mathbf{g}_i) \right]^{-1}$$

$$S^* \approx S^{\text{Reuss}} = \left[ \sum_i V_i S(\mathbf{g}_i) \right]$$

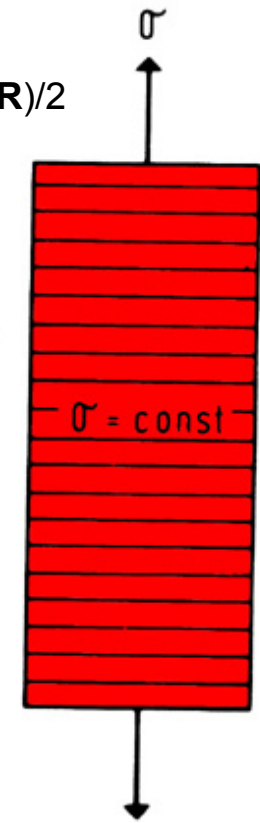
$$C^{\text{Voigt}} \neq C^{\text{Reuss}} \text{ and } C^{\text{Voigt}} \neq [S^{\text{Reuss}}]^{-1}$$

# Voigt & Reuss micro-structural models



Voigt-Reuss-Hill (VRH) or Hill =  $(V+R)/2$

No microstructural or theoretical basis, but in practice it is close to the experimentally measured values

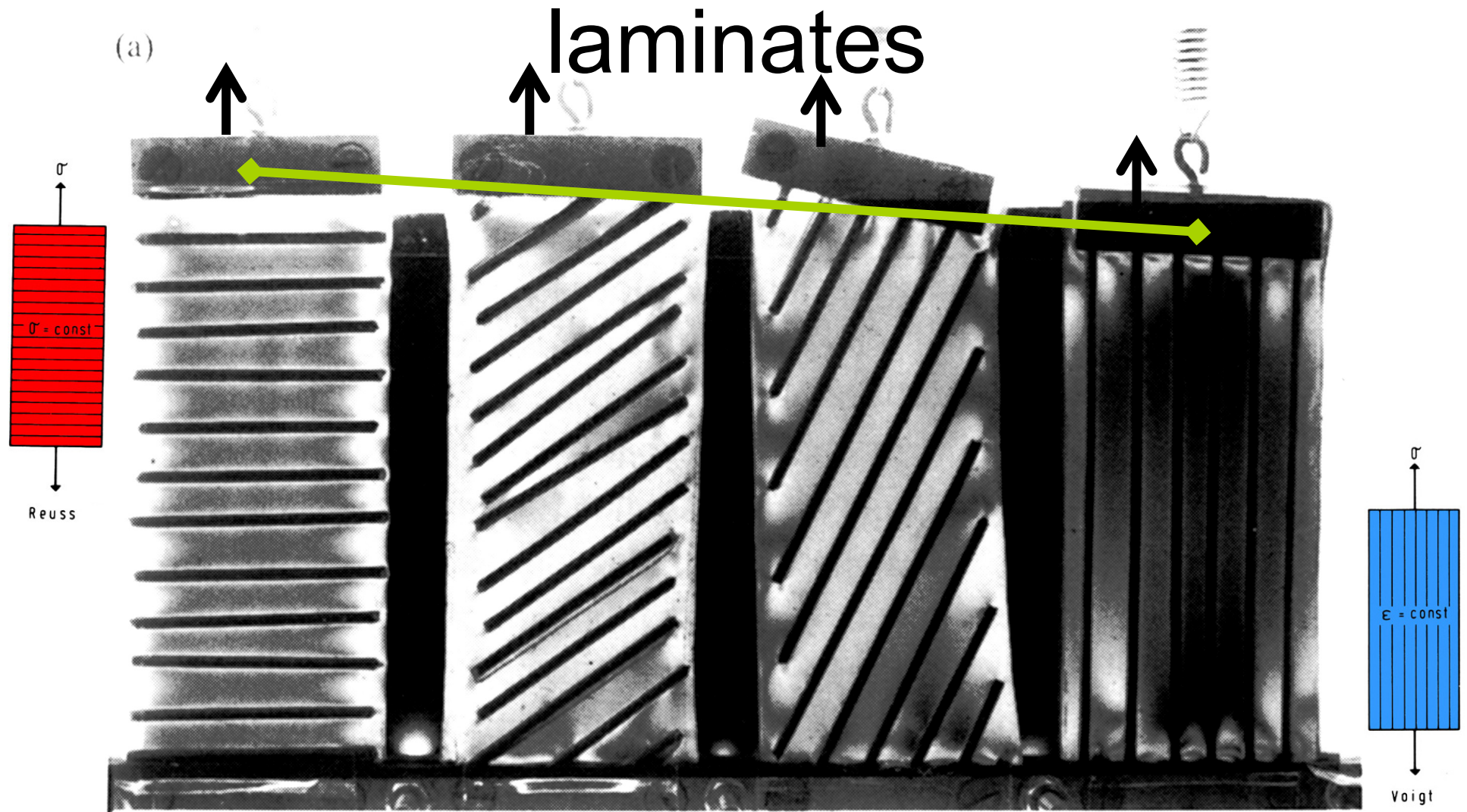


Voigt (a)

Reuss (b)

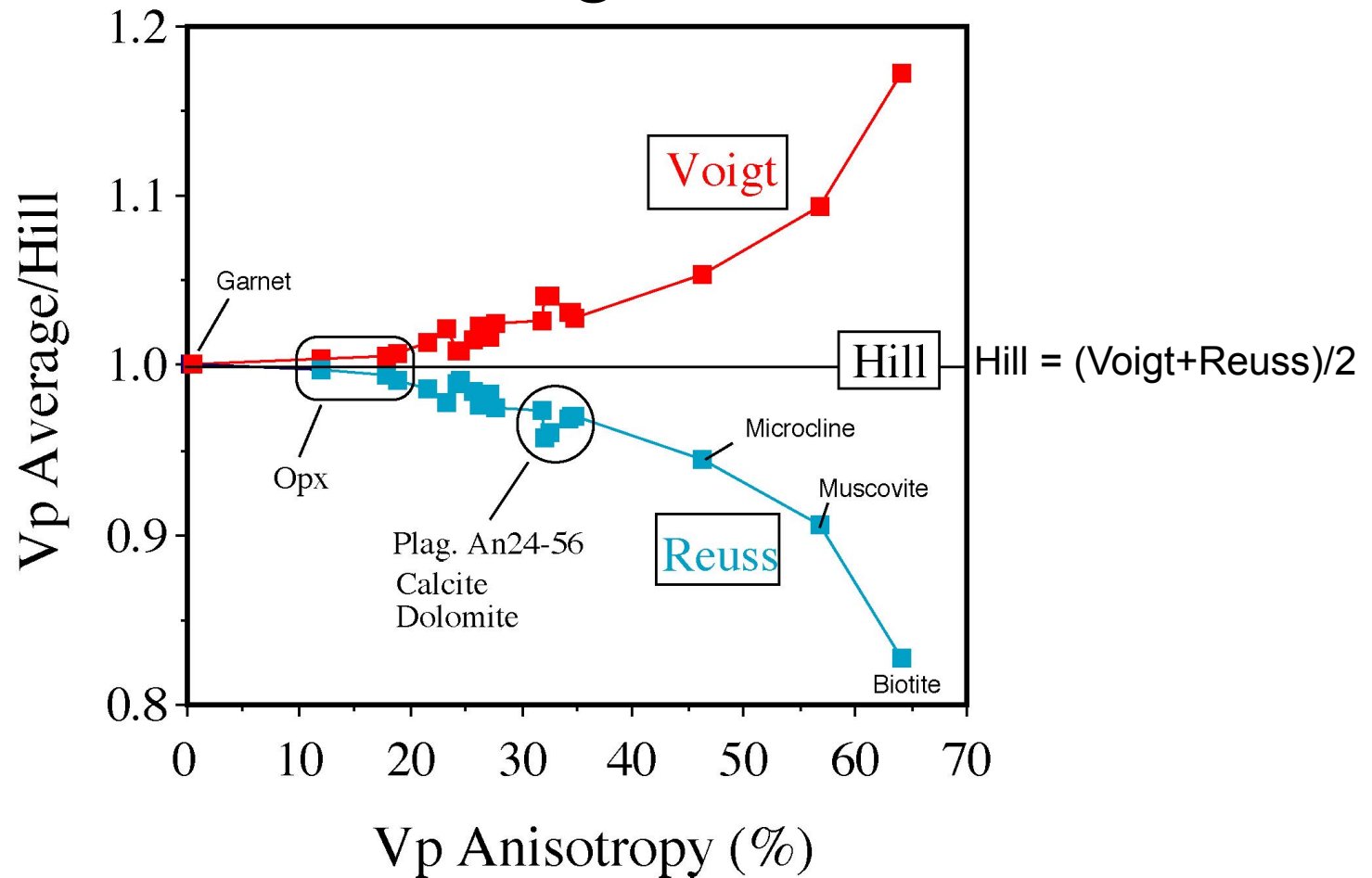
(a) Grain structure of a polycrystal for which Voigt's assumption is valid. (b) Grain structure of a polycrystal for which Reuss's assumption is valid.

# Macromodel composite



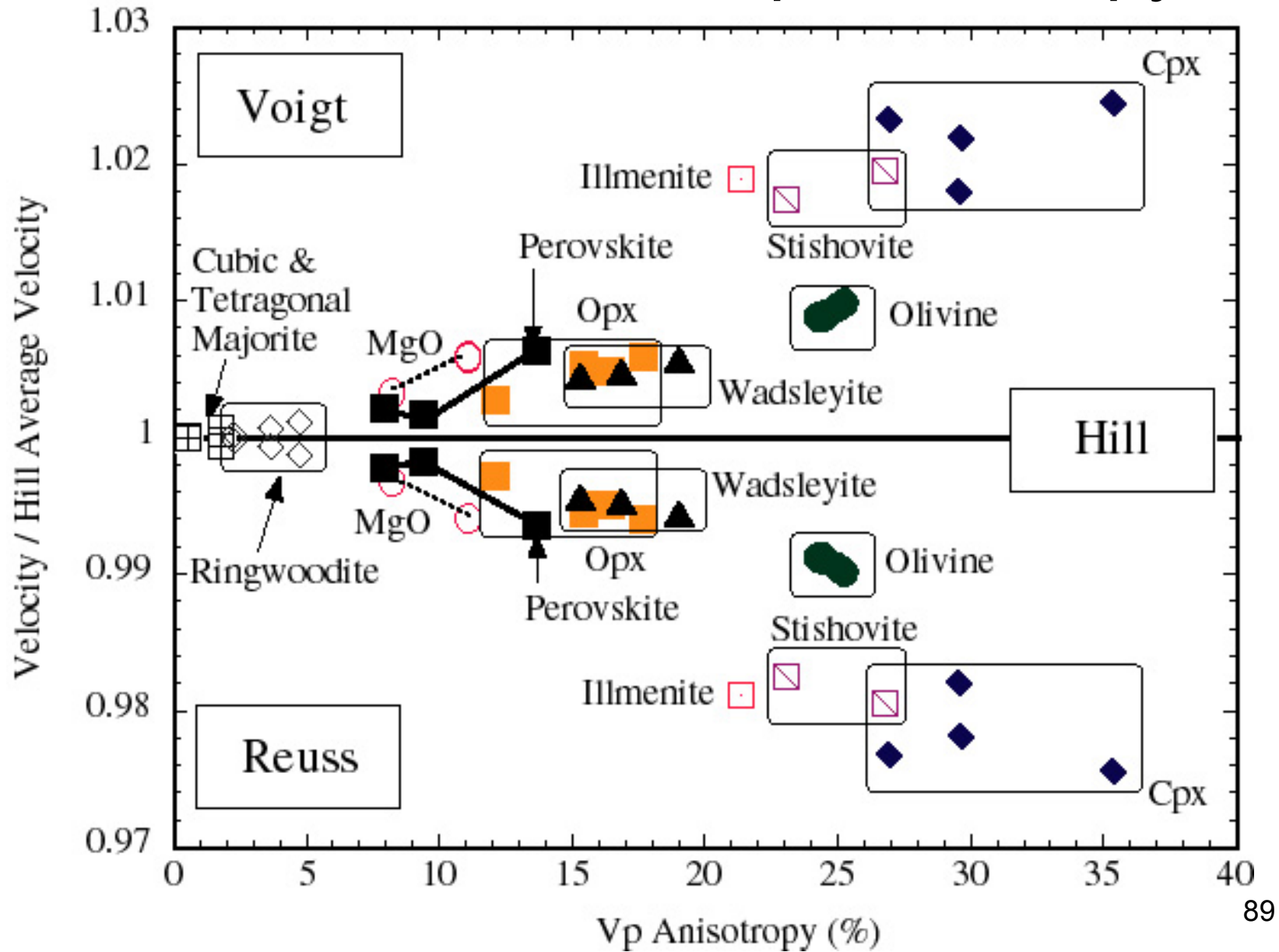
Aligned metal rods(black) in polyurethane matrix grey  
where lighter greys - larger shear strains

# Vp Randomly Oriented Polycrystal Averages

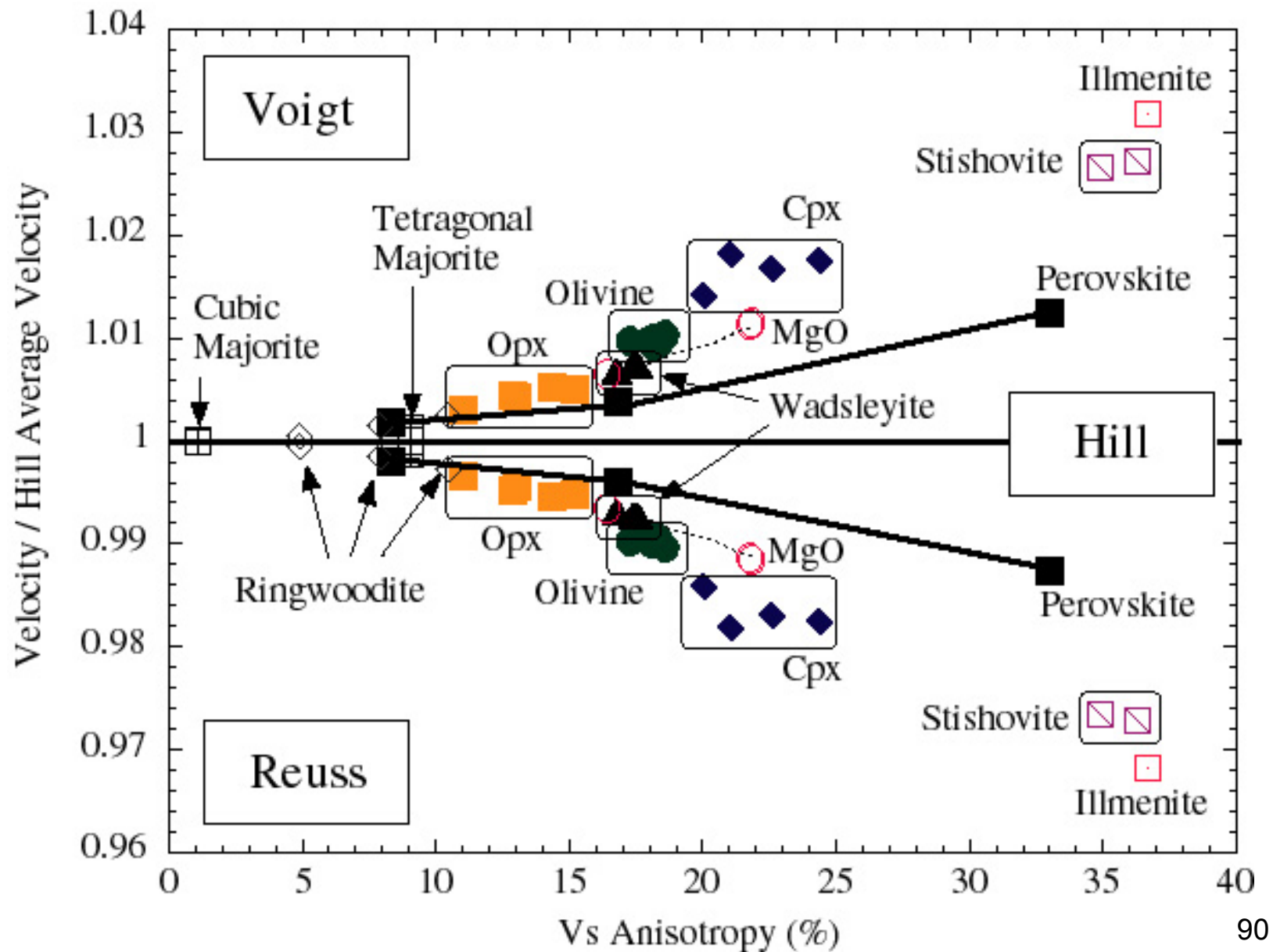




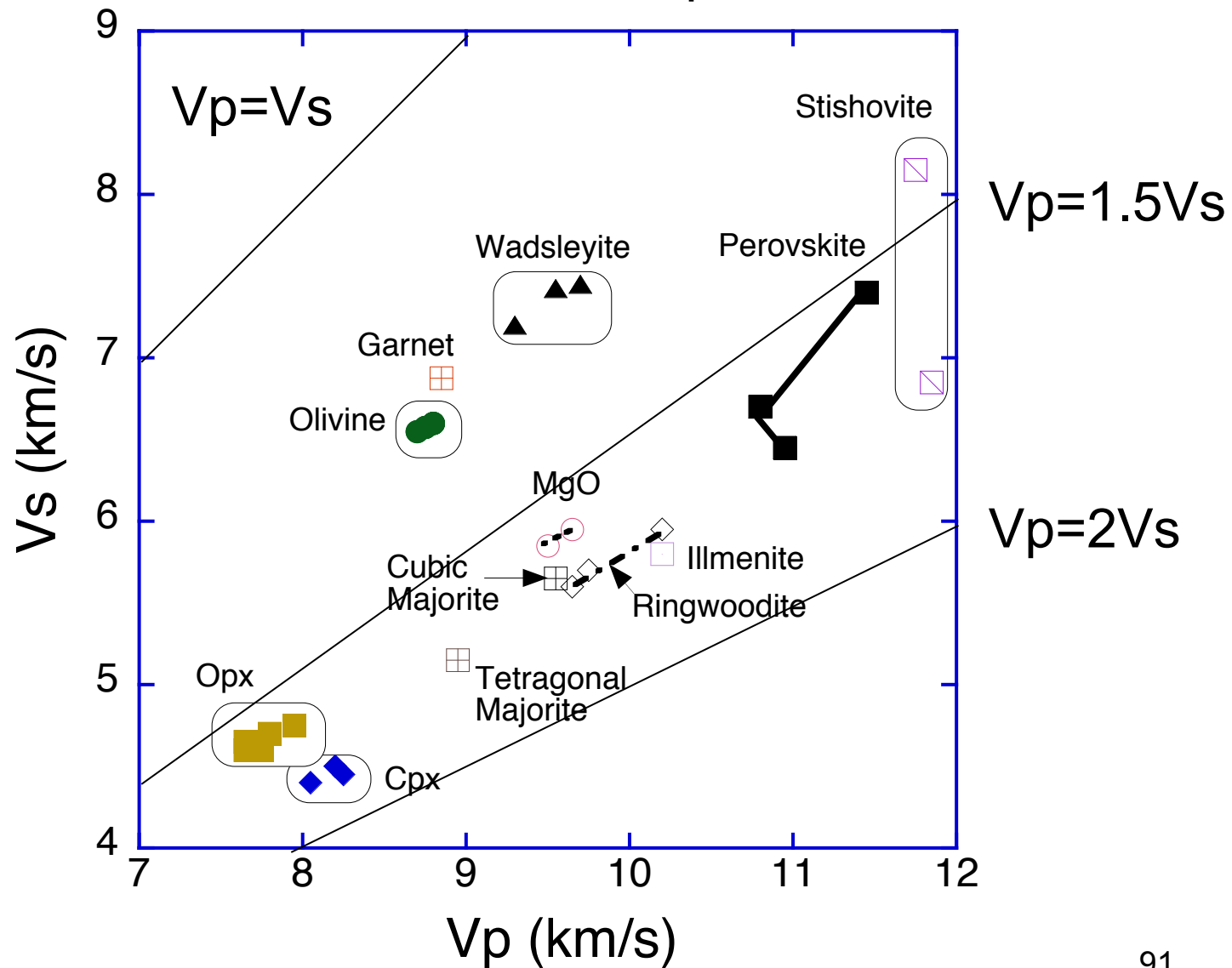
# Mantle Minerals: Vp Anisotropy



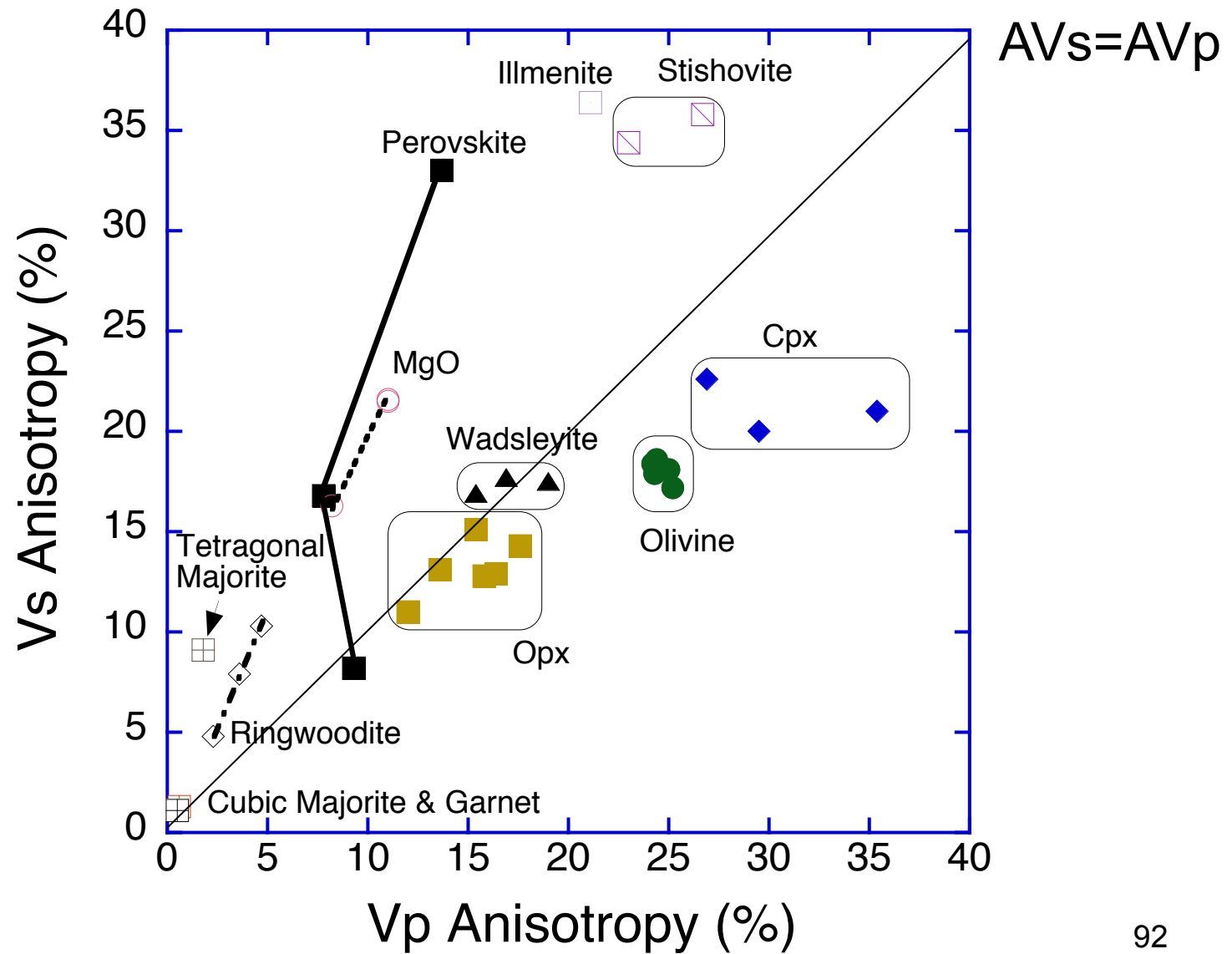
# Mantle Minerals: Vs anisotropy



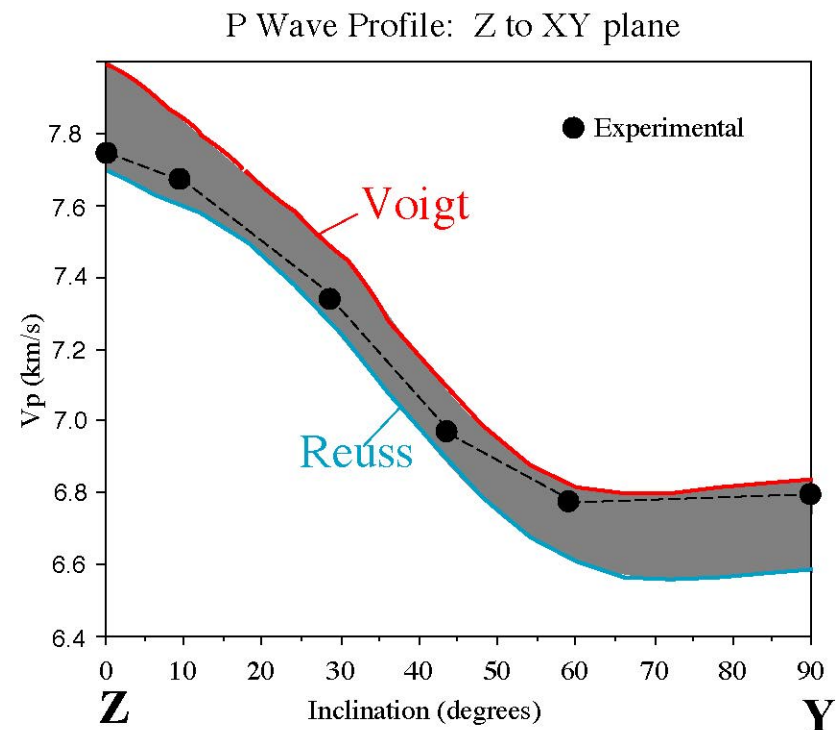
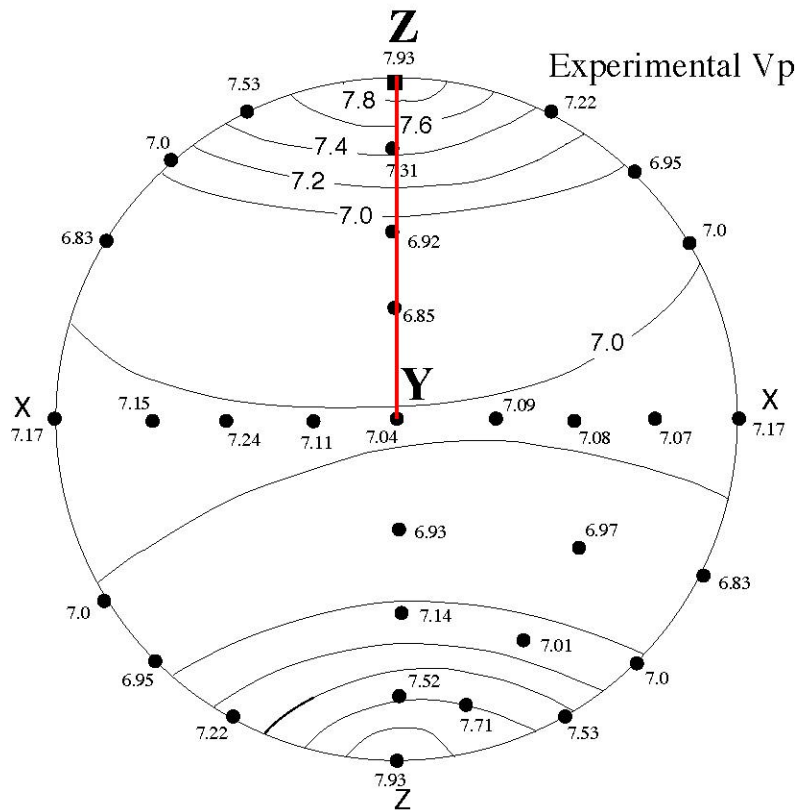
# Mantle Minerals: Vp & Vs



# Mantle Minerals: Vp & Vs anisotropy

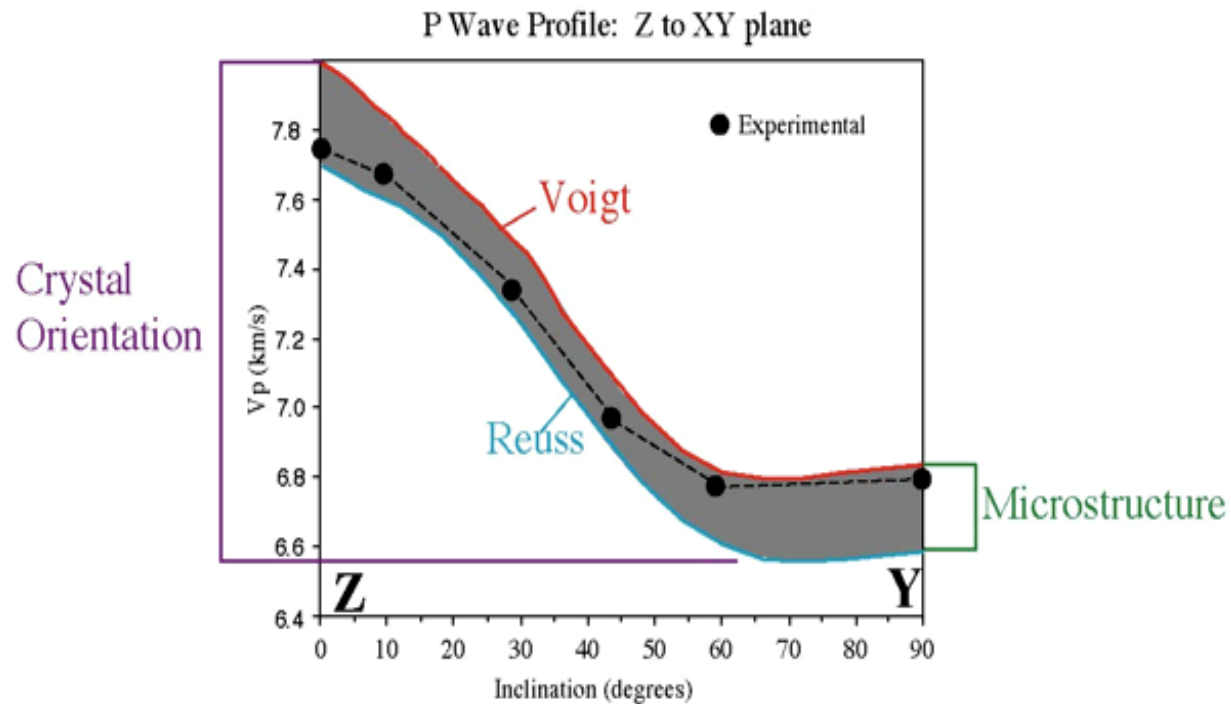


# Oklahoma Gabbro Stereo & Vp YZ profile



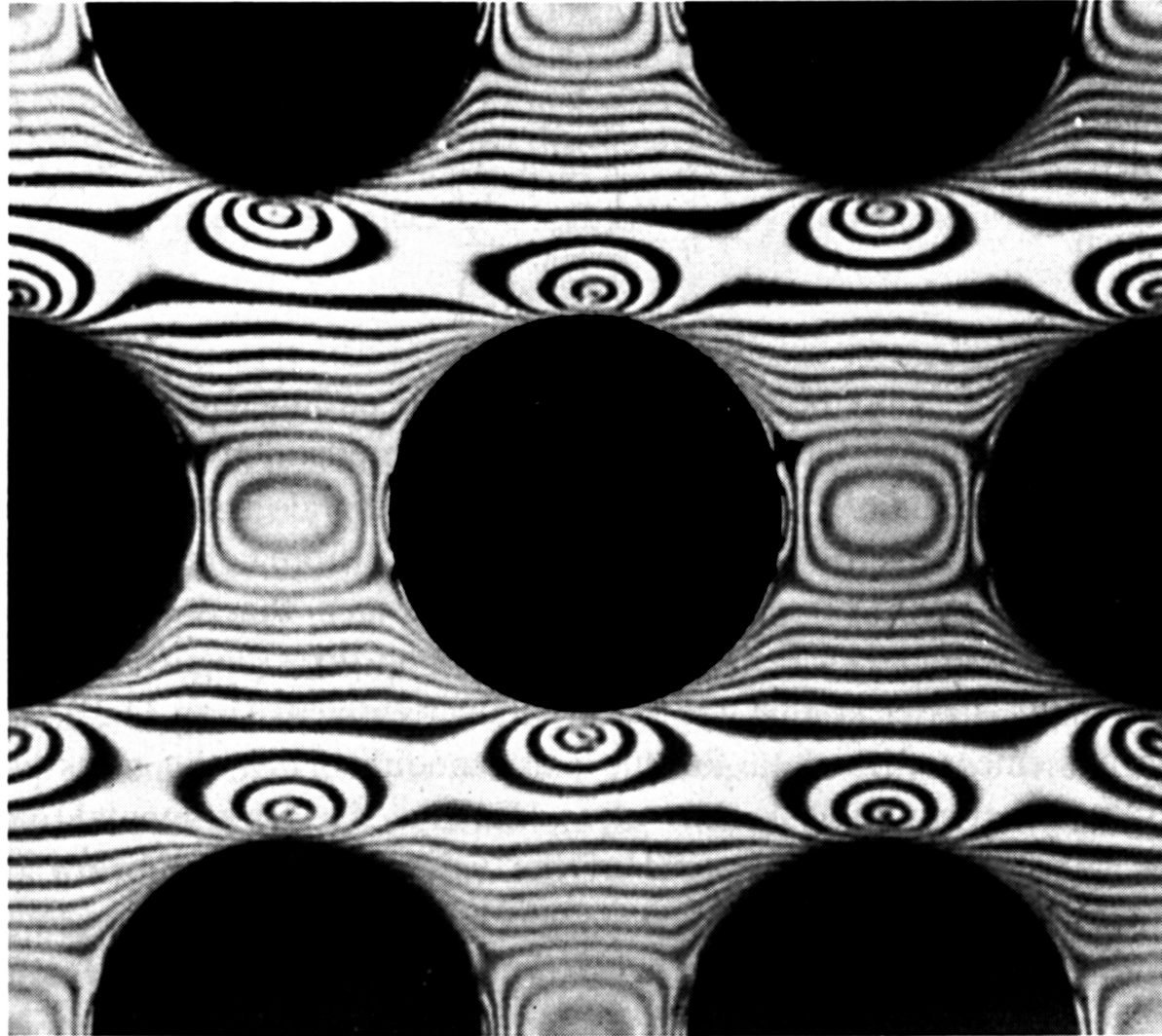
Confining pressure = 800 MPa

## Physical Properties & Microstructure - Composite materials approach



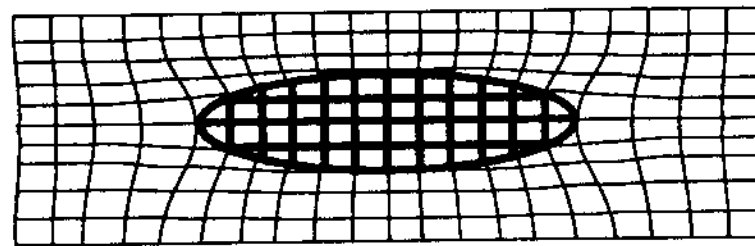
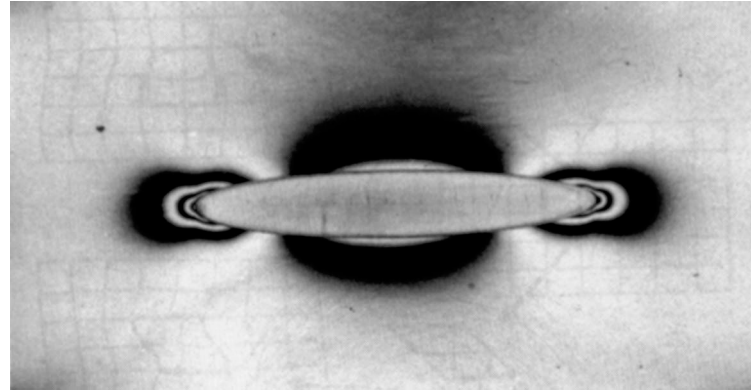
Crystal Orientation "signal" = 5 x Microstructure "signal"

# Interaction between inclusions



A photoelastic image (Puck 1967), showing isochromatic (equal bi-refringence) fringes for a macromodel composite loaded in transverse tension (vertical direction).

# Ellipsoidal inclusion - special properties



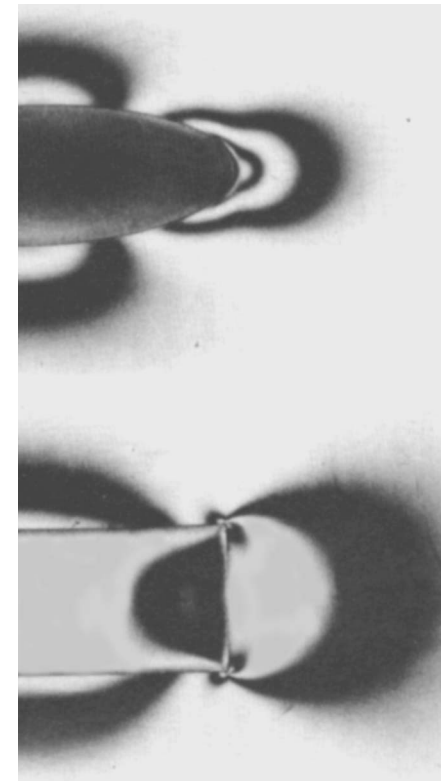
Uniform stress and strain field within the inclusion



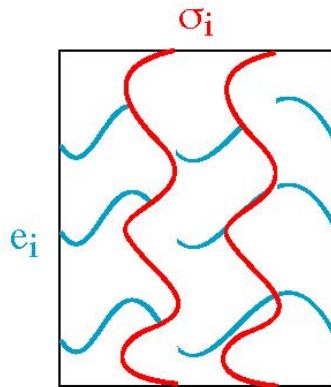
# Ellipsoid and Cylinder

homogeneous stress & strain fields

heterogeneous stress & strain fields



# Self Consistent Method



$$e_i \neq \text{constant} \quad \sigma_i \neq \text{constant} \quad : \quad C^* = \langle \sigma \rangle \langle e \rangle^{-1}$$

$$C^* = \langle \sigma \rangle \langle e \rangle^{-1}$$

$$\langle \sigma \rangle = \sum v_i (C_i e_i) \quad \langle e \rangle = \sum v_i e_i$$

$$C^* \approx C^{SC} = \langle \sigma \rangle \cdot \langle e \rangle^{-1} = [\sum v_i (C_i e_i)] \cdot [\sum v_i e_i]^{-1}$$

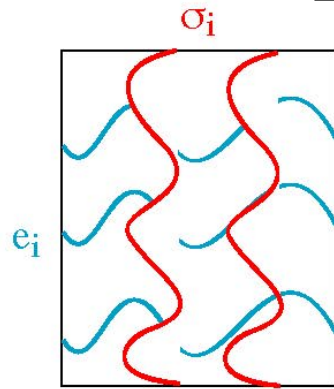
The value of  $e_i$  is found using a generalization of Eshelby's inclusion theory to anisotropic inclusions in anisotropic back ground media.

$$e_i = [I + G (C_i - C^*)]^{-1}$$

Where  $I$  is the 4<sup>th</sup> rank identity tensor and  $G$  is a Green's tensor involving elliptical integrals over the inclusion shape.

- 
- much more complex to calculate
  - "best" bounds for strongly anisotropic minerals
  - can introduce microstructure e.g. shape via Green's tensor as ellipsoidal inclusion
  - "best" bounds for mixtures with very different stiffnesses e.g. solids, liquids and voids
  - treats every object (grain, void, fracture) in an identical manor
  - further information about neighbour interaction could be introduced via two-point correlation functions

# DEM Method



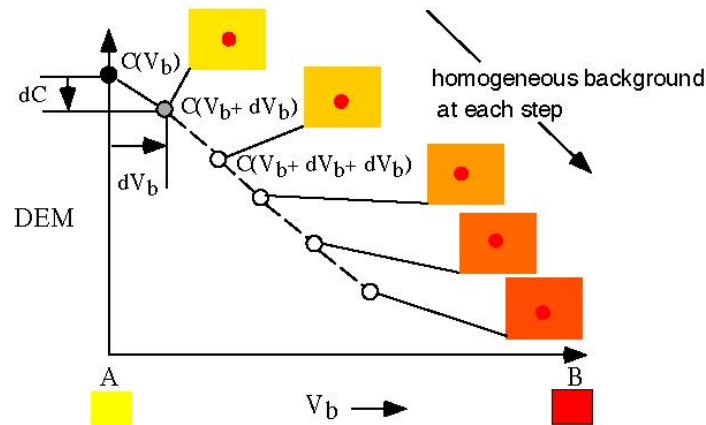
$$e_i \neq \text{constant} \quad \sigma_i \neq \text{constant} \quad : \quad C^* = \langle \sigma \rangle \langle e \rangle^{-1}$$

$$C^* = \langle \sigma \rangle \langle e \rangle^{-1}$$

$$\langle \sigma \rangle = \sum v_i (C_i e_i) \quad \langle e \rangle = \sum v_i e_i$$

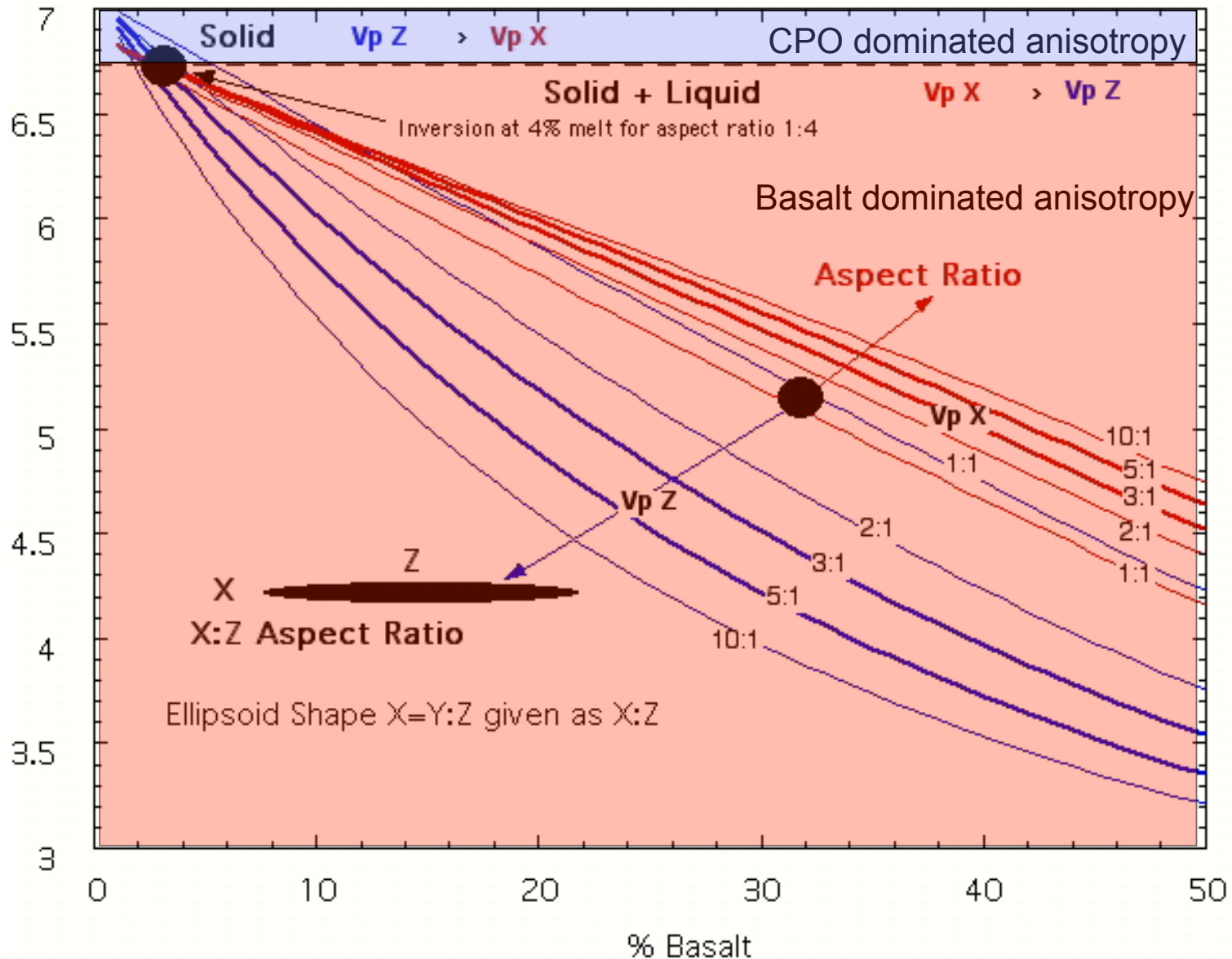
$$dC^* \approx dC^{\text{DEM}} = \frac{dV_b}{(1-V_b)} (C_i - C^{\text{DEM}}) e_i$$

Developed for two phase aggregates type AB (McLaughlin,1977) where  $V_b = 1-V_a$ . The phase B is the included phase and phase A is the host phase. Requires some initial value of  $C^*$  at a composition  $V_b$  - this may be a pure end member. The incremental nature of the differential approach preserves the percolating (connectivity) properties of the initial estimate. For example if starting from pure A, A will always be connected. If starting from 50:50 A:B then both phases will be connected.

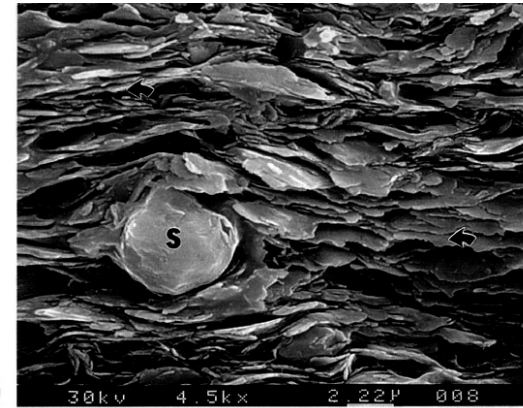


- now phases A and B are microstructurally different (A=Host, B= Inclusion)
- preserves connectivity of initial estimate

# VpX and VpZ Gabbro - Basalt : Effect of Aspect Ratio



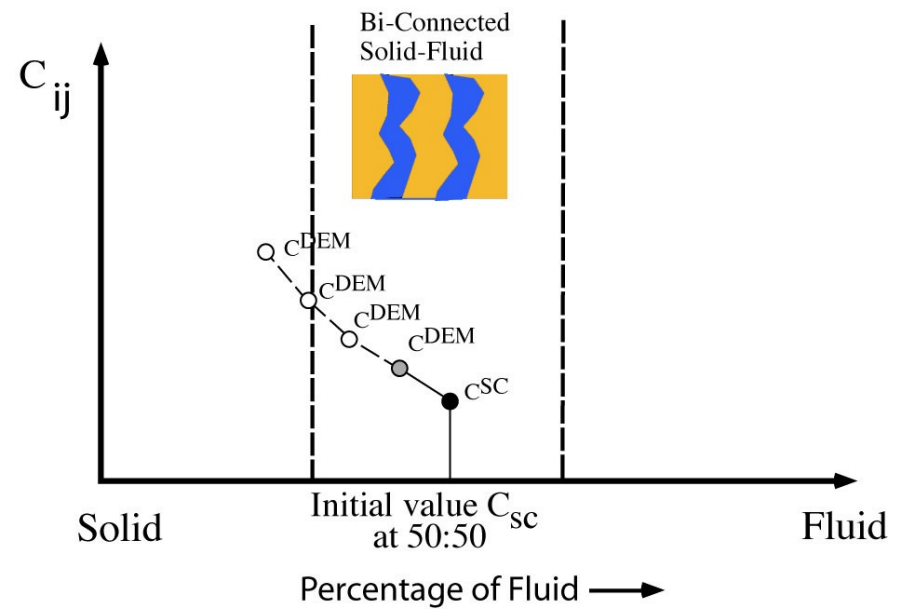
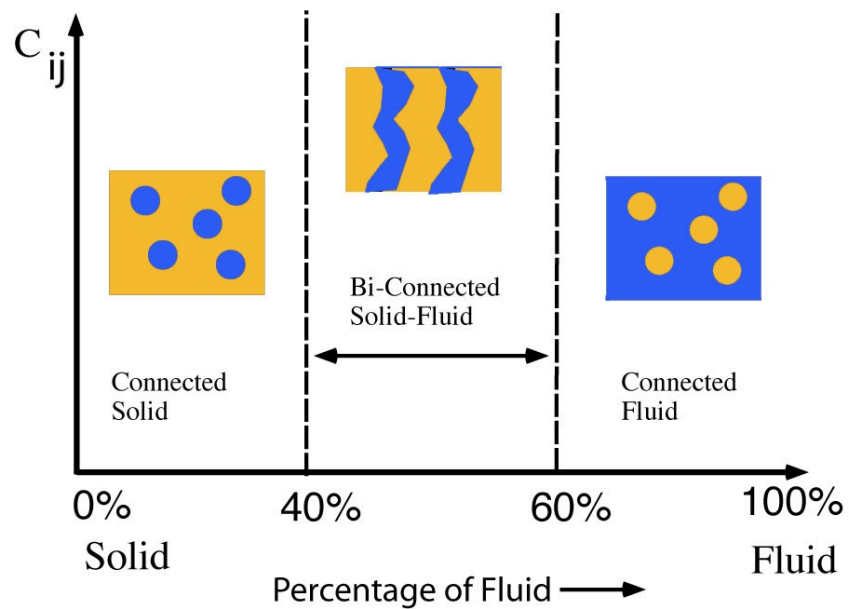
# Hybrid Methods



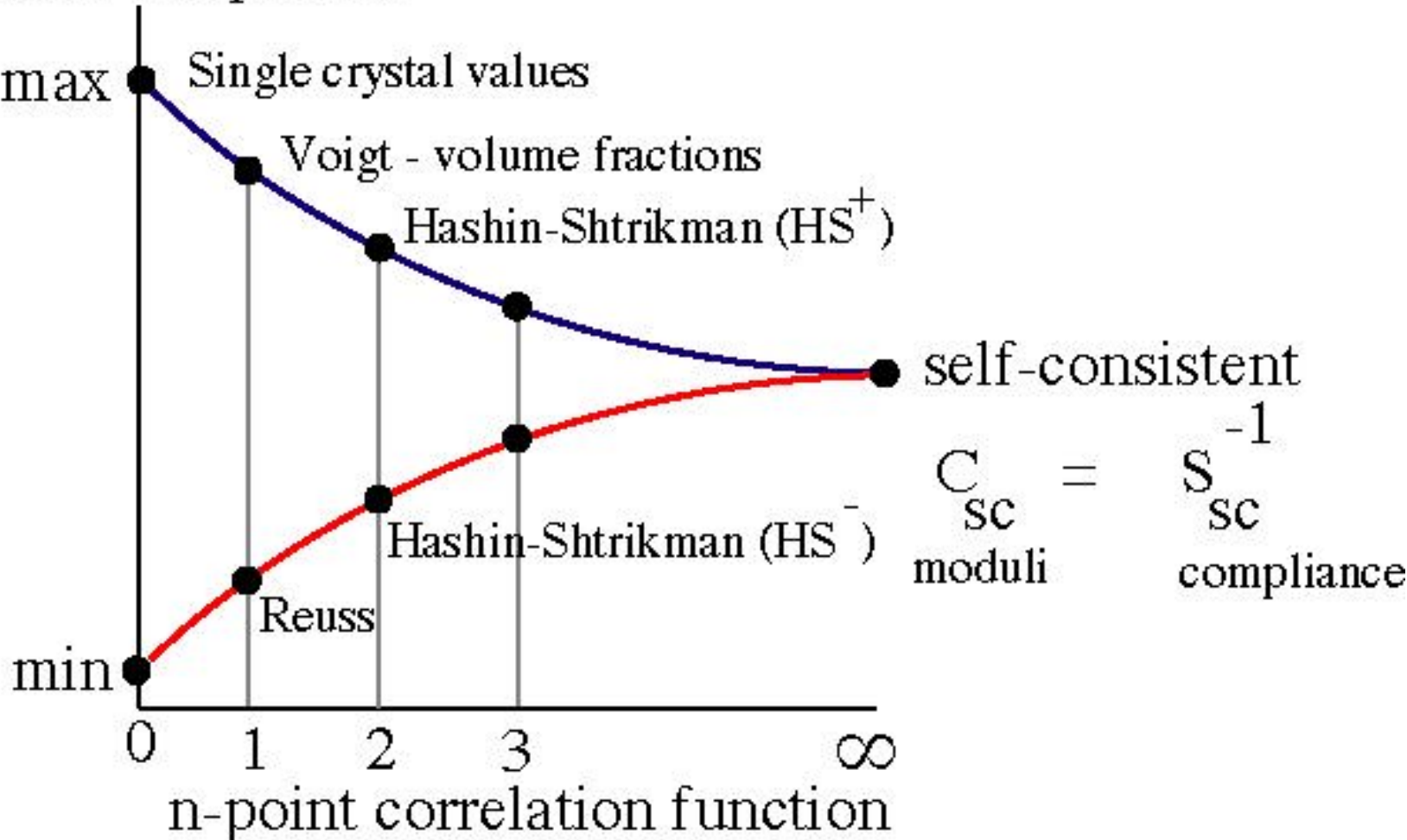
**B** SEM photomicrographs of the fabric of the Williamson Shale- basinward facies graptolite black shale

## Application of Combined SC and DEM : Percolating Fluid

Percolation or Connectivity In SC Method

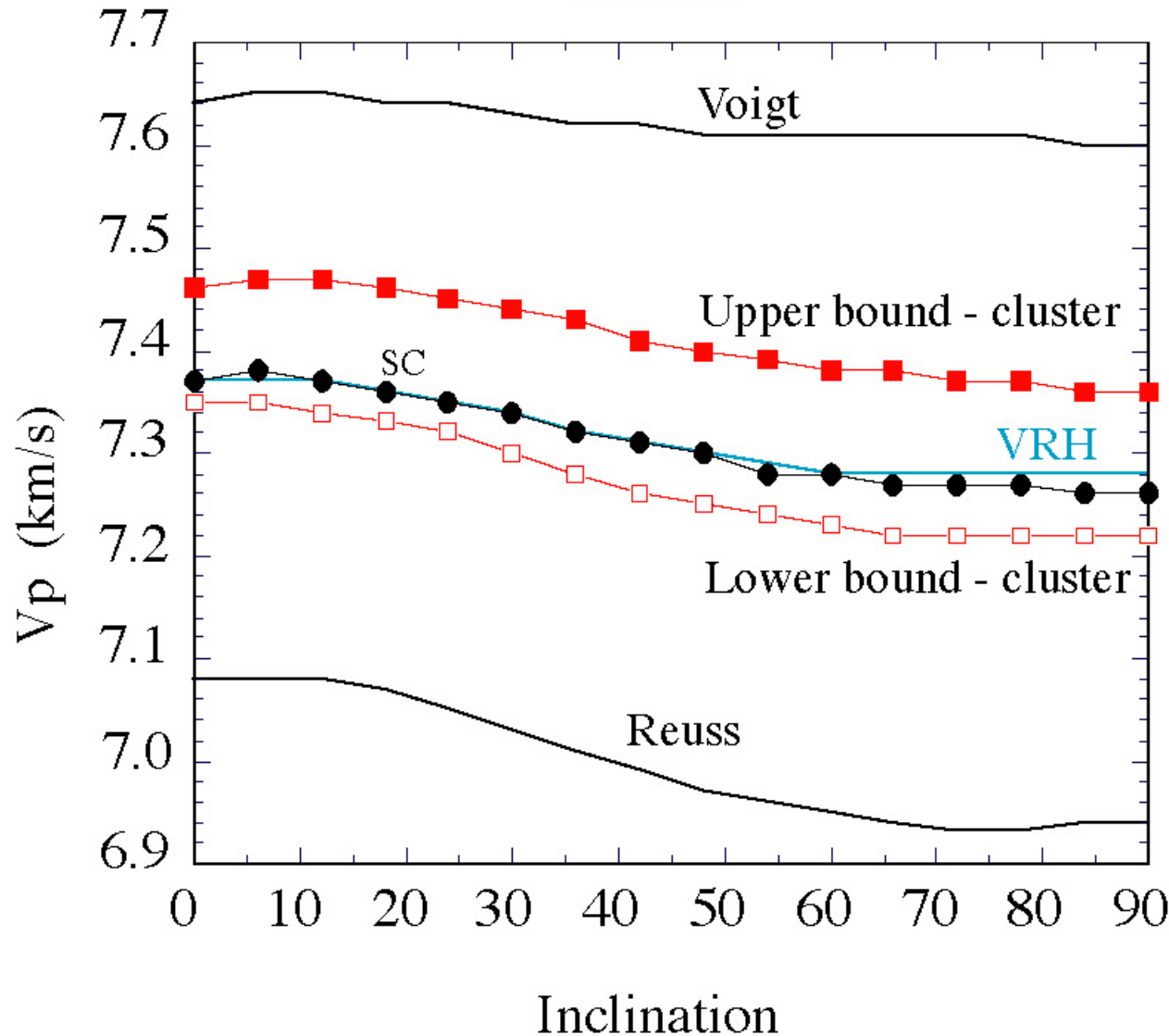


# Elastic Properties

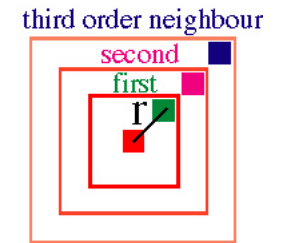
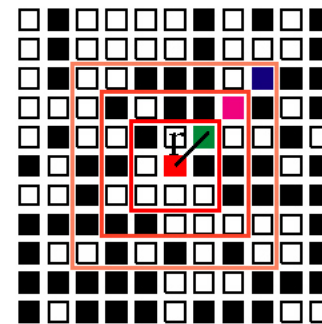


# Cluster method

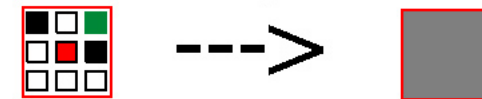
## Gabbro



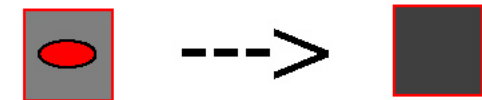
Sobel filter approach - a local cluster model



calculate local background



introduce "inclusion"



move to next point  
sum overall positions

# Quantitative texture analysis and calculation of physical properties

- Orientation  $g$
- Orientation distribution function (O.D.F.)
- Elastic properties of aggregate
- Extrapolation of elastic constants to Pressure & Temperature
- Extrapolation of density to Pressure & Temperature



# Orientation $g$

Orientation of crystals in a polycrystal can be measured by volume diffraction techniques (e.g. X-ray or neutron diffraction) or individual orientation measurements (e.g. U-stage & Optical microscope, electron channeling or EBSD).

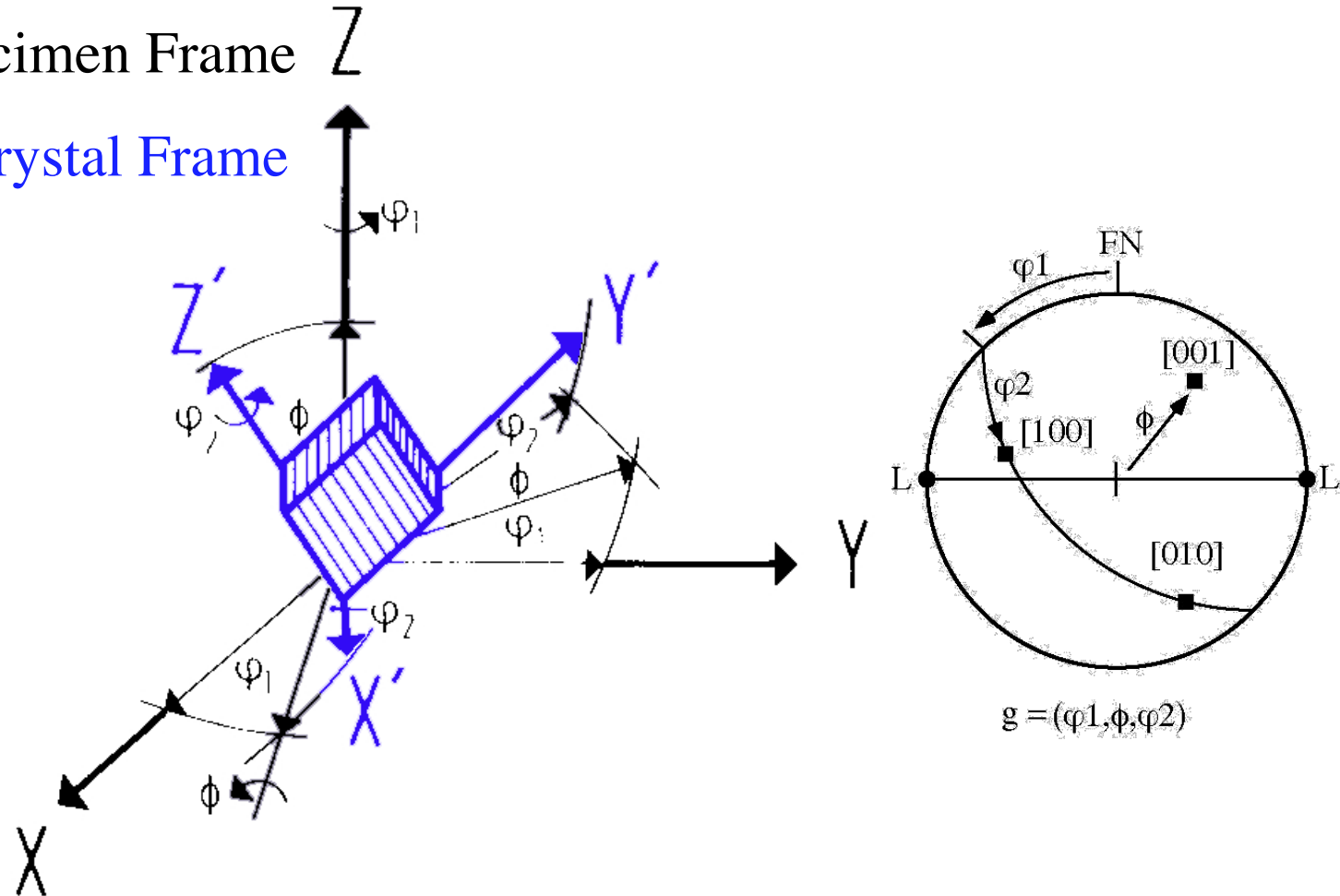
An orientation, often given the letter  $g$  (gefüge = fabric or structure), of a grain or crystal in sample co-ordinates can be described by the rotation (or orientation) Matrix between crystal and sample co-ordinates. In practice it is convenient to describe the rotation by a triplet of Euler angles, for example  $g = \phi_1 \phi \phi_2$  used by Bunge (1982).

*N.B. One should be aware there are many different definitions of Euler angles that are used in the physical sciences, here we will use the definition given by Bunge (1982).*

# Orientation of a crystal defined by 3 Euler angles

XYZ - Specimen Frame  $Z$

$X'Y'Z'$  - Crystal Frame



# Orientation Distribution Function

The orientation distribution function (O.D.F.)  $f(\mathbf{g})$  is defined as the volume fraction of orientations with an orientation in the interval between  $\mathbf{g}$  and  $\mathbf{g}+d\mathbf{g}$  in a space containing all possible orientations given by

$$\Delta V/V = \int f(\mathbf{g}) d\mathbf{g}$$

where  $\Delta V/V$  is the volume fraction of crystals with orientation  $\mathbf{g}$ ,  $f(\mathbf{g})$  is the texture function and  $d\mathbf{g} = 1/8\pi^2 \sin \phi d\phi_1 d\phi_2$  is the volume of the region of integration in orientation space.

The function  $f(\mathbf{g})$  is given in terms of symmetrical generalized spherical harmonics (Bunge,1982) as

$$f(\mathbf{g}) = \sum_{l=0}^{L_{\max}} \sum_{m=1}^{M(l)} \sum_{n=-1}^1 C_l^{mn} T_l^{mn}(\mathbf{g})$$

$C_l^{mn}$  are the coefficients of the series development of the texture function  $f(\mathbf{g})$ ,  $T_l^{mn}(\mathbf{g})$  are the generalised spherical harmonic functions,  $M(l)$  is the number of linearly independent harmonics and  $L_{\max}$  is the maximum degree used in the expansion. For tensor properties of rank 2 or 4 the maximum degree of expansion is  $L_{\max}=2$  or  $L_{\max}=4$  respectively.

# X-ray analysis program !

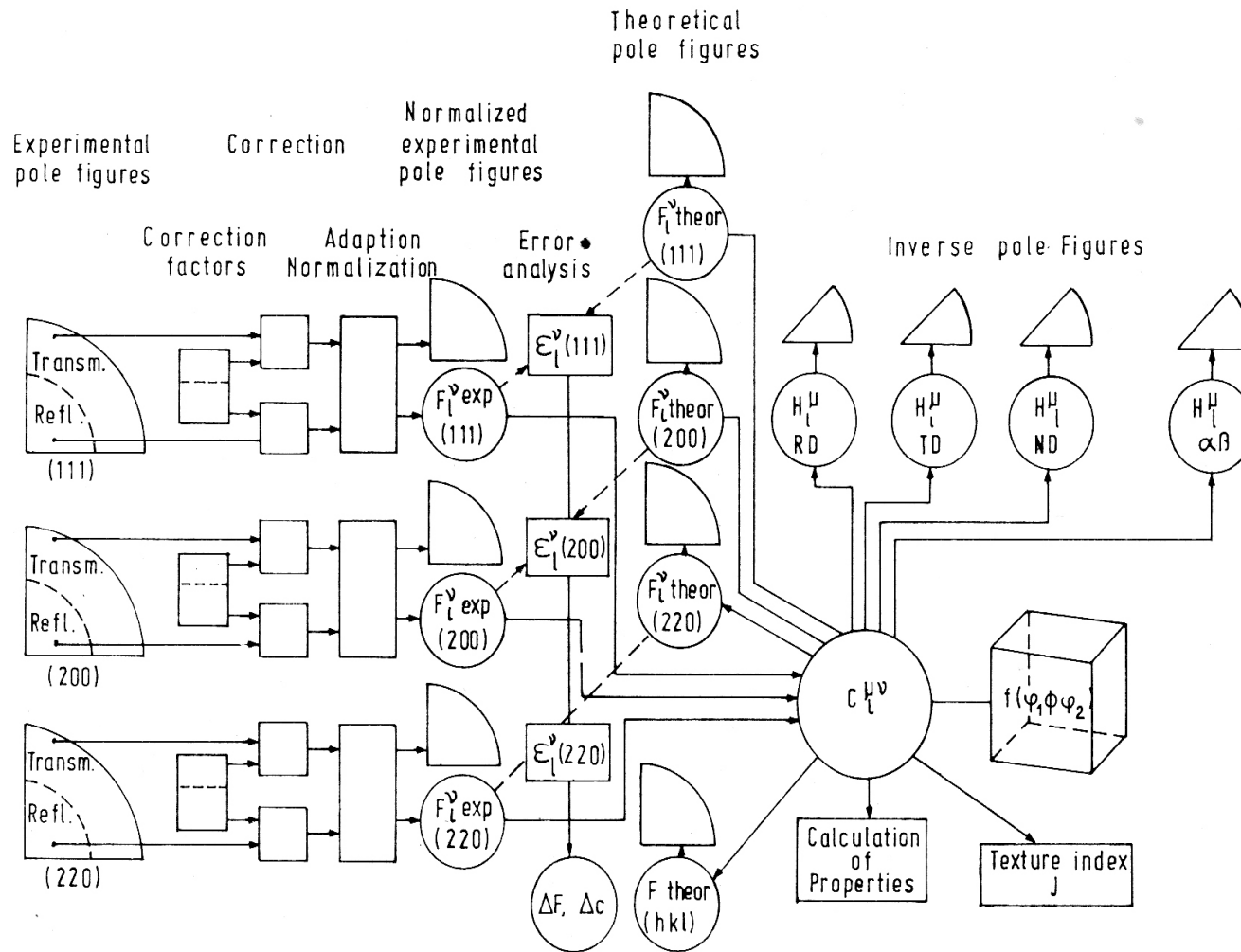
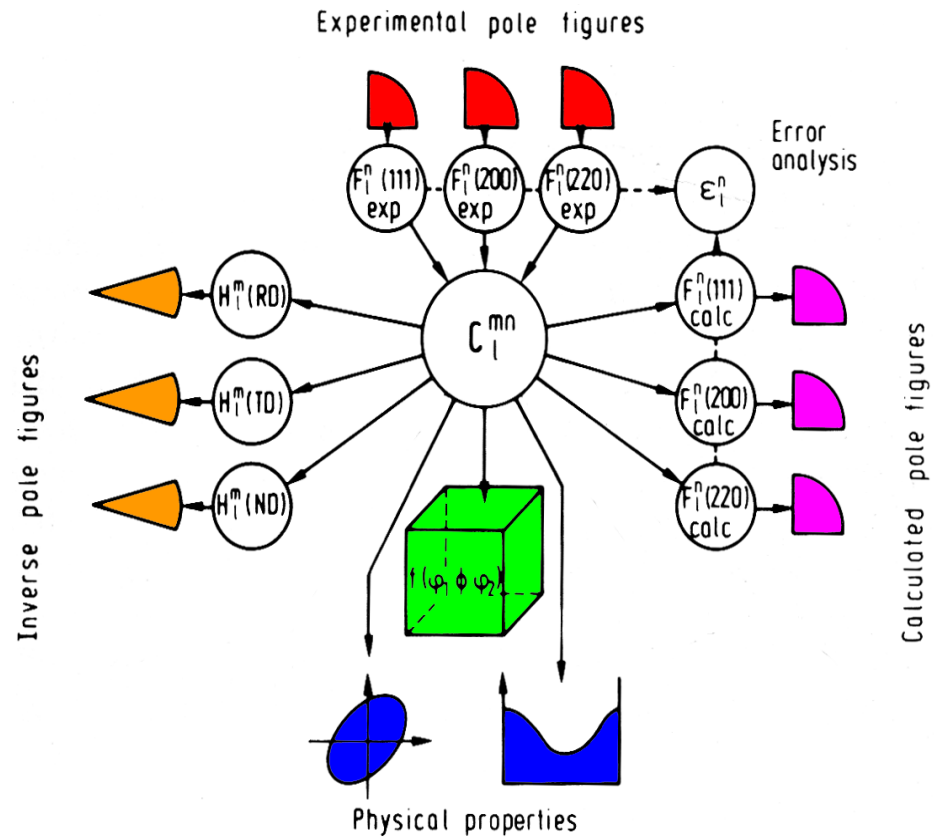


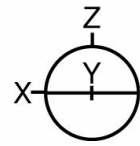
Figure 9.3 Structure of the texture analysis program system<sup>171</sup>

# Harmonic Method

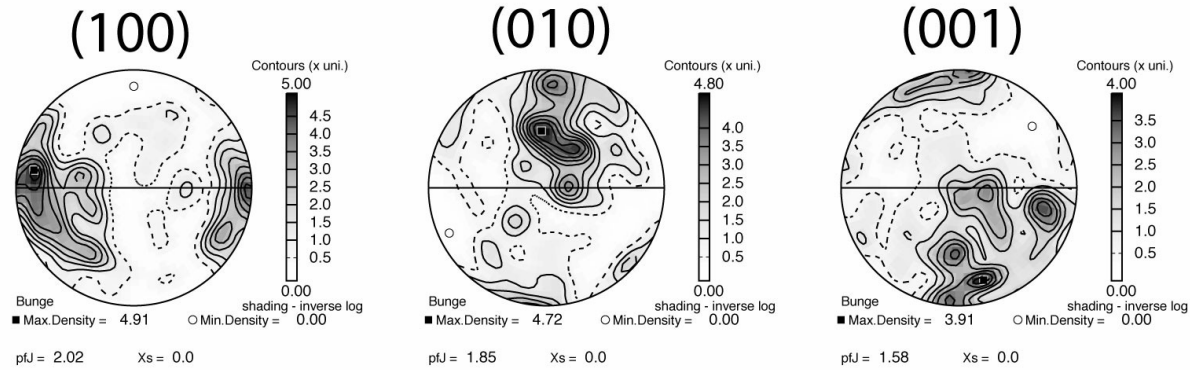


The general scheme of harmonic texture analysis.

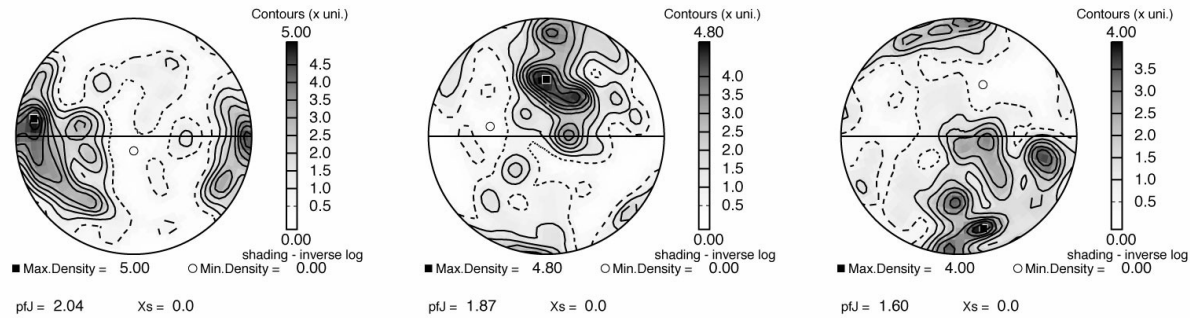
# Pole figure comparison : FB085 Olivine100 grains (U-stage)



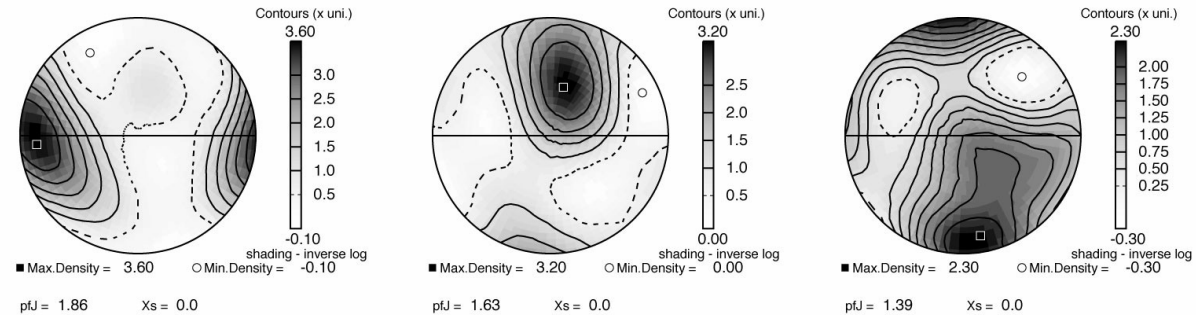
PF2k  
(box counting with  
gaussian half-width = 8.5°)



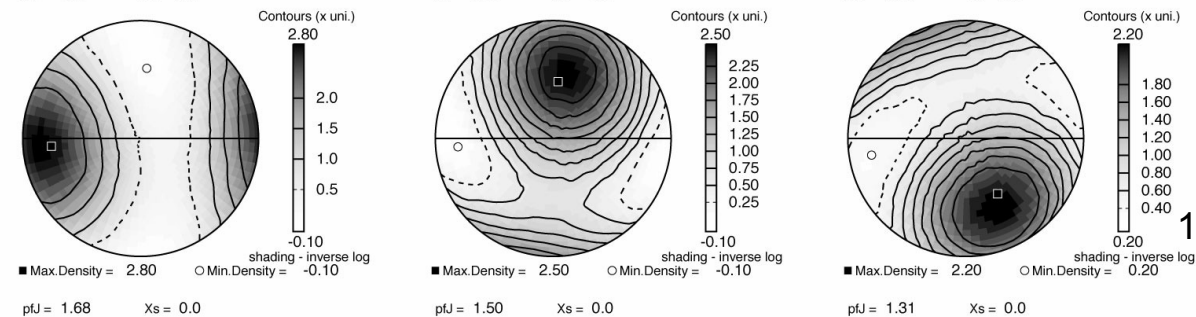
CGRAIN  
(SH Even Coeffs Lmax=22  
by F.Wagner Metz  
gaussian half-width = 10°)



CGRAIN  
(SH Even Coeffs Lmax=4  
by F.Wagner Metz  
gaussian half-width = 10°)



CGRAIN  
(SH Even Coeffs Lmax=2  
by F.Wagner Metz  
gaussian half-width = 10°)



## Cartesian references frames: Euler and Tensor 1

A potential complication is the fact that the Cartesian frame defined by orthogonal crystallographic directions used report elastic tensor of the single crystal may not be the same as those used for Euler angle reference frame used in texture analysis (e.g. MTEX, BearTex ...) or measurement (e.g. EBSD) packages. To account for this difference a rotation may be required to bring the crystallographic frame of tensor into coincidence with the Euler angle frame,

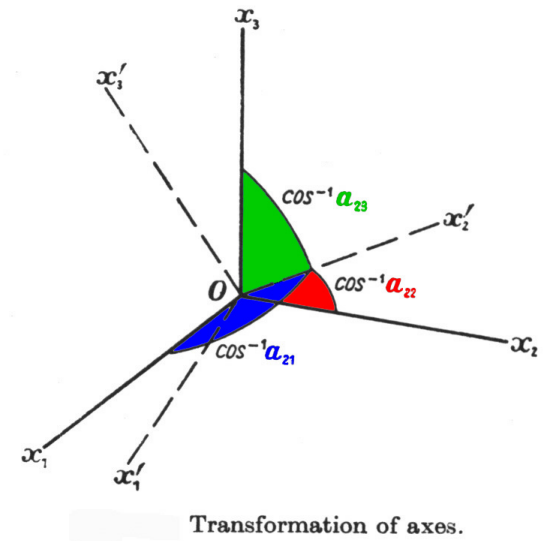
$$C_{ijkl}(\mathbf{g}^E) = T_{ip} \cdot T_{jq} \cdot T_{kr} \cdot T_{lt} C_{pqrt}(\mathbf{g}^T)$$

where  $C_{ijkl}(\mathbf{g}^E)$  is the elastic property in the Euler reference and  $C_{pqrt}(\mathbf{g}^T)$  is the elastic property in the original tensor reference frame, BOTH frames are in crystal co-ordinates.

## Cartesian references frames: Euler and Tensor 2

The orthogonal transformation (or rotation) matrix  $T_{ij}$  is constructed from the angles between the two sets perpendicular crystallographic axes, forming rows and columns of the matrix (see Nye, 1957).

		Tensor Axes			
		Tx	Ty	Tz	
Euler Axes	Ex	[	$T_{11}$	$T_{12}$	$T_{13}$
	Ey		$T_{21}$	$T_{22}$	$T_{23}$
	Ez		$T_{31}$	$T_{32}$	$T_{33}$





# Rotation into specimen co-ordinates

To calculate the seismic properties of a polycrystal, one must evaluate the elastic properties of the aggregate. In the case of an aggregate with a crystallographic fabric, the anisotropy of the elastic properties of the single crystal must be taken into account. For each orientation  $\mathbf{g}$ , the single crystal properties have to be rotated into the specimen co-ordinate frame using the orientation or rotation matrix  $g_{ij}$ ,

$$C_{ijkl}(\mathbf{g}) = g_{ip} \cdot g_{jq} \cdot g_{kr} \cdot g_{lt} C_{pqrt}(\mathbf{g}^E)$$

where  $C_{ijkl}(\mathbf{g})$  is the elastic property in sample co-ordinates,  
 $g_{ij} = g(\phi_1 \phi_2)$  the measured orientation in sample co-ordinates  
and  $C_{pqrt}(\mathbf{g}^E)$  is the elastic property in crystal co-ordinates of the Euler frame (E).

# Elastic properties of aggregate

The elastic properties of the polycrystal may be calculated by integration over all possible orientations of the ODF.

Bunge (1985) has shown that integration is given as:

$$\langle C_{ijkl} \rangle^m = \int g_{ip} \cdot g_{jq} \cdot g_{kr} \cdot g_{lt} \cdot C_{pqrt}^m(\mathbf{g}^E) \cdot f(\mathbf{g}) \, d\mathbf{g} = \int C_{ijkl}^m(\mathbf{g}) \cdot f(\mathbf{g}) \, d\mathbf{g}$$

where  $\langle C_{ijkl} \rangle^m$  is the elastic properties of the aggregate of mineral  $m$ .

Alternatively it may be determined by simple summation of individual orientation measurements (e.g. U-stage or EBSD),

$$\langle C_{ijkl} \rangle^m = \sum g_{ip} \cdot g_{jq} \cdot g_{kr} \cdot g_{lt} \cdot C_{pqrt}^m(\mathbf{g}^E) \cdot V(\mathbf{g}) = \sum C_{ijkl}^m(\mathbf{g}) \cdot V(\mathbf{g})$$

Where  $V(\mathbf{g})$  is the volume fraction of the grain in orientation  $\mathbf{g}$ .

# Christoffel equation & Tensor

The final step is the calculation of the three seismic phase velocities by the solution of the Christoffel equation,

$$| \langle C_{ijkl} \rangle^{\text{Voigt}} \mathbf{n}_j \mathbf{n}_l - \rho V^2 \delta_{ik} | = | \mathbf{T}_{ik} - \rho V^2 \delta_{ik} | = 0$$

where  $\mathbf{n}$  is the plane wave propagation direction,

$\mathbf{T}_{ik}$  is the Christoffel tensor

$\rho$  is the density of the rock,

$\delta_{ik}$  is the Kronecker delta and

the three values of  $V$  are the three seismic phase velocities.

The calculation is of the three seismic phase velocities from the Christoffel tensor

$$\mathbf{T}_{ik} = \langle C_{ijkl} \rangle^{\text{Voigt}} \mathbf{n}_j \mathbf{n}_l$$

where the three eigenvalues  $E_i$  of the symmetric Christoffel tensor ( $\mathbf{T}_{ik}$ ) are related to three seismic phase velocities  $V_i$  (qP,qS1,qS2) by  $V_i = (E_i/\rho)^{1/2}$ , and qS1 > qS2.

# Extrapolation elastic constants to PT

To calculate the elastic constants at pressures and temperatures of Earth's interior the single crystal elastic constants are extrapolated to pressure and temperature using the following relationship:

$$C_{ij}(PT) = C_{ij}(P_o T_o) + (dC_{ij}/dP) \cdot (P-P_o) + 1/2 (d^2C_{ij}/dP^2) \cdot (P-P_o)^2 + (dC_{ij}/dT) \cdot (T-T_o)$$

where  $C_{ij}(PT)$  are the elastic constants at pressure  $P$  and temperature  $T$ ,  
 $C_{ij}(P_o T_o)$  the elastic constants at reference pressure  $P_o = 0.1$  MPa and  
temperature  $T_o = 25$  °C;  
 $dC_{ij}/dP$  is the first order pressure derivative  
and  $dC_{ij}/dT$  is the first order temperature derivative.

The second order pressure derivatives  $d^2C_{ij}/dP^2$  are available for  
an increasing number of mantle minerals and first order temperature  
derivatives seem to adequately describe the temperature dependence of most minerals.

# Mg-Perovskite $C_{ij}$ at lower mantle PT

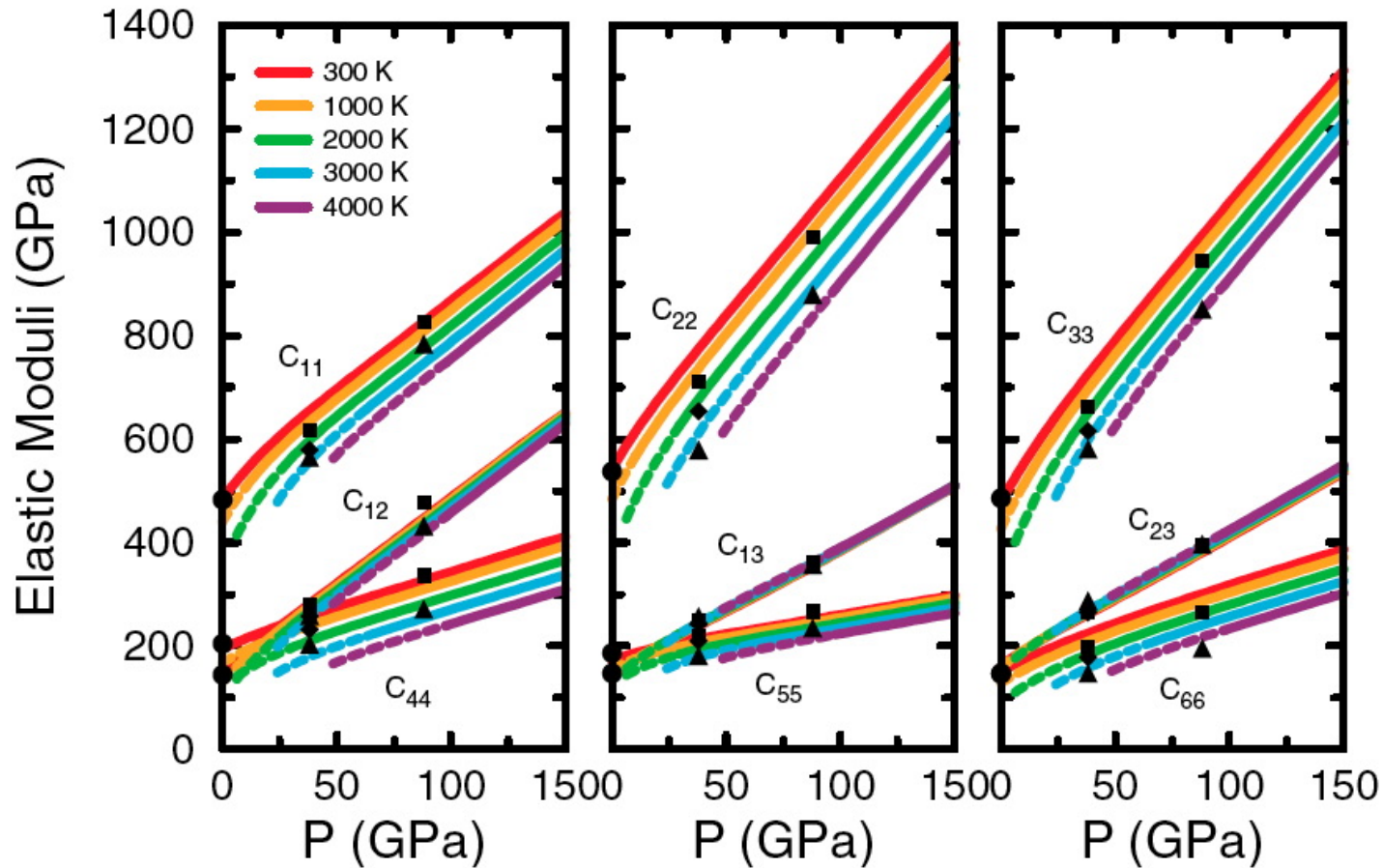


FIG. 1 (color). Pressure dependence of the adiabatic elastic constants of MgSiO<sub>3</sub>. The line's thickness represents uncertainties caused mainly by the use of the local density approximation (LDA) (see [11]). Full (dashed) lines correspond to results within (outside) the  $(P, T)$  regime of validity of the QHA. Measurements are represented by full circles at 0 GPa [5]. Full symbols at  $P > 0$  GPa are the results of Oganov *et al.* [4] at 38 and 88 GPa (squares: 1500 K, diamonds: 2500 K, up triangles: 3500 K).

# Extrapolation of density to PT

The seismic velocities also depend on the density of the minerals at pressure and temperature which can be calculated using an appropriate equation of state. The Murnaghan equation of state derived from finite strain is sufficiently accurate at moderate compression (Knittle, 1995) of the upper mantle and leads to the following expression for density as a function of pressure,

$$\rho(P) = \rho_o(1+(K'/K).(P-P_o))^{1/K'}$$

where  $K$  is bulk modulus,  $K' = dK/dP$  the pressure derivative of  $K$ ,  $\rho_o$  is the density at reference pressure  $P_o$  and temperature  $T_o$ .  
For temperature the density varies as

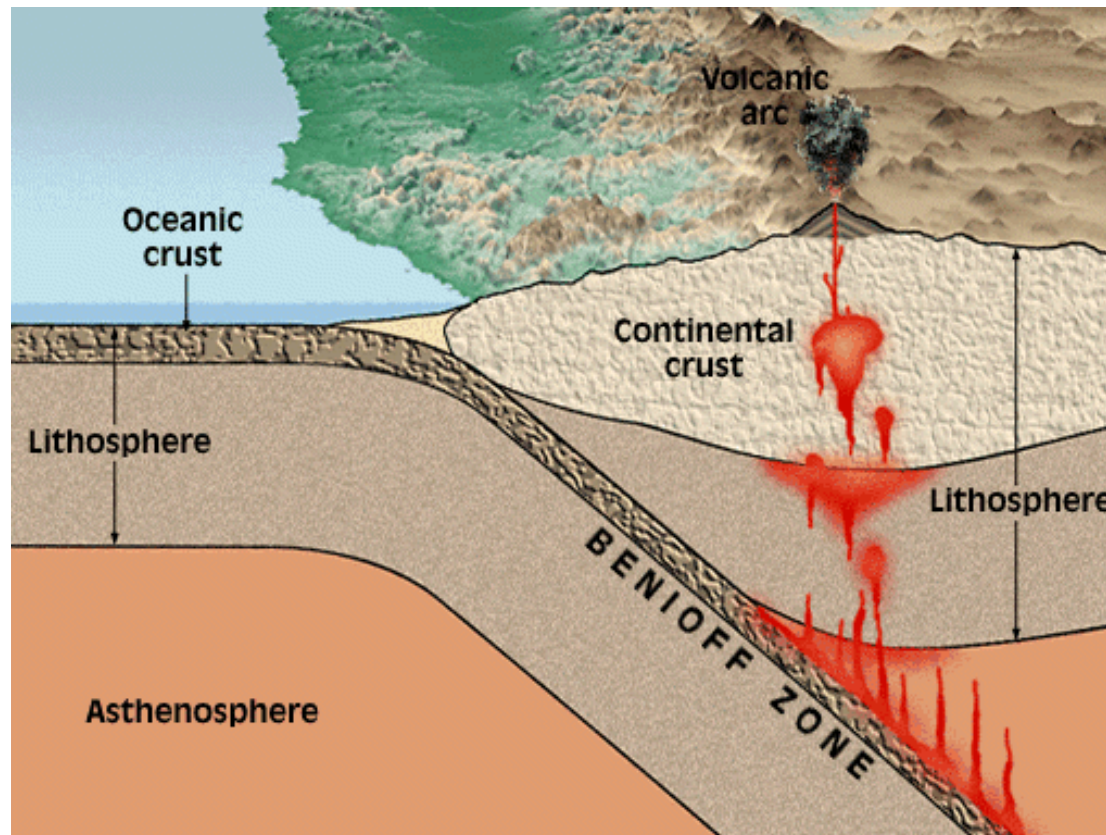
$$\rho(T) = \rho_o[1 - \int \alpha_v(T) dT] = \rho_o[1 - \alpha_{av} (T-T_o)]$$

where  $\alpha_v(T) = 1/V(\partial V/\partial T)$  is the volume thermal expansion coefficient as a function of temperature and  $\alpha_{av}$  is an average value of thermal expansion which is constant over the temperature range (Fei, 1995). According to Watt (1988) an error of less 0.4% on the P and S velocity results from using  $\alpha_{av}$  to 1100K for MgO.  
For temperatures and pressures of the mantle, the density can be described by,

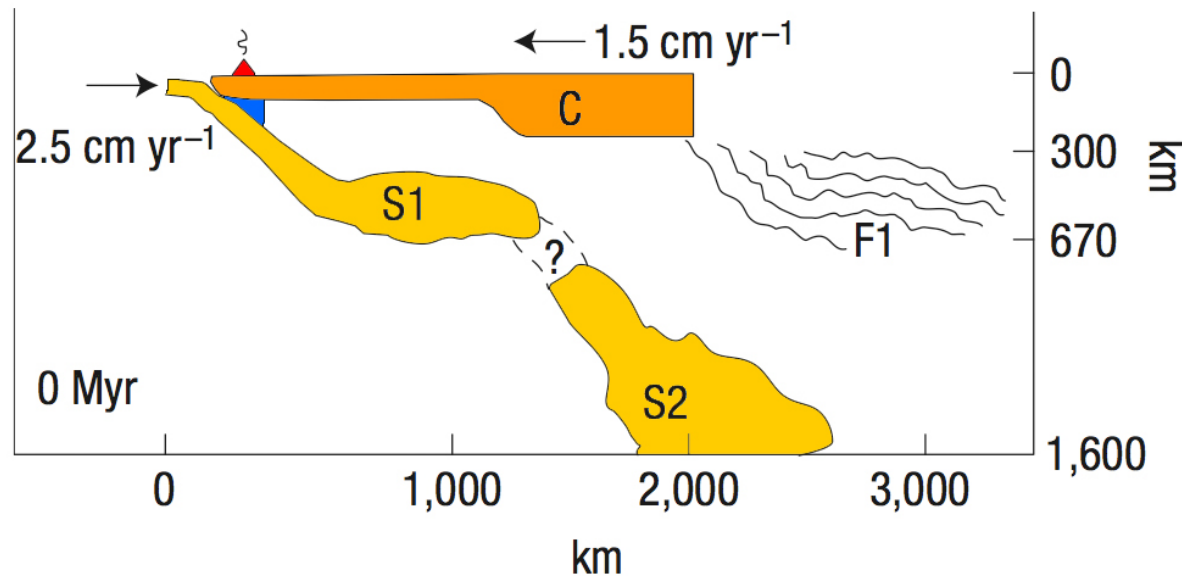
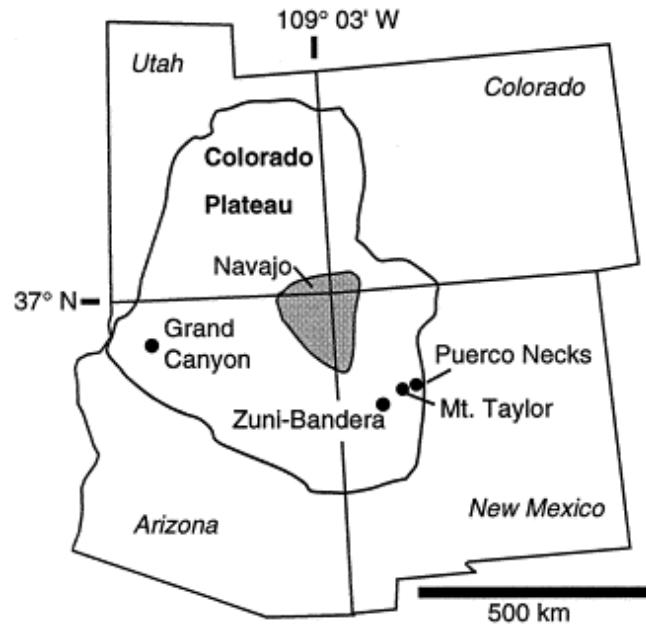
$$\rho(P,T) = \rho_o \{ (1+(K'/K).(P-P_o))^{1/K'} [1 - \alpha_{av} (T-T_o)] \}$$

# Topotactic relationships between olivine and antigorite: Implications for subduction zone seismology and geodynamics

David Mainprice, **Francoise Boudier**, **Alain Baronnet** and **Roland Pellenq**



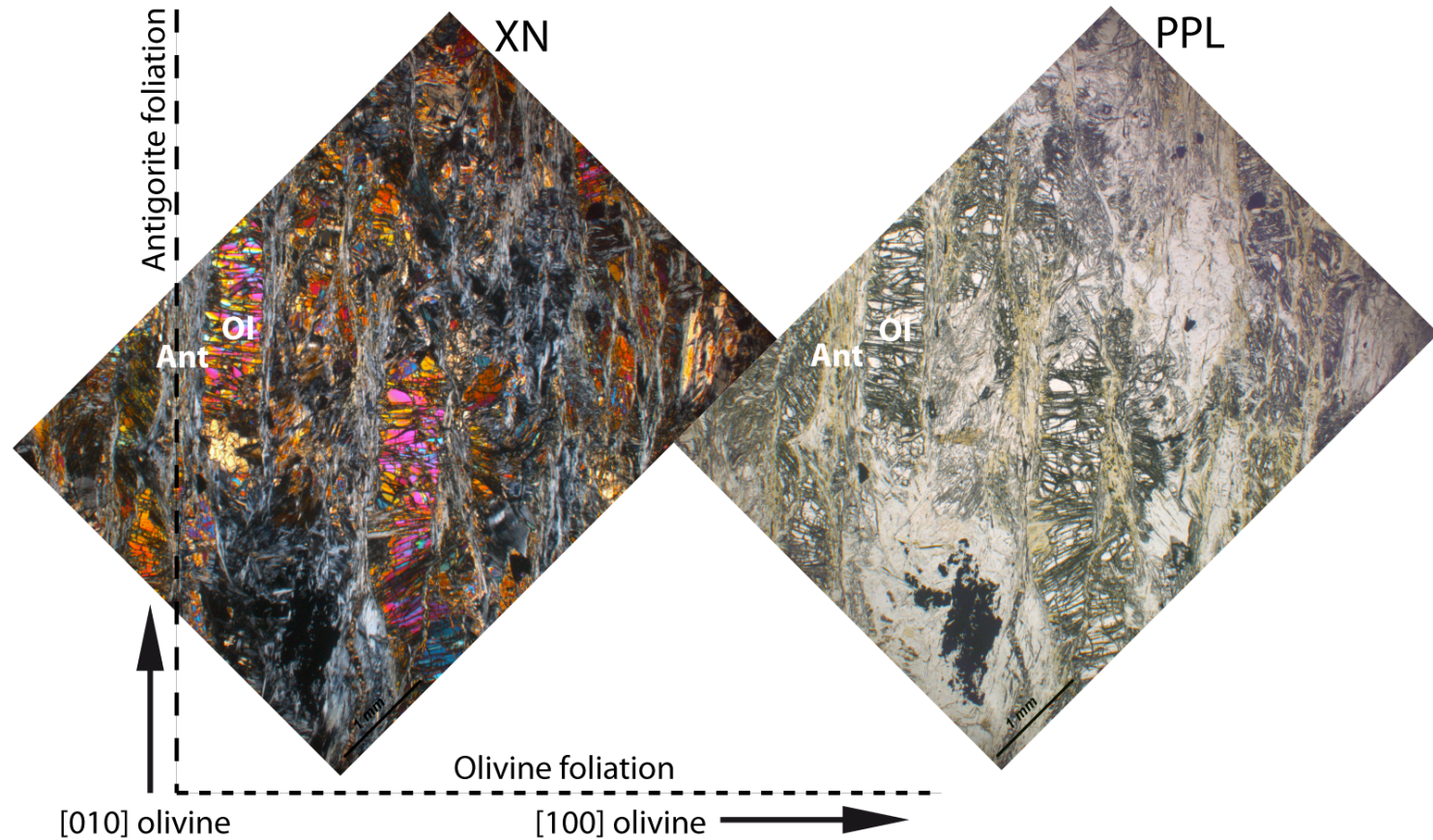
# Moses Rock Utah - Farallon plate subduction



Sigloch et al.08 Nature Geosciences

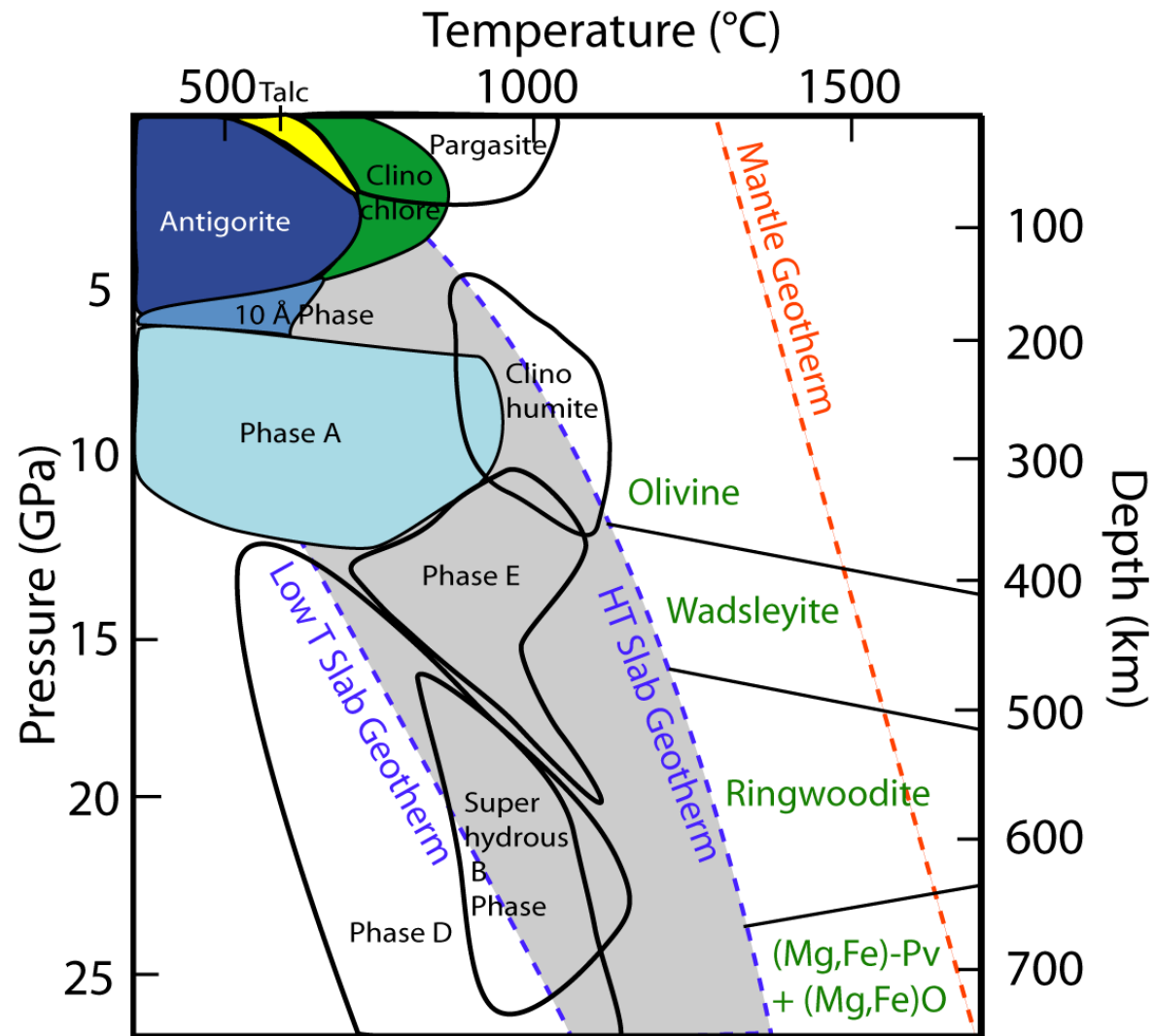


# A sample from a subduction zone

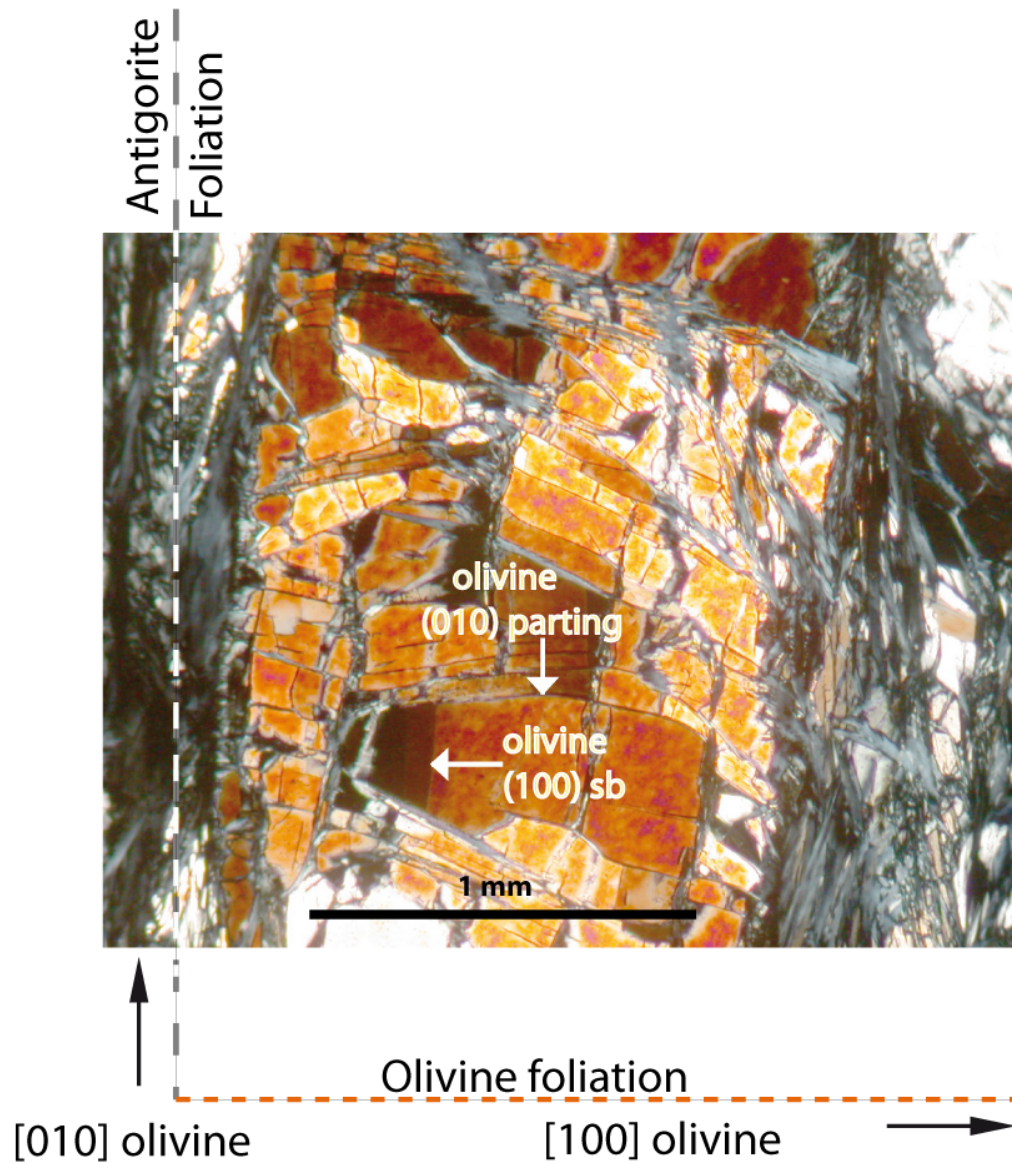


**Olivine - Antigorite Schist  
from Moses Rock, Utah**

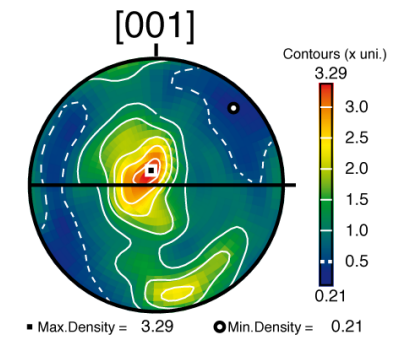
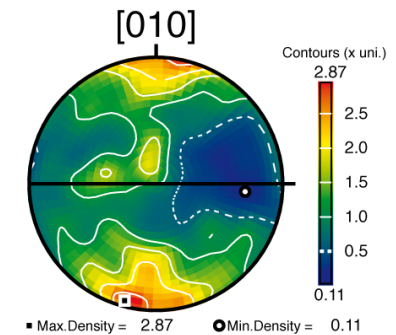
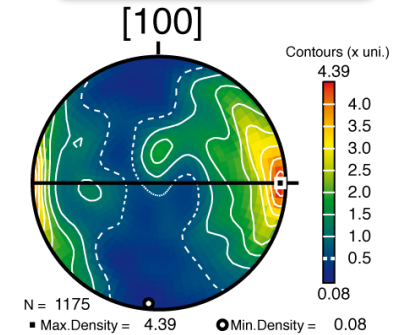
# Phase diagram & Geotherm



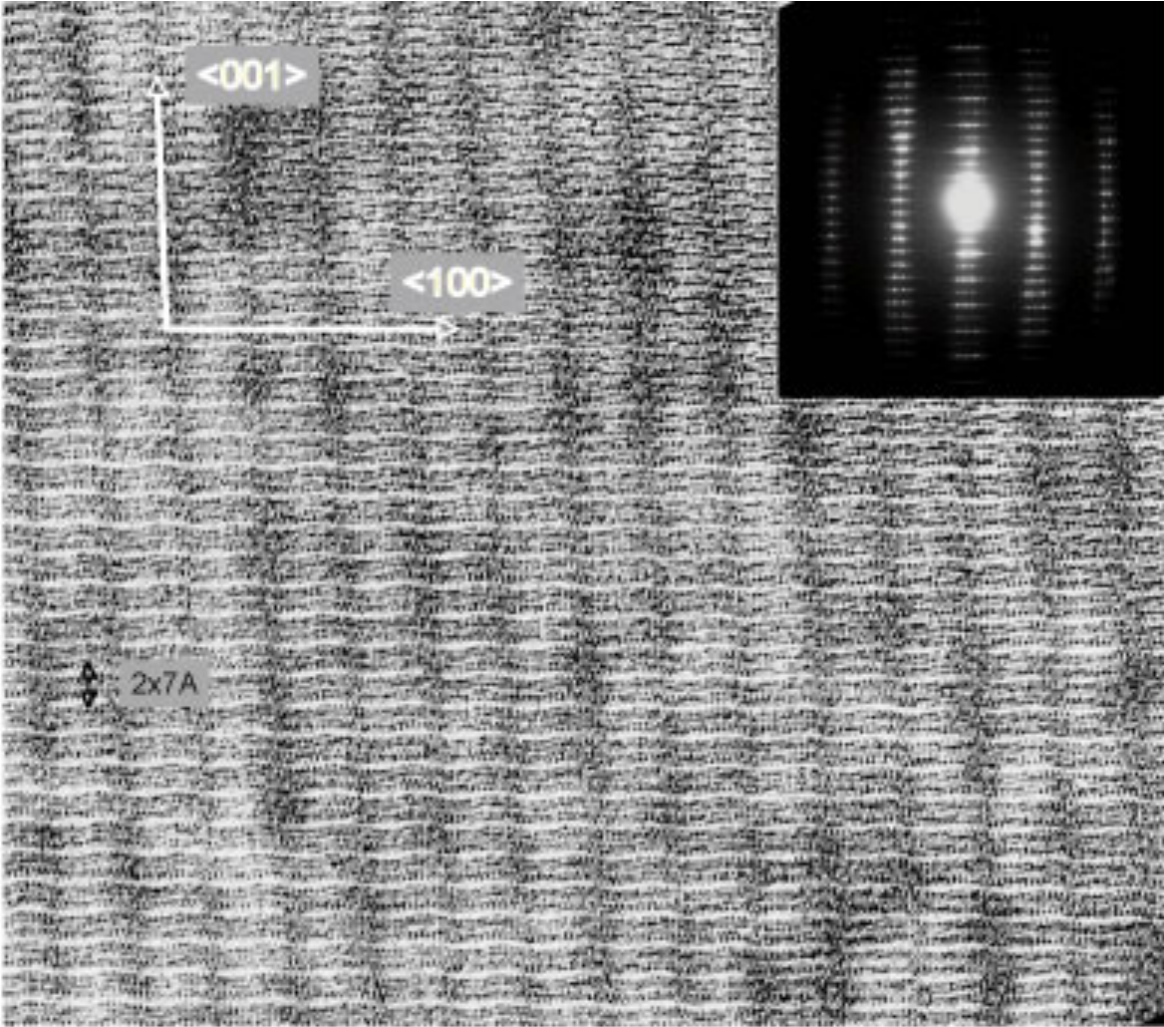
# Relationship olivine microstructure & CPO



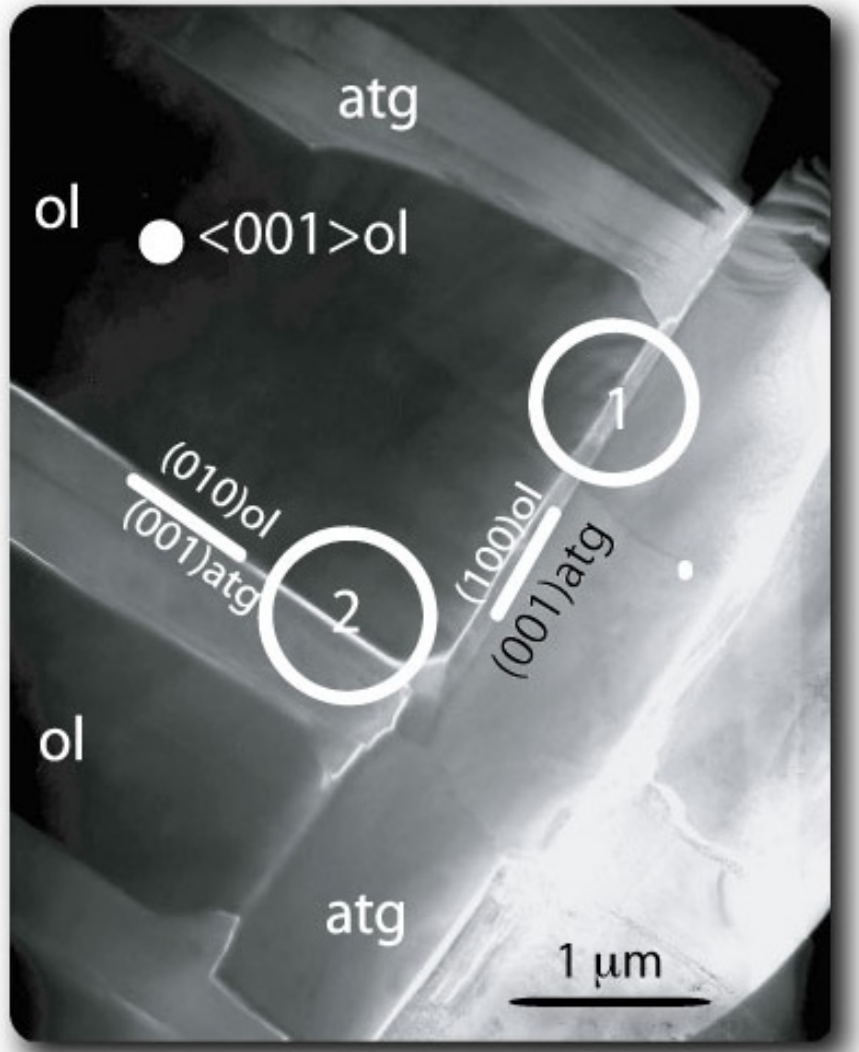
## EBSD data



HRTEM  
View along  $[010]_{atg}$



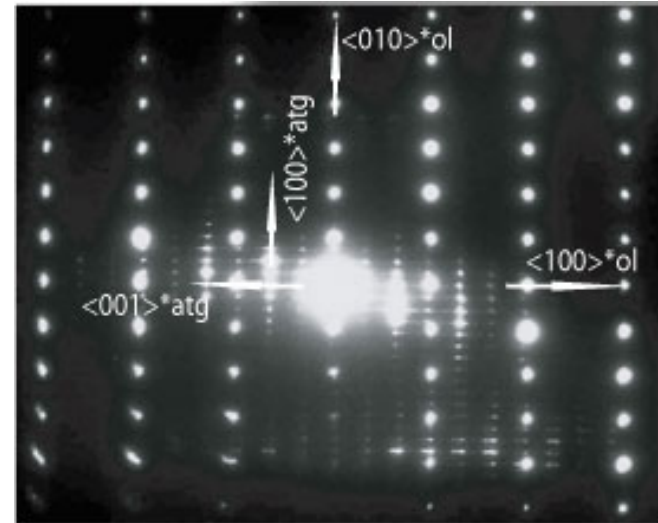
CTEM data



relation 1

$$(100)_{\text{Olivine}} \parallel (001)_{\text{Antigorite}}$$

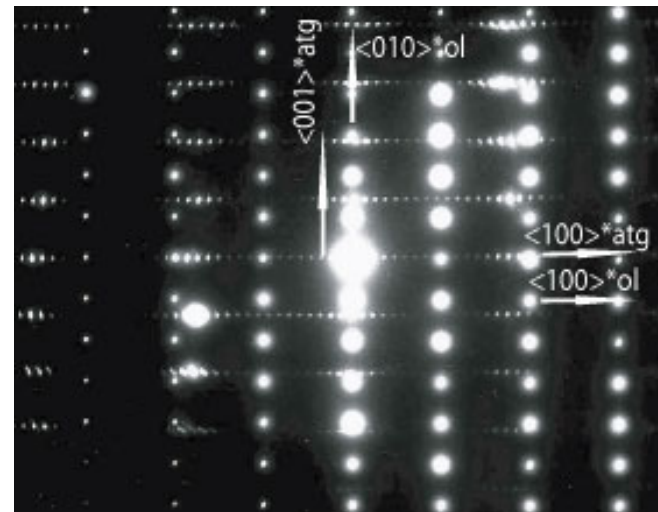
$$[001]_{\text{Olivine}} \parallel [010]_{\text{Antigorite}}$$



relation 2

$$(010)_{\text{Olivine}} \parallel (001)_{\text{Antigorite}}$$

$$[001]_{\text{Olivine}} \parallel [010]_{\text{Antigorite}}$$



# Summary of Olivine - Antigorite Schist

- Hydrated Moses Rock peridotite has classical "A-type" olivine CPO, typical of "dry" olivine, but sample contains hydrated minerals antigorite, chlorite ... Do we need "wet" olivine to explain the subduction zone dynamics ?

- Antigorite has topotactic relationships with olivine, two types of relations are observed:

(1)  $(100)_{ol} \parallel (001)_{atg}$  with  $[001]_{ol} \parallel [010]_{atg}$   
and (2)  $(010)_{ol} \parallel (001)_{atg}$  with  $[001]_{ol} \parallel [010]_{atg}$

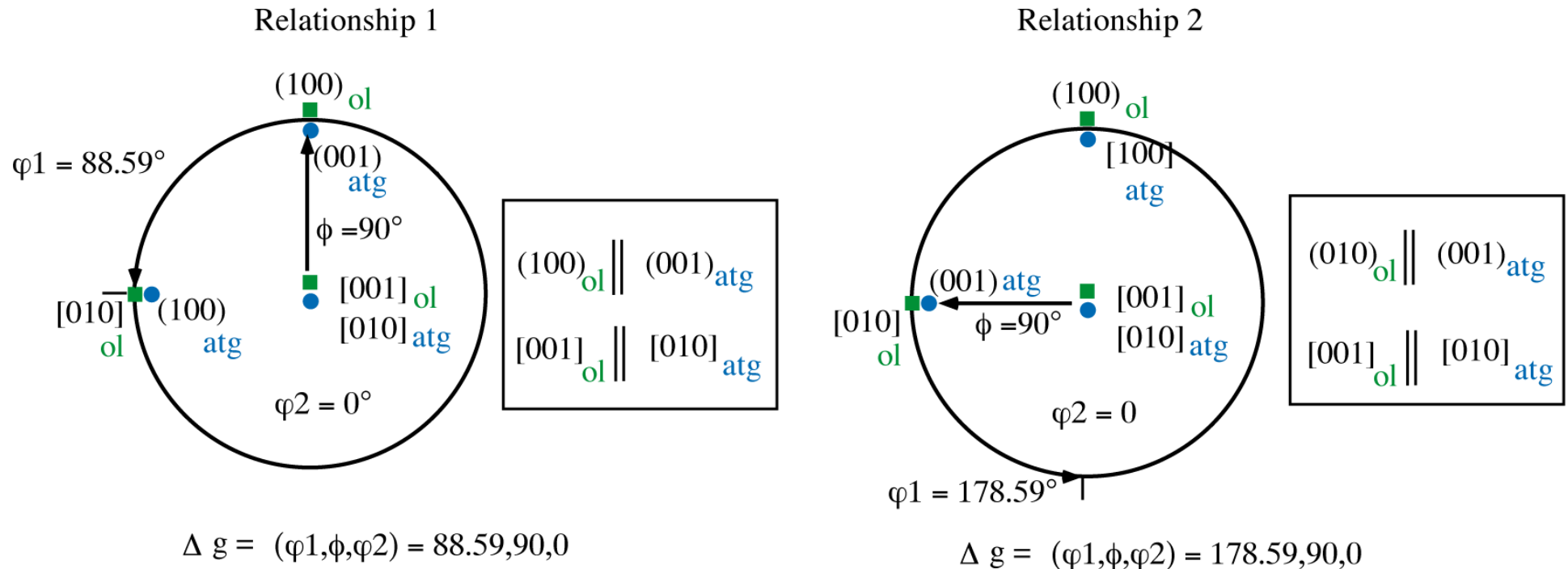
- Dominant antigorite orientation (1) develops a schistosity perpendicular to the mantle flow fabric of the olivine aggregate

- Develops coeval with hydrous fluid induced micro-cracking controlled by the olivine crystallography:

(010) parting, (100) dislocation walls, fast hydrogen diffusion  $\parallel [100]$

# Crystallographic relationships

Misorientation  $\Delta g$  between lattices : olivine - antigorite



Antigorite has topotactic relationships with olivine,  
two types of relations are observed:

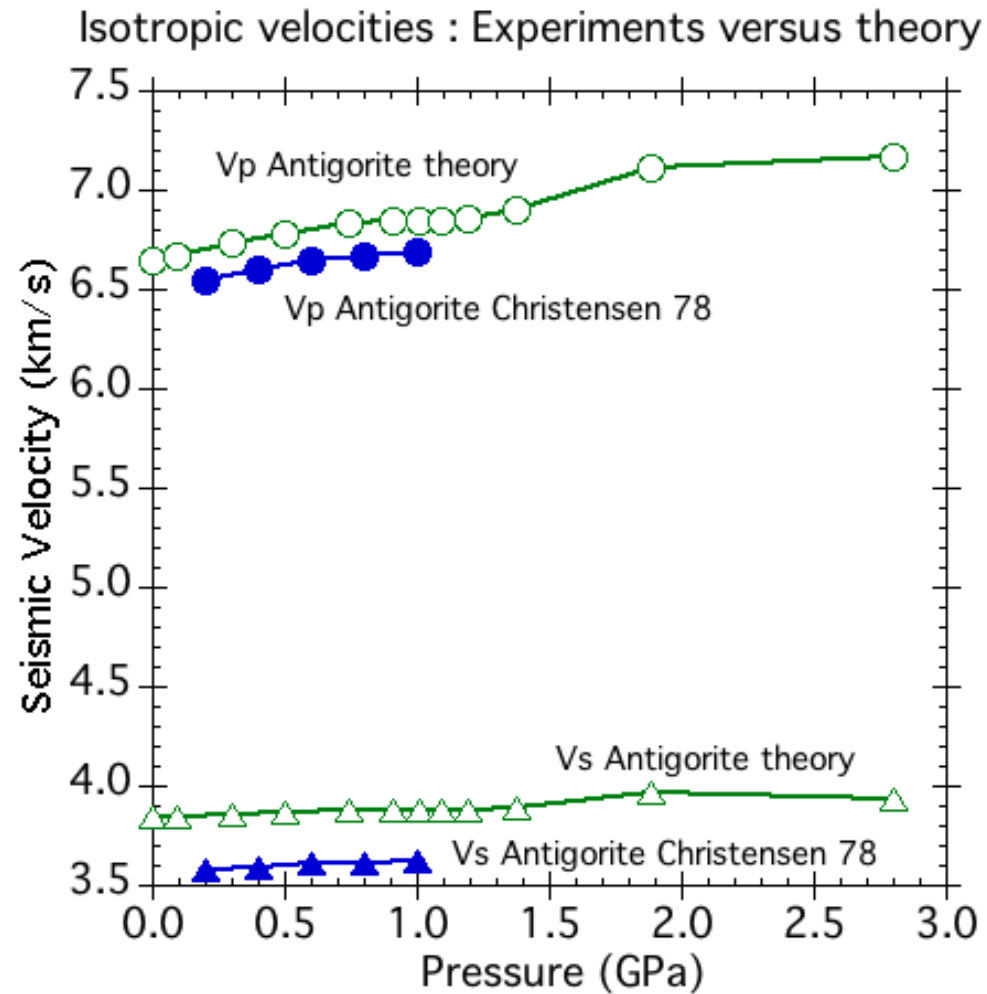
- (1)  $(100)_{ol} \parallel (001)_{atg}$  with  $[001]_{ol} \parallel [010]_{atg}$   
and (2)  $(010)_{ol} \parallel (001)_{atg}$  with  $[001]_{ol} \parallel [010]_{atg}$

# Calculate antigorite orientation from olivine

$$g_{n=1 \text{ to } 4}^{\text{Antigorite}} = \Delta g \cdot S_n^{\text{Olivine}} \cdot g^{\text{Olivine}}$$

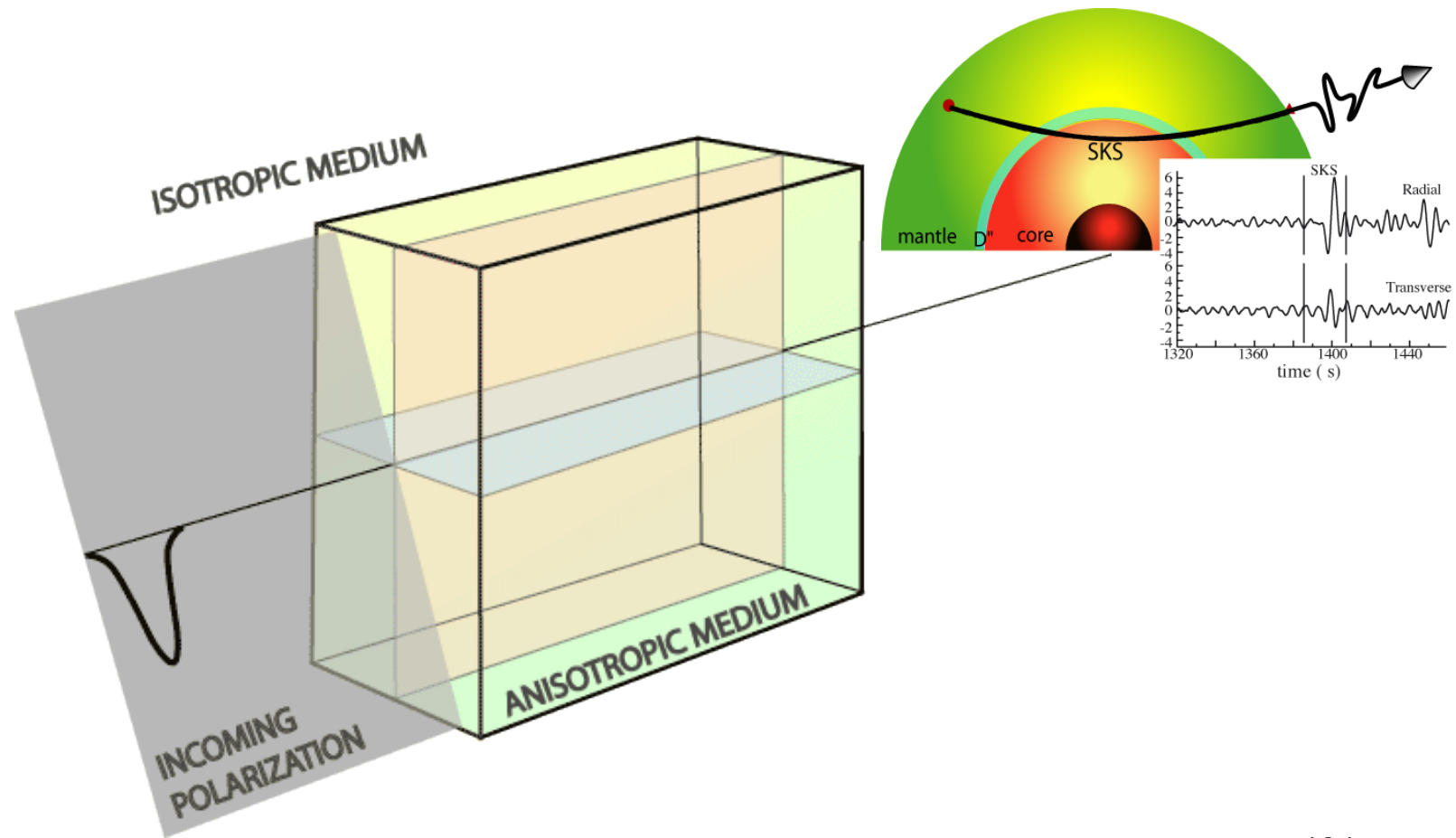


# SEISMIC IMPLICATIONS

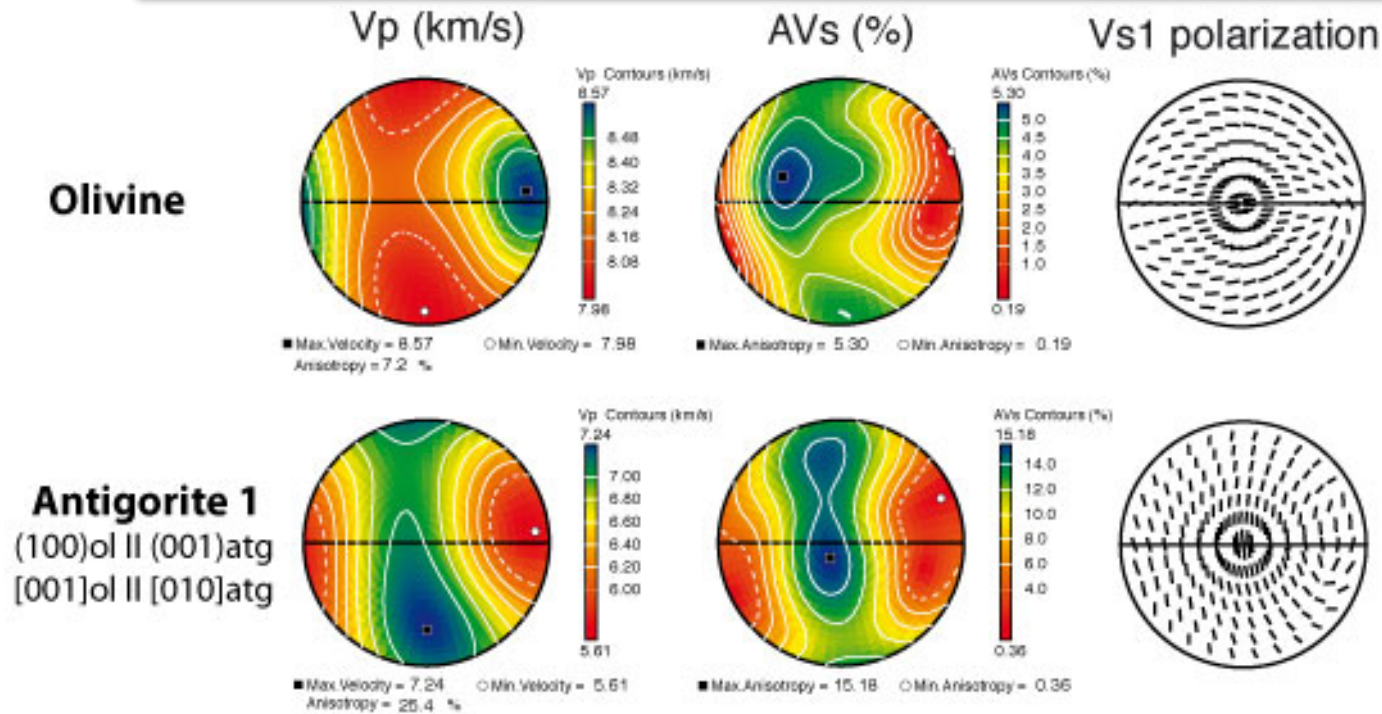


- a) Experimentally Lizardite and Antigorite have different velocities (Christensen,2004,1978,1966).
- b) Agreement of Antigorite elastic constants of Pellenq et al. (GULP) with experimental data is good.

# Shear Wave Splitting



# ANTIGORITE MODEL for Moses rock schist



$$\text{Thickness} = 100 \cdot dt \cdot \langle Vs \rangle / Avs$$

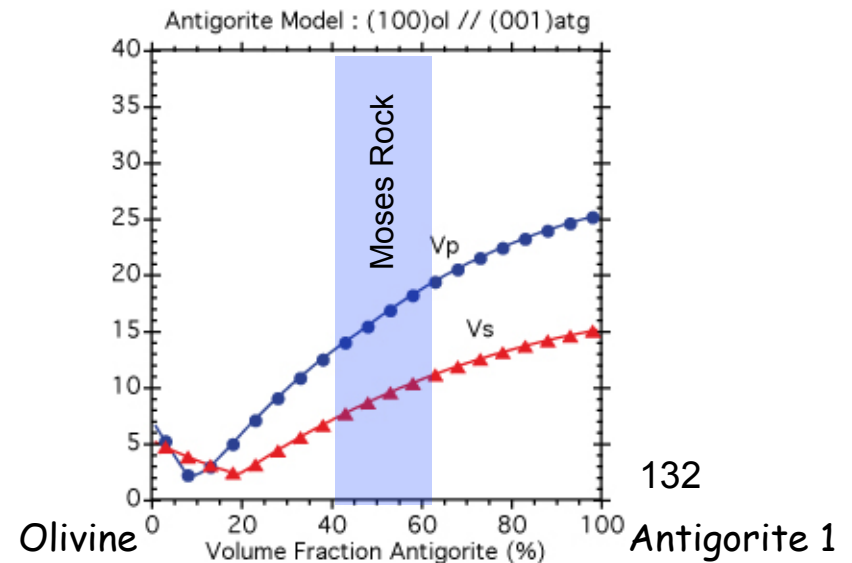
$$dt = 0.25 - 1.0s$$

$$\langle Vs \rangle = 4.19 \text{ km/s}$$

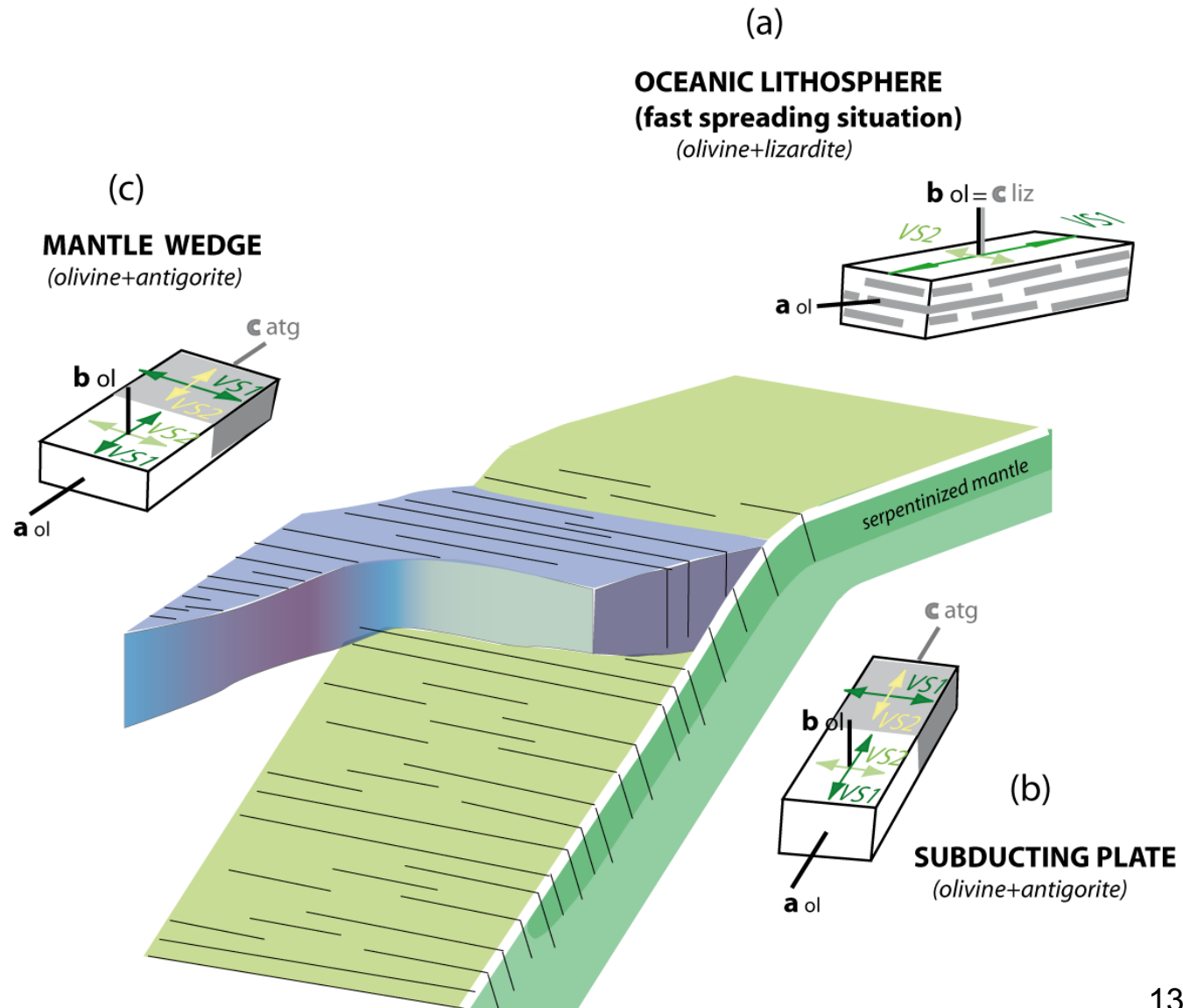
$$Avs = 10\%$$

$$\text{Thickness (dt 1.0s)} = 41.9 \text{ km}$$

$$\text{(dt=0.25s)} = 10.5 \text{ km}$$

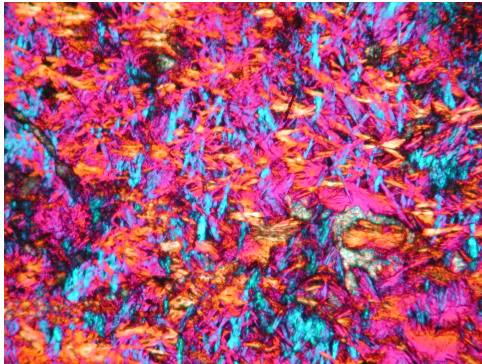


# Overview of Serpentine contribution to anisotropy

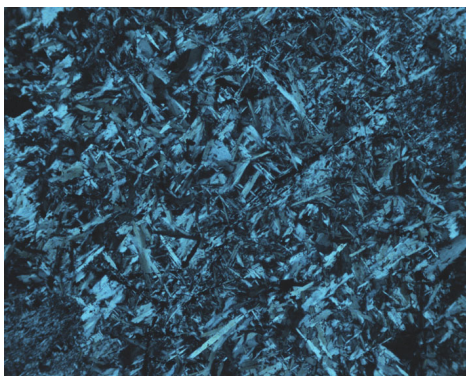


# ANTIGORITE MODEL : mixture of Antigorite 1 & 2 (100% antigorite)

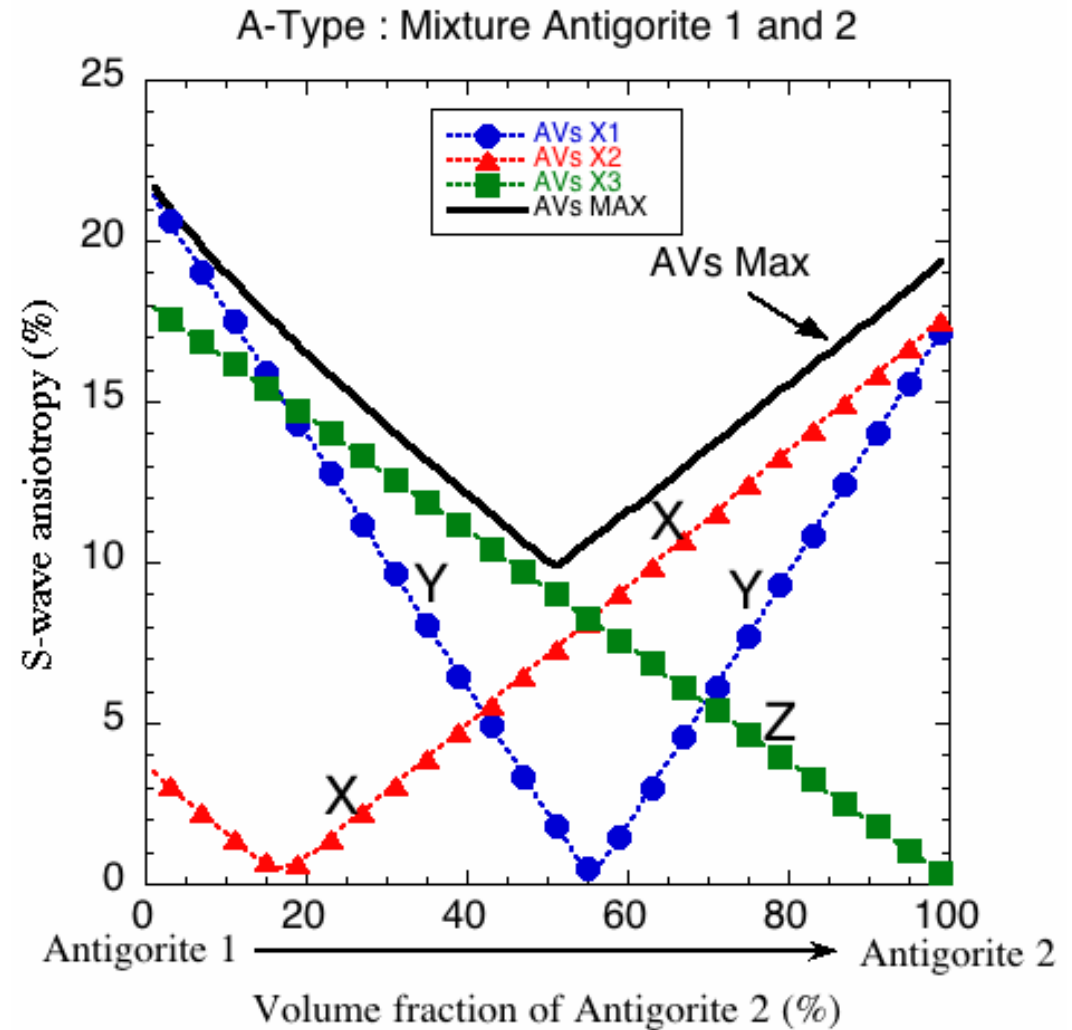
Z - vertical S-waves  
X & Y in horizontal plane



Orange - Atg(001) EW  
Blue - Atg(001) NS



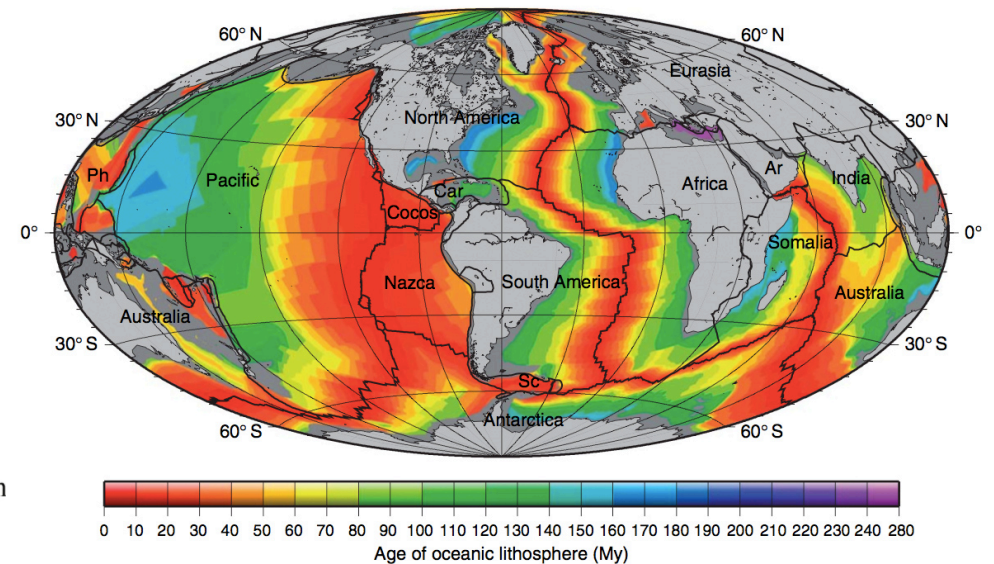
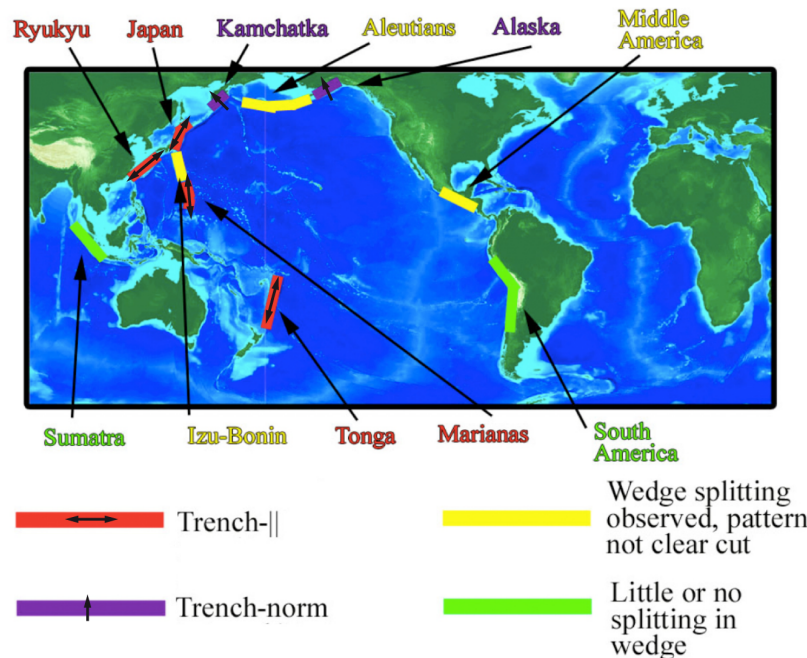
N.I. Christensen (per.comm.)



100% Antigorite type 2 (100)ol || (001)atg would result in ZERO vertical (Z direction) anisotropy

# Mantle wedge S-wave anisotropy

- a) Trench parallel in western Pacific (Japan, Ryuku, Marianas & Tonga).
- b) Trench normal in northern Pacific (Kamchatka & Alaska).
- c) *Low or no anisotropy in Aleutians, Central and South America = warm lithosphere*



After Long & Silver (2008) Science

Müller et al (1997)

# Summary of Seismic results

- Using type (1) (100)ol || (001)atg antigorite topotactic relationships with olivine produces TRENCH PARALLEL anisotropy for "A-type" olivine CPO, (as suggested by Moses Rock sample), with the correct order of magnitude to explain seismic delay times of 1s in the Mantle wedge.
- Using type (2) (010)ol || (001)atg antigorite topotactic relationships with olivine produces ZERO vertical anisotropy for "A-type" olivine at 100% antigorite, which may explain subduction of young (warm) slabs on western Mexico and South America with no clear anisotropy and probably high Antigorite volume fractions.
- Why type (1) or type (2) may be favoured is not yet determined... But you can have both !